Spintronics --- Half metal and Colossal Magnetoresistance

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Chapter One

Half Metal

Outline

1. *Definition of half metal*
2. *Material classification*
3. *Polarization measurement*
4. *Applications*

New Class of Materials: Half-Metallic Ferromagnets

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and

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(Received 21 March 1983)

The band structure of Mn-based Heusler alloys of the $C1_b$ crystal structure (MgAgAs type) has been calculated with the augmented-spherical-wave method. Some of these magnetic compounds show unusual electronic properties. The majority-spin electrons are metallic, whereas the minority-spin electrons are semiconducting.

Compound	$N(E)\uparrow$	$N(E)\downarrow$	$n_{3d}^{Mn}\uparrow$	$n_{3d}^{Mn}\downarrow$	μ_{tot}^{cacl}	μ_{tot}^{exp}
NiMnSb	9.90	0	4.51	0.87	4.00	3.85
PtMnSb	10.05	0	4.57	0.79	4.00	3.97
PdMnSb	9.04	2.97	4.58	0.71	4.05	3.95
PtMnSn	9.78	19.31	4.40	0.78	3.60	3.42

Half metals are the extreme case of strong ferromagnet, where not only 3d electrons are fully polarized, but also other (sp) down-spin bands do not cross the Fermi level.
(examples: NiMnSb, PtMnSb--- Hesuler phases.)

JOURNAL OF APPLIED PHYSICS VOLUME 91 (2002)

Half-metallic ferromagnetism: Example of CrO2 (invited)

J. M. D. Coey and M. Venkatesan

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A **half metal** is a solid with an unusual electronic structure. For electrons of one spin it is a metal with a Fermi surface, but for the opposite spin there is a gap in the spin-polarized density of states, like a semiconductor or insulator. This definition presupposes a magnetically ordered state to define the spin quantization axis. The responses of a half metal to electric and magnetic field at zero temperature are quite different. There is electric conductivity, but no high-field magnetic susceptibility.

TABLE I. Summary of the classification of half-metals.

Type	Density of states	Conductivity	↑ electrons at E_F	↓ electrons at E_F	
IA	Half-metal	Metallic	Itinerant	None	(CrO_2,NiMnSb)
IB	Half-metal	Metallic	None	Itinerant	(Sr2FeMoO6)
IIA	Half-metal	Nonmetallic	Localized	None	
IIB	Half-metal	Nonmetallic	None	Localized	
IIIA	Metal	Metallic	Itinerant	Localized	(Magnetite)
IIIB	Metal	Metallic	Localized	Itinerant	($La_{0.7}Sr_{0.3}MnO_6$)
IVA	Semimetal	Metallic	Itinerant	Localized	
IVB	Semimetal	Metallic	Localized	Itinerant	($Tl_2Mn_2O_7$)
VA	Semiconductor	Semiconducting	Few, itinerant	None	(Doped EuO & EuS)
VB	Semiconductor	Semiconducting	None	Few, itinerant	(GaAsMn)

