Superconducting SrSnP with Strong Sn–P Antibonding Interaction: Is the Sn Atom Single or Mixed Valent?

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Supporting Information

ABSTRACT: The large single crystals of SrSnP were prepared using Sn self-flux method. The superconductivity in the tetragonal SrSnP is observed with the critical temperature of \(~2.3\,K\). The results of a crystallographic analysis, superconducting characterization, and theoretical assessment of tetragonal SrSnP are presented. The SrSnP crystallizes in the CaGaN structure type with space group \(P4/nmm\) (S.G. 129, Pearson symbol \(tP\)) according to the single-crystal X-ray diffraction characterization. A combination of magnetic susceptibility, resistivity, and heat capacity measurements confirms the bulk superconductivity with \(T_c = 2.3(1)\,K\) in SrSnP. According to the X-ray photoelectron spectroscopy measurement, the assignments of Sr\(^{2+}\) and P\(^{3-}\) are consistent with the chemical valence electron balance principles. Moreover, it is highly likely that Sn atom has only one unusual oxidation state. First-principles calculations indicate the bands around Fermi level are hybridized among Sr d, Sn p, and P p orbitals. The strong Sn–P and Sr–P interactions pose as keys to stabilize the crystallographic structure and induce the superconductivity, respectively. The physics-based electronic and phononic calculations are consistent with the molecular viewpoint. After inclusion of the spin–orbit coupling into the calculation, the band degeneracies at \(\Gamma\)-point in the first Brillouin zone split into two bands, which yield to the van Hove singularities around Fermi level.

INTRODUCTION

The discovery of new materials that possess novel properties, especially superconductivity, is a long-standing topic for solid-state material scientists. There are various approaches to new superconductors, one of which is to assume that superconductivity appears in the structural families. Many superconductors, including high-\(T_c\) cuprates, boride carbides, and iron-based pnictides, crystallize in the layered structure, which is widely accepted to be one of the most critical factors to induce the superconductivity. The discovery of these new structural motifs hosting the superconductivity often leads to the rapid development of new superconductors. Both perovskites and the Fe-based superconductor structure types are good examples. On the other hand, suppressing the charge disproportionation is another strategy to find new superconductors. For example, the Mott insulator, BaBiO\(_2\), can be chemically formulated as Ba[Bi\(^{3+}\)\(_{0.5}\)Bi\(^{5+}\)\(_{0.5}\)]O\(_3\). After K is doped on Ba or Pb on Bi sites, the mixed valence state will vanish; meanwhile, the superconductivity will appear. In 2014, Hosono’s group investigated the superconductivity in SnAs and focused on the valence state of Sn in SnAs to prove that Sn has the single valence state, likely Sn\(^{3+}\)(5s\(^0\)) and Sn\(^{2+}\)(5s\(^2\)). Later, in 2017, Tokura’s group studied the superconducting SnP under high pressure and emphasized the critical role of the nominal valence Sn\(^{3+}\)(5s\(^2\)) in stabilizing the phase and inducing superconductivity.

Considering these anomalous oxidation states in Sn-based superconductors, we focused on ternary equiatomic SrSnP with an even shorter Sn–P bonding distance (\(~2.46\,\AA\)
compared to $\sim 2.642$ Å Sn–P in SnP at 2.2 GPa. A previously reported antiferromagnetic EuSnP has the isotype structure to SrSnP. The crystal structure can be described as built up from tin atoms that form nets of puckered four-membered rings by means of long Sn–Sn bonds ($\sim 3.26$ Å). Assuming the oxidation states of Sr to be $+2$ and P to be $-3$, each Sn will be left with only three valence electrons for forming four equivalent Sn–Sn bonds in the puckered net, if all bonds are assumed to be normal two-center two-electron ($2c$–$2e$) bonds. The observation of four long Sn–Sn bonds then implies some sort of fractional bonding in which the bonds are weaker than normal $2c$–$2e$ bonds. This will lead to an interesting question regarding the Sn valence: does the Sn atom have a single valence as we proposed above or mixed valences, like Sn$^0$ ($5s^25p^2$) and Sn$^{+2}$ ($5s^2$)? Therefore, can the unusual single valent Sn be related to some unique physical properties, such as superconductivity?

With these structural, valent, and superconducting features in mind, we report herein a thorough structural characterization, investigation of the superconducting properties, and theoretical electronic and phononic structures of SrSnP, with an emphasis on the Sn valence and interplay between superconductivity and chemical bonding. We discovered the superconductivity in SrSnP with $T_c = 2.3(1)$ K for the first time. What’s more, Sn has only one valent state, not mixed valent states like Sn$^0$ ($5s^25p^2$) and Sn$^{+2}$ ($5s^2$). A combination of chemistry and physics interpretations confirm both Sn–P and Sn–P bonding interactions are critical for the structural stability and superconductivity.

**EXPERIMENTAL SECTION**

**Single-Crystal Growth of SrSnP.** The single crystals of SrSnP were grown from Sn flux. Elemental strontium (99%, granules, Beantown Chemical), red phosphorus (99%, ~100 meshes, Beantown Chemical), and tin granules (99.5%, Alfa Aesar) were put into an alumina crucible with the molar ratio of 1:1:10. The crucible was subsequently sealed into an evacuated ($10^{-5}$ Torr) quartz tube. A treatment was performed at a rate of $30$ °C/h to $600$ °C and held for 12 h to avoid the explosion of red phosphorus. After that, the mixture was heated up to $1050$ °C and annealing for 24 h followed by slowly cooling to $600$ °C at a rate of $3$ °C/h. Excess Sn flux was removed by centrifuging the samples. Single crystals of SrSnP with a size of ~0.2 × 2 × 2 mm$^3$ tetragonal shape were obtained as shown in the inset of Figure 1. All operations were carried out in the Ar-protected glovebox because SrSnP is extremely unstable toward decomposition in air or moisture.

**Phase Identification.** A Rigaku MiniFlex 600 powder X-ray diffractometer (XRD) equipped with Cu Kα radiation ($\lambda = 1.5406$ Å, Ge monochromator) was used to examine the phase information. Due to the fact that the SrSnP single crystal is extremely air- and moisture-sensitive, we were only able to perform a short scan which the Bragg angle ranged from $10^\circ$ to $70^\circ$ in a step of $0.010^\circ$ at a rate of $0.05$°/min. The powder XRD pattern shown in Figure 1 was matched with the calculated pattern generated from the single-crystal X-ray data and smoothed using Savitzky-Golay smoothing filters.

**Structure Determination.** Multiple pieces of single crystals (~20 × 40 × 40 μm$^3$) were picked up to perform the routine structural examination. The structure was determined using a Bruker Apex II diffractometer equipped with Mo radiation ($\lambda = 0.71073$ Å) at low temperatures (215 and 100 K). Glycerol and liquid nitrogen gas were used to protect the sample which was mounted on a Kapton loop. Seven different angles were chosen to take the measurement with an exposure time of 15 s per frame and the scanning $2\theta$ width of 0.5°. The direct methods and full-matrix least-squares on $F^2$ models with SHELXTL package were employed to solve the crystal structure. Data acquisition was made via Bruker SMART software with the corrections on Lorentz and polarization effect. Numerical absorption corrections using face-index model were approached by XPREP.

**Scanning Electron Microscopy (SEM).** The determination of the chemical composition and stoichiometry were performed using a high-vacuum scanning electron microscope (JSM-6610 LV) and energy-dispersive spectroscopy (EDS). Samples were mounted on the carbon tape in the argon-protected glovebox and quickly moved into the SEM chamber, and then evacuated immediately. Multiple points and areas were taken to test on the same single crystal of SrSnP with an accelerating voltage of 15 kV and the collecting time of 100 s for each point/area via TEAM EDAX software.

**Superconducting Properties Measurements.** The superconducting Quantum Interference Device (SQUID) (Quantum Design, Inc.) magnetometer is used to measure the temperature-dependent susceptibility under different magnetic fields with the temperature range of 1.8–10 K. The susceptibility is defined as $\chi = M/H$. Here, $M$ is the magnetization in units of emu, and $H$ is the magnetic field applied in Oe. Both field cooling (FC) and zero field cooling (ZFC) methods were used to test susceptibility. Resistivity and heat capacity measurements were performed using a quantum design physical property measurement system (PPMS) between 1.8 and 300 K with and without applied fields. Heat capacity was measured using a standard relaxation method.