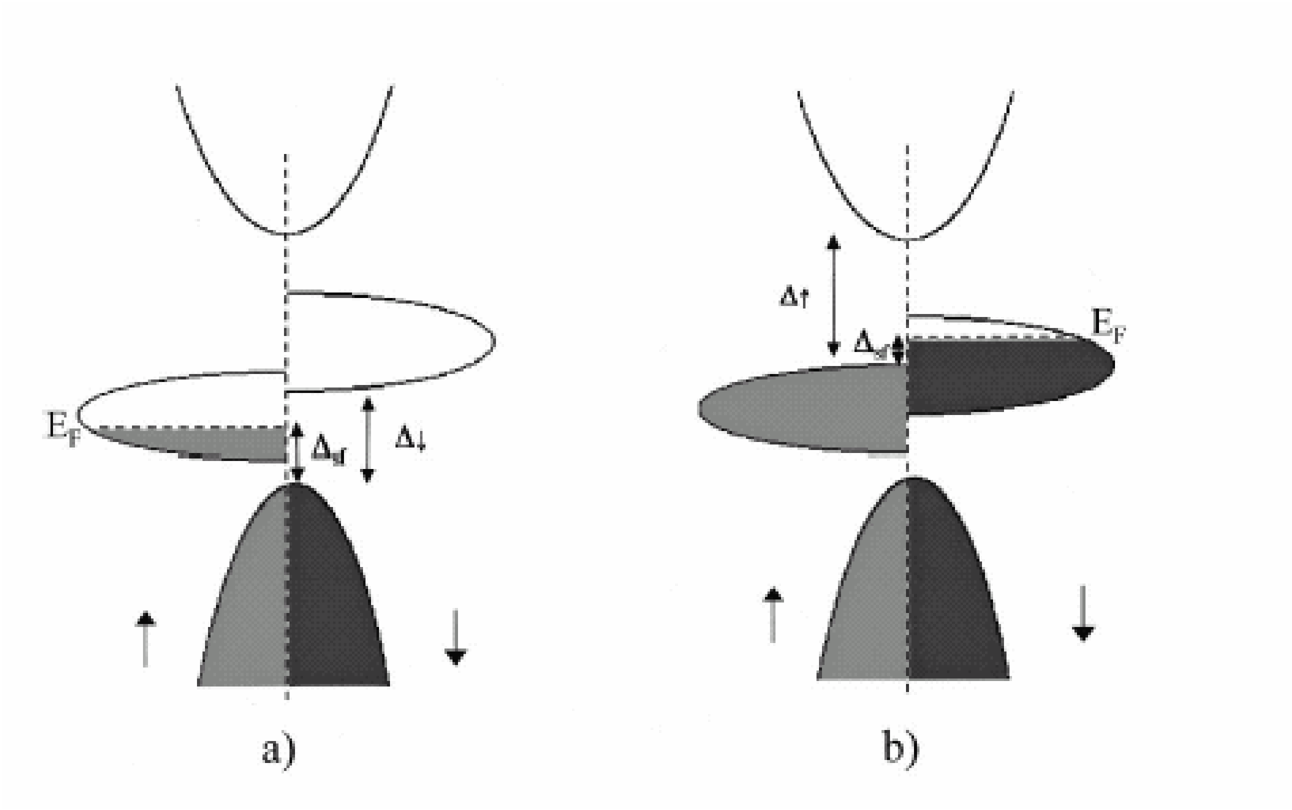


FIG. 1. Schematic density of states for a half metal, (a) Type I_A with only ↑ electrons at E_F and (b) Type I_B with only ↓ electrons at E_F . In narrow d bands, the states at E_F may be localized (type II).

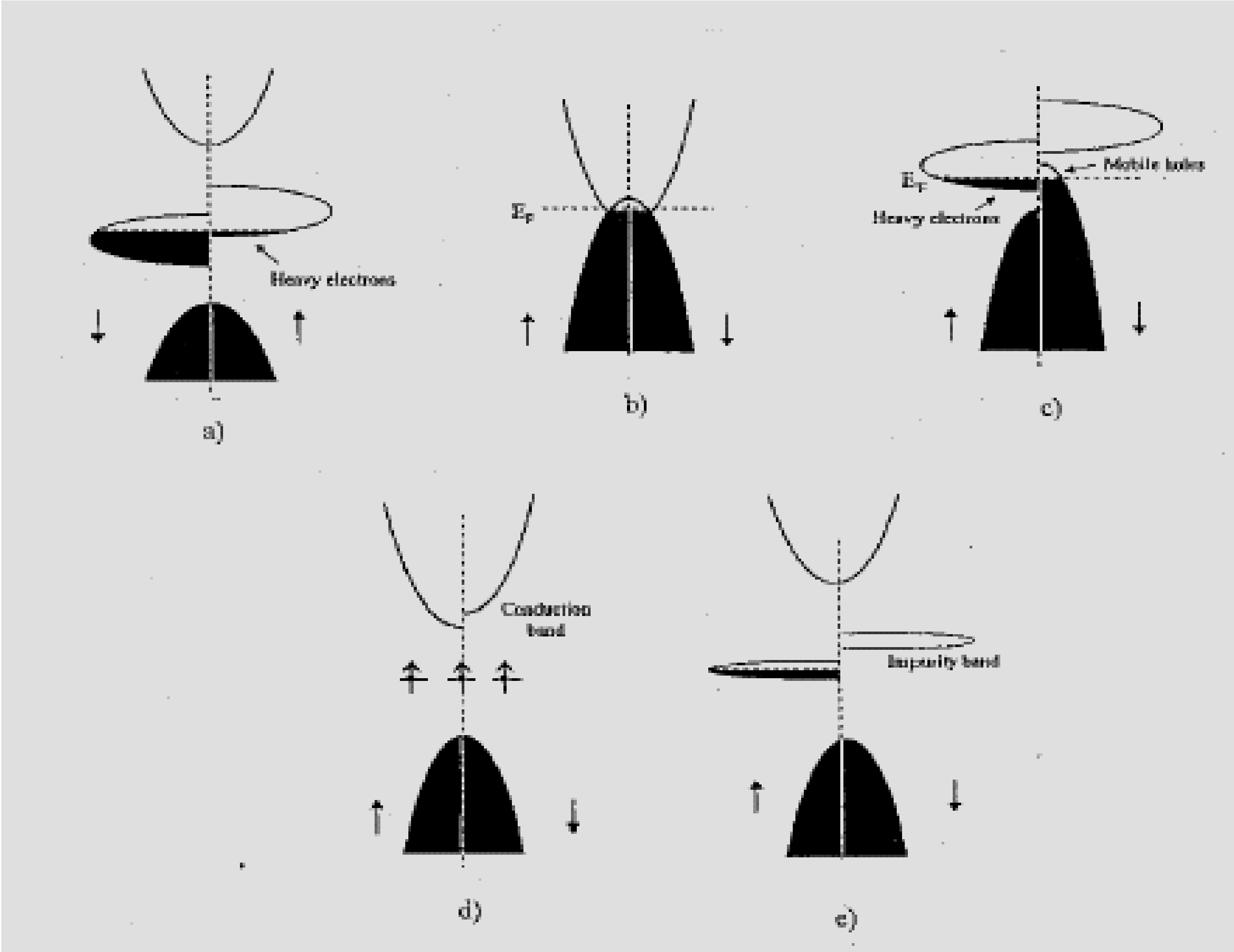


FIG. 2. Schematic density of states for (a) a type III_A half metal, where electrons of one spin direction are itinerant and the others are localized, (b) a semimetal, (c) a type IV_A half metal, and (d), (e) two types of ferromagnetic semiconductor.

Definition of polarization

$$P_0 = (N^{\uparrow} - N^{\downarrow})/(N^{\uparrow} + N^{\downarrow}),$$

--- straightforward

$$P_n = (\langle N^{\uparrow} v^{\uparrow n} \rangle - \langle N^{\downarrow} v^{\downarrow n} \rangle)/(\langle N^{\uparrow} v^{\uparrow n} \rangle + \langle N^{\downarrow} v^{\downarrow n} \rangle).$$