**X-ray Photoelectron Spectroscopy (XPS).** A Kratos AXIS 165 XPS/AES equipped with standard Mg/Al and high-performance Al monochromatic source was used to characterize the chemical states of elements on the surface of SrSnP in an evacuated ($10^{-9}$ Torr) chamber at room temperature. To keep the surface of our sample clean, an Ar ion gun was utilized to remove the possible oxidized layer of SrSnP.

**Electronic Structure Calculations.** Tight-Binding, Linear Muffin-Tin Orbital-Muffin-Spheres Approximation (TB-LMTO-ASA). Calculations of Crystal Orbital Hamiltonian Population (COHP) curves were performed by Tight-Binding, Linear Muffin-Tin Orbital-Atomic Spheres Approximation (TB-LMTO-ASA) using the Stuttgart code. The convergence criterion was set as 0.05 meV and a mesh of 64 k points was used to generate all integrated values. In the ASA method, space is filled with overlapping Wigner-Seitz (WS) spheres. The symmetry of the potential is treated as spherical in each WS sphere with a combined correction on the overlying part. The WS radii are 2.081 Å for Sr, 1.486 Å for Sn, and 1.367 Å for P. Empty spheres are required for the calculation, and the overlap of WS spheres is limited to no larger than 16%. The basis set for the calculations included Sr 5s4d; Sn 5s5p; and P 3s3p wave functions.

![Figure 1. Powder X-ray diffraction pattern of SrSnP where the blue and red lines represent observed pattern and calculated patterns based on single-crystal data, respectively. Trace Sn residual can be found. (Inset) Single crystal of SrSnP with dimension.](image-url)
Vienna Ab initio Simulation Package (VASP). The electronic structure was computed based on projector-augmented-wave (PAW) pseudopotentials that were adopted with the Perdew–Burke–Ernzerhof generalized gradient approximation (PBE-GGA) implemented in VASP. The experimental lattice parameters were used to perform the calculations. The energy cutoff was 400 eV. The 11 × 11 × 6 Monkhorst–Pack k-point mesh was used in calculation. The spin–orbit coupling effects are included.

Quantum Espresso. The electron–phonon coupling was completed using Quantum Espresso code.26,29 based on Density Functional Perturbation Theory (DFPT).30,31 which employs norm-conserving pseudopotentials. Reciprocal space integrations were completed over a 24 × 24 × 12 k-point mesh, 4 × 4 × 4 q-point mesh with the linear tetrahedron method. The cutoff energies were set at 40 Ry (400 Ry) for wave functions (charge densities). With these settings, the calculated total energy converged to less than 0.1 Ry per atom. The electron–phonon coupling strength \( \lambda_{\omega\nu} \) was calculated using

\[
\lambda_{\omega\nu} = \frac{1}{N_{\omega\nu}} \int \omega_{\nu}(\mathbf{q}) \exp[-\frac{1.04(1 + i)}{2-\mu} \lambda_{\omega\nu}^2] d\mathbf{q}
\]

where \( N_{\omega\nu} \) is the density of states (DOS) at the Fermi level and \( \omega_{\nu}(\mathbf{q}) \) is the phonon frequency of mode \( \nu \) at wave vector \( \mathbf{q} \). The critical temperature \( T_c \) can be estimated by

\[
T_c = \frac{\mu}{2 \omega_{\nu}(\mathbf{q})} \exp\left(\frac{\lambda_{\omega\nu}^2}{2} + \frac{1}{2} \int \omega_{\nu}(\mathbf{q}) d\mathbf{q}\right)
\]

\( \lambda_{\omega\nu} \) and \( \omega_{\nu}(\mathbf{q}) \) are obtained from calculation using Quantum Espresso. The superconductivity parameter \( \mu \) is determined from the McMillan formula, 32

\[
\mu = 0.128 \ln\left(\frac{\mu + 1}{\mu - 0.82}\right)
\]

SrSnP as an Elongation of SnP. The structure of SrSnP can be regarded as the elongation after the insertion of Sr atoms into the cubic SnP frame followed by the Sr-induced bond inversion, as described in Figure 3. To make the figure clear, we only show one-quarter of the structure of cubic SnP.

RESULTS AND DISCUSSION

Phase, Crystal Structure, and Chemical Composition Determination. According to the previous phase study in the Sr–Sn–P system, three different ternary phases that included orthorhombic SrSnP\(_4\) and SrSnP\(_6\) and tetragonal SrSnP were reported. Our synthetic approach did not yield anything other than crystalline tetragonal SrSnP phase. The powder X-ray diffraction pattern of SrSnP was shown in Figure 1. It can be found that a pure SrSnP phase was obtained with slight Sn-impurity. The broad peaks located between 25 and 30° are due to the decomposition of SrSnP in the air. The result matches with the calculated pattern generated from our single-crystal X-ray diffraction data and reference as well.15

To obtain further insights into the structural features of SrSnP, single crystals were investigated to extract atomic distributions and coordination environments. The results of single-crystal diffraction including atomic positions, site occupancies, and isotropic thermal displacements are summarized in Tables S1 and S2. Corresponding anisotropic displacement parameters are summarized in Table S3. The result is consistent with the previous report. The lattice parameters are slightly smaller because of lower testing up to 100 K. Both vacancy and mixed models were compared using Hamilton test24 to confirm the full occupancy on each site. The chemical compositions of Sr\(_{0.99(9)}\)Sn\(_{1.00(8)}\)P\(_{1.1(2)}\) examined by EDS are listed in Table S4. SrSnP crystallizes in the CaGaN structure type with the tetragonal space group P4/nmm which is the same as that for EuSnP.16 Each Sn atom in the puckered sheet is surrounded by a distorted tetrahedron consisting of four Sn atoms with the Sn–Sn distance \( \sim 3.21 \) Å. The connection to the upper and lower layers of the Sn layer is produced by an axial connection to the P atom at \( \sim 2.46 \) Å and an opposite axial connection to Sr at \( \sim 3.45 \) Å (Figure 2A). On adjacent Sn atoms the arrangement of axial Sr and P atoms is opposing, forming Sn–Sr–P–Sn stacking (Figure 2B). There are two layers comprised of Sr and P in a square array between every sheet of Sn atoms.
an electronic specific heat estimated by subtraction of the lattice contribution ($C_{\text{lat}}$) from the experimental data ($C_p$). Large anomaly that is visible in the specific heat data confirms bulk superconductivity of SrSnP. An equal area construction, denoted by solid lines, performed in order to determine superconducting transition temperature gave a value of $T_c = 2.3$ K. This temperature is in agreement with results obtained from magnetic susceptibility and resistivity measurements.

Figure 6B presents the results of measurement of $C_p/T$ vs $T^2$ for the data obtained under 0 T and 50 mT magnetic field. The normal state was fitted to the equation $C_p/T = \gamma + \beta T^2$, represented by a solid line, where $\gamma$ is an electronic contribution and $\beta$ is a phonon contribution to the specific heat. The estimated Sommerfeld coefficient is equal to $\gamma = 2.77(7)$ mJ mol$^{-1}$ K$^{-2}$ and $\beta = 0.39(1)$ mJ mol$^{-1}$ K$^{-4}$, which is connected to the Debye temperature through the relation

$$
\Theta_D = \left( \frac{12\pi^4}{5\beta} \right)^{1/3}
$$
where \( R = 8.31 \text{ J mol}^{-1} \text{ K}^{-1} \) and \( n = 3 \) for SrSnP. The Debye temperature calculated from this formula is \( \Theta_D = 246(1) \text{ K} \). Knowing this value allows calculation of the electron–phonon coupling constant from the inverted McMillan formula:

\[
\lambda_{\text{ep}} = \frac{1.04 + \mu^* \ln \left( \frac{\Theta_D}{1.45T_c} \right)}{(1 - 0.62\mu^*) \ln \left( \frac{\Theta_E}{1.45T_c} \right) - 1.04}
\]

Estimated value of \( \lambda_{\text{ep}} = 0.55 \) suggests that SrSnP is a weak coupling superconductor. For this calculation, the Coulomb repulsion constant was taken as \( \mu^* = 0.13 \). A weak coupling is also suggested by the normalized superconducting anomaly jump that was calculated to be about \( \Delta C/\gamma T_c = 1.48 \), which is slightly above the expected value (1.43) for weak-coupling BCS superconductors.

Having both the electron–phonon coupling constant \( \lambda_{\text{el-ph}} \) and the Sommerfeld coefficient \( \gamma \), the density of states at the Fermi energy can be calculated from

\[
N(E_F) = \frac{3\gamma}{\pi^2k_B^2(1 + \lambda_{\text{ep}})}
\]

The value for SrSnP is 0.86 states eV\(^{-1}\) per f.u. Figure 6C presents the temperature dependence of \( C_p/T \) divided by \( T^3 \). A maximum of \( C_p/T^3 \) occurs at about \( T_{\text{max}} \approx 14 \text{ K} \), and since \( T_{\text{max}} = \Theta_E/5 \), the Einstein temperature is estimated to be \( \Theta_E \approx 70 \text{ K} \).