--- real measurement

ν: fermi velocity of electrons

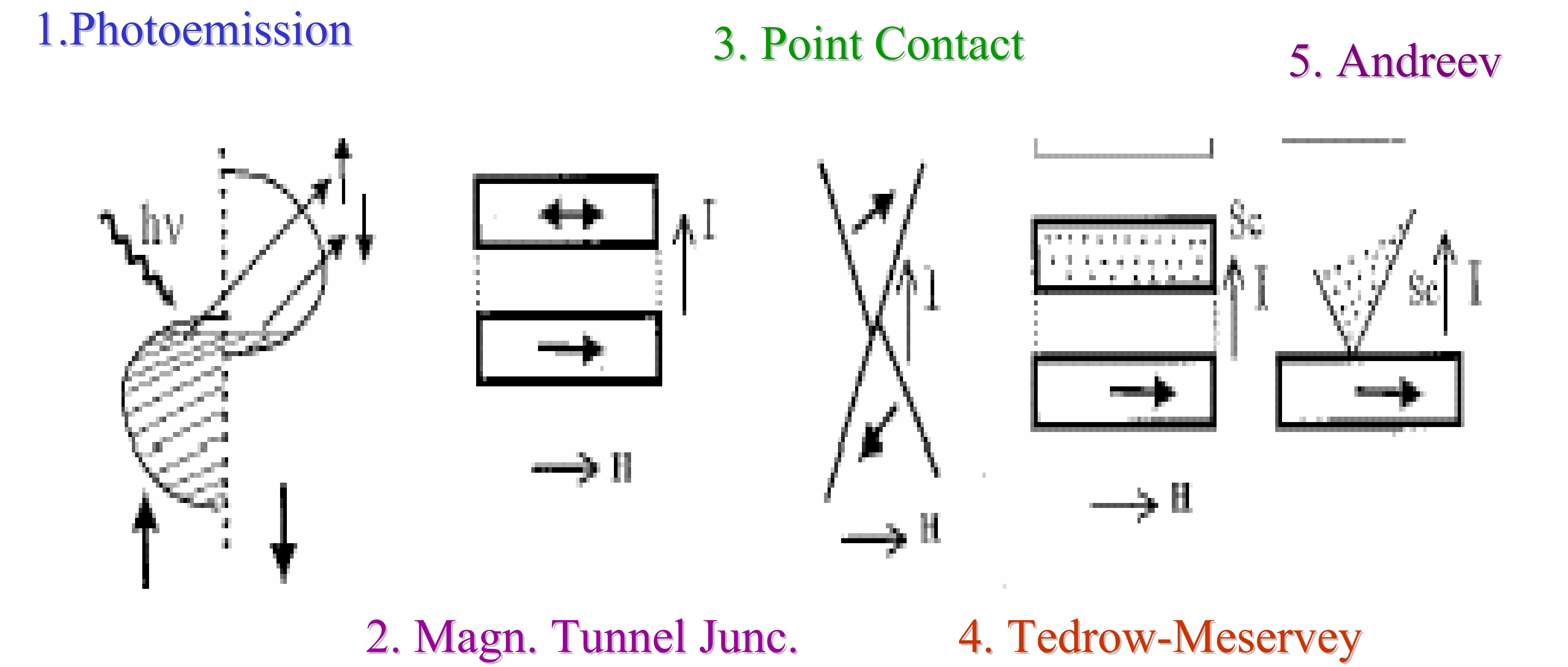


FIG. 3. Comparison of five methods of measuring *P*: Photoemission, tunnel junction, point contact, Tedrow–Meservey experiment, Andreev reflection.