**Single Valuence or Mixed Valuence of Sn?** With the question of the valence state of Sn in mind, we performed the X-ray photoelectron spectroscopy (XPS) experiment on SrSnP. Figure 7 shows the XPS spectrum of Sn 3d\(_{5/2}\) core-level region. The asymmetric line shape of Sn 3d\(_{5/2}\) is due to the metallic properties. To clarify the valence state of Sn, we assume Sn has mixed valences of Sn\(^0\) and Sn\(^{2+}\). With use of this hypothesis, the fitting peaks of the Sn 3d\(_{5/2}\) core-level are 484.62 and 485.40 eV with the atomic ratio of 5.18:1 between Sn\(^0\) and Sn\(^{2+}\). However, according to Figure S1 (Supporting Information), the shape of Sn\(^{2+}\) peak we assumed is not reasonable because of the broad characteristic and the full width at half-maximum (fwhm) of peaks are too large (1.43 and 1.59 eV), which confirms the single-valent hypothesis of Sn. The peak position of Sn 3d\(_{5/2}\) (484.65 eV) is slightly higher than Sn metal (484.38 eV), which is also tested as standard (see Figure S2) but lower than Sn-related oxides such as SnO (485.8 eV). The measurement indicates that Sn in SrSnP has only one valence state and the XPS result is consistent with the fact that SrSnP has only one Sn site without any structural distortion. Is Sn in the +1 oxidation state? Usually, Sn\(^0\) with four valence electrons would be four-bonded with other Sn atoms, and Sn\(^+\) will be left with only three valence electrons to form four Sn–Sn bonds with puckered nets, if all bonds are assumed to be normal two-center two-electron (2c–2e) bonds. However, the observation of four long Sn–Sn bonds implies partially fractional bonding in which the bonds are weaker than normal 2c–2e bonds. To further confirm this observation, we integrated the partial DOS of Sn, ~2.39 and 2.38 valence
electrons per Sn without and with spin–orbit coupling (SOC), respectively.

**Electronic Structure of SrSnP.** To gain further insights into the electronic influences on the stability and superconducting properties of SrSnP, VASP calculations were carried out to evaluate and analyze the band structures. Figure 8A shows the first Brillouin zone (BZ) of a primitive tetragonal lattice with high-symmetry k points marked. The calculated bulk band structure of SrSnP, generated here with the inclusion of SOC, is shown in Figure 8B. The band structure calculation indicates that the Sn 6p and P 3p orbitals dominate the conduction and valence bands around the Fermi level (E_F).

The Fermi surface is predominantly comprised of two symmetric electron pockets along Z−A−Z and Γ−M−Γ and saddle points around Γ point, which are significant for the superconductivity. The “symmetric” band structure is usually originated from the band folding in the first Brillouin zone. Furthermore, the band structure of first BZ applies for the crystal momentum in the compound. To evaluate the real momentum in SrSnP more accurately, we performed the unfolding band structure calculation in the second Brillouin zone. With the Sn−Sr−P−Sn stacking along the c-axis, the closed Sn−Sr−P−Sn chain is repeating along the body diagonal on the ab-plane shown in Figure 8C. In the second BZ picture, electrons encounter Sn atoms in the crystal and form the Bloch waves. The electron interactions with P or Sr can be treated as a small perturbation to the Bloch wave. Thus, the second Brillouin zone and band structure are generated in Figure 8D,E. The unfolding of the new BZ marked in a blue line shows the actual measured gap is at the M point. A big band gap appears at the Γ point instead. To evaluate the SOC effects on the electronic structure of SrSnP, we also calculated the DOS shown in Figure 9. Interestingly, SOC opens up the electronic gap, but it has no effect on N_F (1.59 states/eV for both).