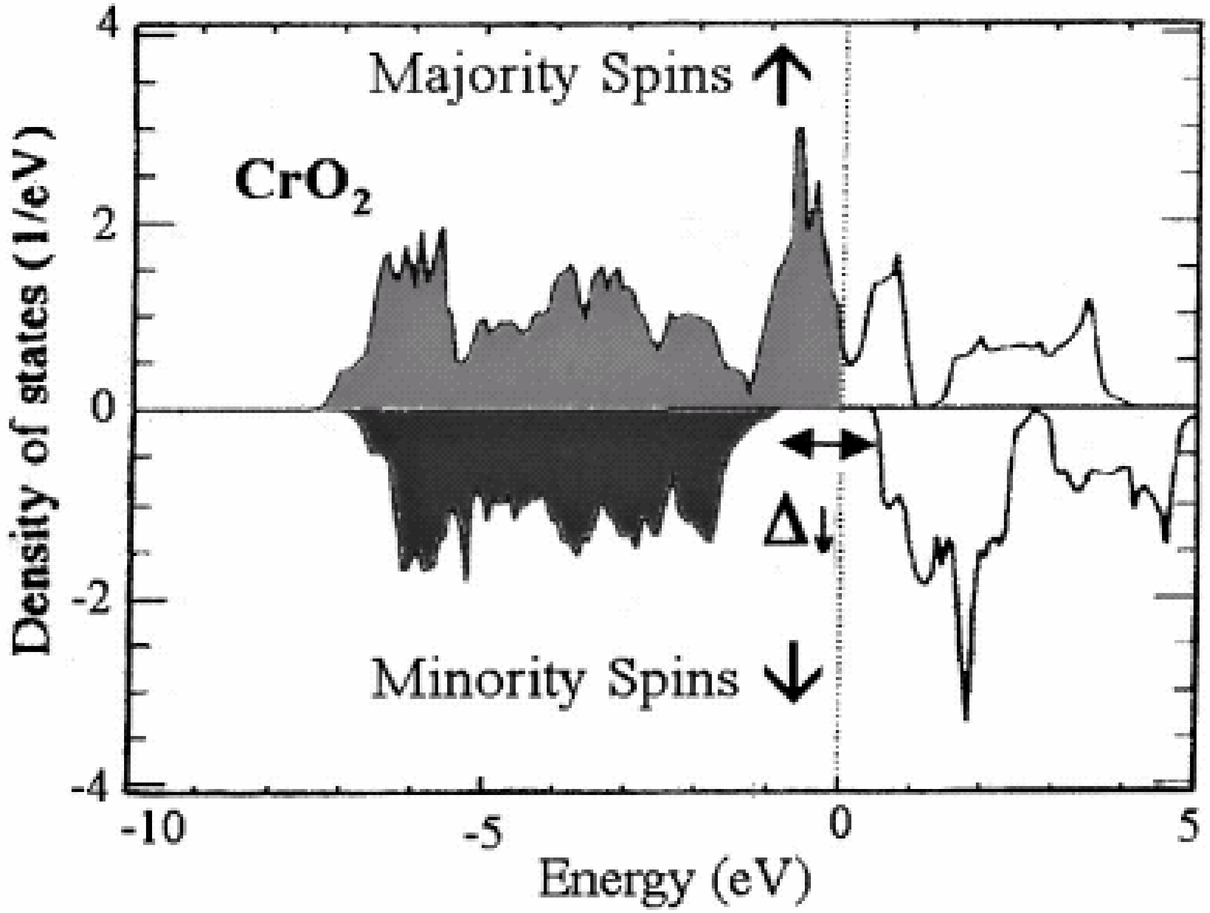


FIG. 5. Spin polarization of the density of states of CrO_2.