**Bonding Interactions in SrSnP.** To understand the atomic interactions in the SrSnP, the corresponding COHP curves were calculated and presented in Figure 10A. It shows that Sr−P bonding and Sn−P antibonding interactions govern the atomic interactions contribution around the Fermi level while Sn−Sn bonding interaction with Sn−Sn ~3.21 Å is not as significant as Sr−P or Sn−P interactions. Moreover, the Sn−Sn interactions are barely found. This indicates the importance of Sr−P bonding interaction in stabilizing the structure and the possibility for Sn−P antibonding interaction to influence the physical properties most. Moreover, the Sn−P bonding
interactions are dominated below the Fermi level along with Sr–P bonding interactions, which signifies that the crucial bonding interaction in SrSnP are Sn–P and Sr–P bonds. Corresponding to the previous structural study between SnP and SrSnP, it can be shown that the valence electron contribution from Sr atom overlaps with Sn–P σ-like antibonding part (Figure 10B). Such overlapping weakens the Sn–P bonding interaction and possibly causes the partially bond breakage of Sn–P which leads to the unusual oxidation state of Sn in SrSnP. This can be an analogous to Sn in SnP or SnAs$^{12-14}$ because of the strong electron transfer from Sn to P or As in the previous study. Therefore, the lower oxidation state delocalizes more valence electrons on Sn atoms which may strengthen electron–phonon coupling and induce the superconductivity.

**Electron–Phonon Coupling in SrSnP.** To test whether this molecule-like hypothesis works for SrSnP in the context of the electron–phonon coupling expected from a physics-based picture, the systematic calculations for the electron–phonon coupling of SrSnP was calculated by use of the program Quantum Espresso. Figure 11A,B shows the phonon spectra of SrSnP with and without SOC. It can be clearly noted that the most contribution comes from Sn and SOC affects the phonon band at M point significantly, which also hosts the saddle points in the electronic band structure. In our calculations in Figure 12, we find the total electron–phonon coupling strength $\lambda_{el-ph}$ to be 0.59 without SOC and 0.62 with SOC, which is consistent with the experimental evaluation. With use of the McMillan formula, the superconducting transition temperatures $T_c$’s are estimated to be 1.95 K SrSnP without SOC and 2.15 K for SrSnP with SOC. The theoretical prediction of $T_c$ with SOC matches with our experimental result $T_c = 2.3$ K very well, which indicates the SOC can help to improve the $T_c$.

**Spin–Orbit Coupling (SOC) on $T_c$.** How does SOC improve the $T_c$? Derived from the BCS theory, the superconducting critical temperature can be expressed as $kT_c = 1.13\hbar \omega \exp(-1/N(E_F)\omega)$, where $V$ is a merit of the electron–phonon interaction, $N(E_F)$ is the DOS at Fermi level, and $\omega$ is a characteristic phonon frequency. The SOC usually affects the superconducting transition temperature by influencing $N(E_F)$ or $\omega$. In SrSnP, since the SOC gap is near the high symmetry point, the impact of SOC on $N(E_F)$ is quite small. Instead, the SOC affects $\omega$ and increases the $T_c$ in SrSnP.

### CONCLUSION

Herein, we successfully grew the single crystal of reported ternary compound SrSnP with a tetragonal space group of $P4/nmm$. Physical properties measurements confirm the existence of bulk superconductivity, which is consistent with the electronic and phononic calculation prediction. We used XPS measurements to examine the single-valent state of Sn in SrSnP, which is critical for the superconductivity. From a chemistry viewpoint, the Sr–P bonding and Sn–P antibonding interactions are related to the anomalous oxidation state of Sn and superconductivity in SrSnP. The systematic electron–phonon coupling calculation from physics perspectives matches well with the molecule-based chemistry picture. The experimental and partial theoretical superconducting and thermodynamic parameters of SrSnP are summarized in Table 1. More interestingly, the strong spin–orbit coupling increases the critical temperature in SrSnP.
Crystallographic data, atomic coordinates, equivalent isotropic displacement parameters, anisotropic thermal displacement, SEM data, and XPS spectra of SrSnP and standard Sn (PDF)

ASSOCIATED CONTENT
* Supporting Information
The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.chemmater.8b02258.

Table 1. Experimental and Partial Theoretical Superconducting and Thermodynamic Parameters of SrSnP

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<th>experimental</th>
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<td>$T_c$ (K)</td>
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<td>$H_c$ (Oe)</td>
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<td>$\gamma$ (mJ mol$^{-1}$ K$^{-2}$)</td>
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<td>$\beta$ (mJ mol$^{-1}$ K$^{-4}$)</td>
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<td>$\lambda_{el-ph}$</td>
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REFERENCES