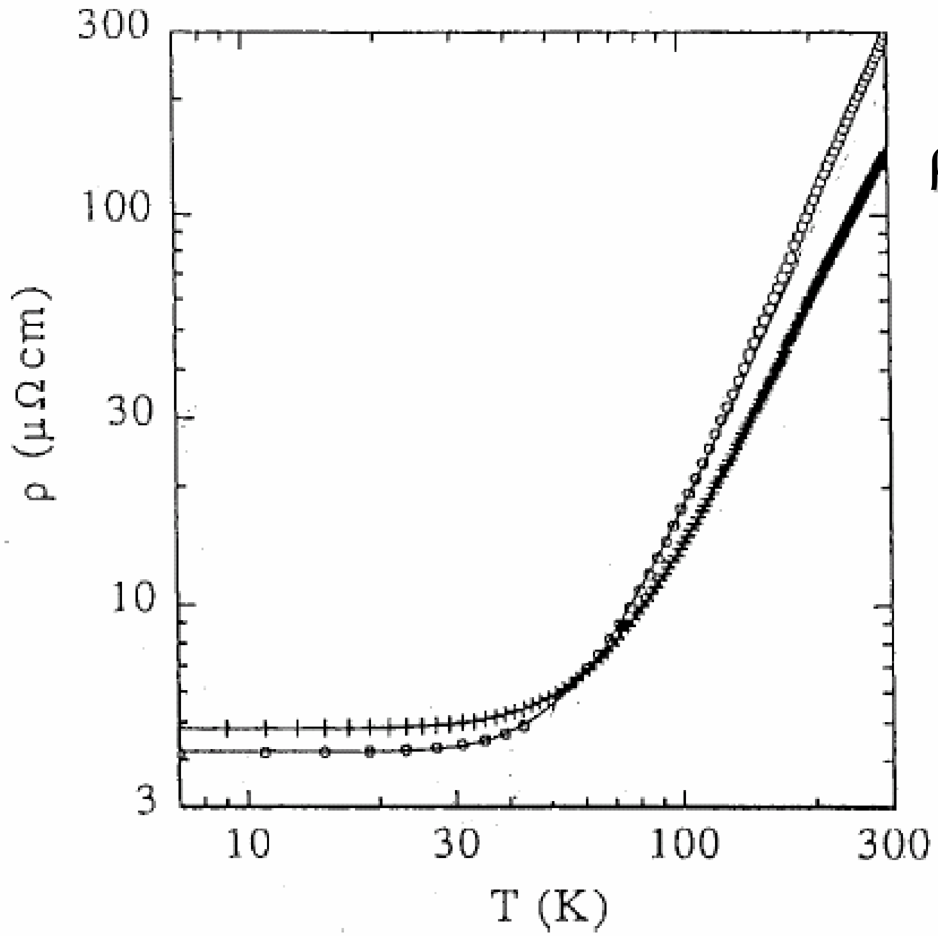
TABLE III. Some electronic structure calculation on CrO_2.

Author		$\Delta_\downarrow$ (eV)	Δ_{sf} (eV)	$N_\uparrow$ eV^{-1} $f.u^{-1}$
Schwarz	Ref. 26 LSDA–ASW	1.3	0.3	0.8
Lewis *et al.*	Ref. 2 LSDA–PWPP	1.4	0.3	0.69
Korotin *et al.*	Ref. 27 LSDA+U(3 eV)	2.4	1.7	0.4
Mazin *et al.*	Ref. 31 LSDA/GGA	1.3	0.2–0.7	0.95
Brener *et al.*	Ref. 32 LSDA–LCGO	1.3	0.2	1.16
Kuneš *et al.*	Ref. 28 GGA	1.8	0.7	0.3

Δ_{Sf} : spin-flip gap

TABLE II. Calculated spin polarization in ferromagnetic oxides.

	CrO_2 (Ref. 2)	$(La_{0.67}Ca_{0.33})MnO_3$ (Ref. 7)	$Tl_2Mn_2O_7$ (Ref. 8)
$N^{\uparrow}$ (eV^{-1} $f.u^{-1}$)	0.69	0.58	1.25
$N^{\downarrow}$ (eV^{-1} $f.u^{-1}$)		0.27	0.24
$V_F^{\uparrow}$ (10^6 ms^{-1})	0.25	0.76	0.06
$V_F^{\downarrow}$ (10^6 ms^{-1})		0.22	0.33
P_0 %	100	36	66
P_1 %	100	76	−5
P_2 %	100	92	−71



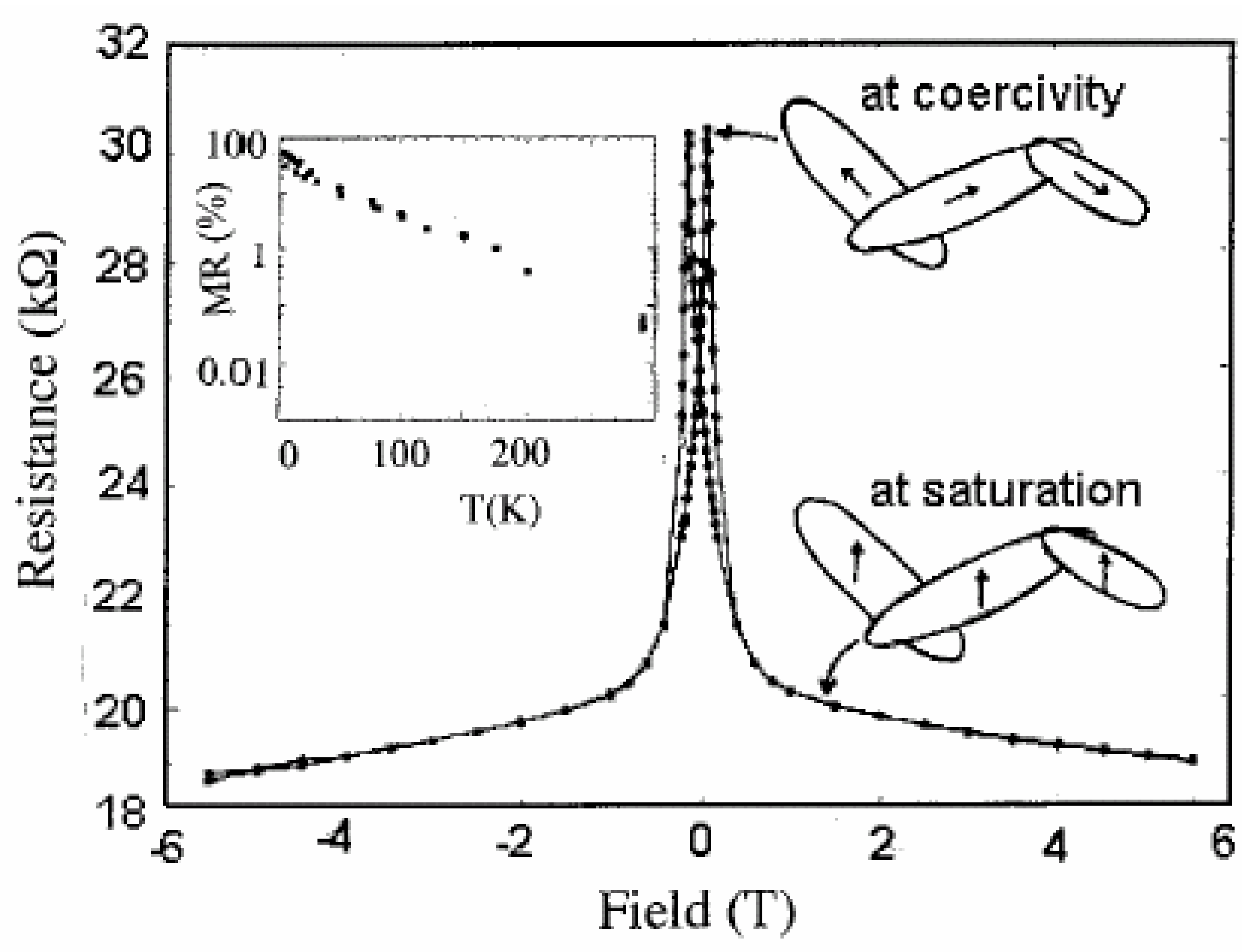
$\rho_o = 4\times10^{-8}$ ohm-m

~ 90 nm mean-free path

$\rho = \rho_o T^2 e^{-\Delta/T}$

with $\Delta \sim 80$K

FIG. 6. Resistivity of CrO_2 thin films.



MR

= [R(H)-R(0)]/R(0)

= $P^2/(1+P^2)$

FIG. 7. Magnetoresistance of a CrO_2–Cr_2O_3 pressed powder compact, with temperature dependence shown in the inset.

Applications of Half metals

a. Magneto-optical effects

Large Kerr rotation in PtMnSb.

b. **GMR applications**

Spin-valve system --- pick-up head, MRAM

c. **Spin electronics**--- Injection of polarized carriers

i) The spin injection in a normal metal can give information on the spin diffusion length in this metal.

ii) Spin injection may act as a pair-breaking agent in a super-conductor.

iii) Half metals can also be used to build a spin transistor

iv) Another possible application is as polarized tips in STM, in order to visualize the orientation of magnetic domains.

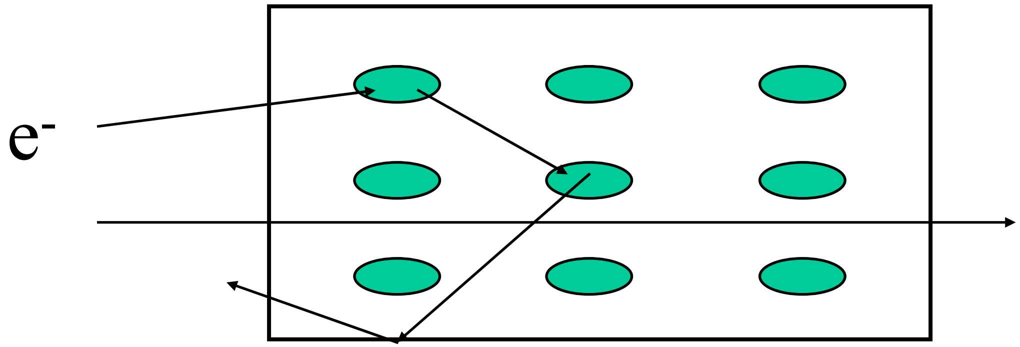
Chapter Two

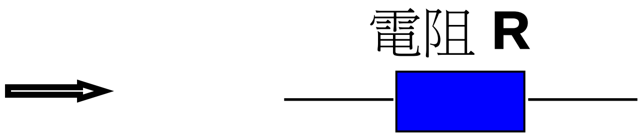
Colossal Magnetoresistance

Outline

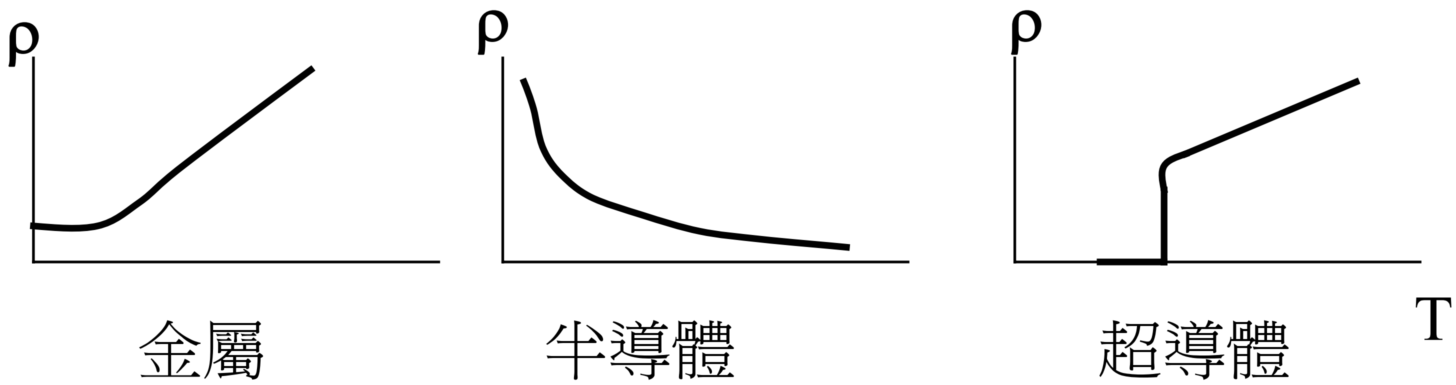
1. *Introduction*
2. *Material*
3. *Physical Properties*
4. *Experiments*
   a. *bulk*
   b. *film, bilayers*

1. Introduction ---何謂電阻(率)

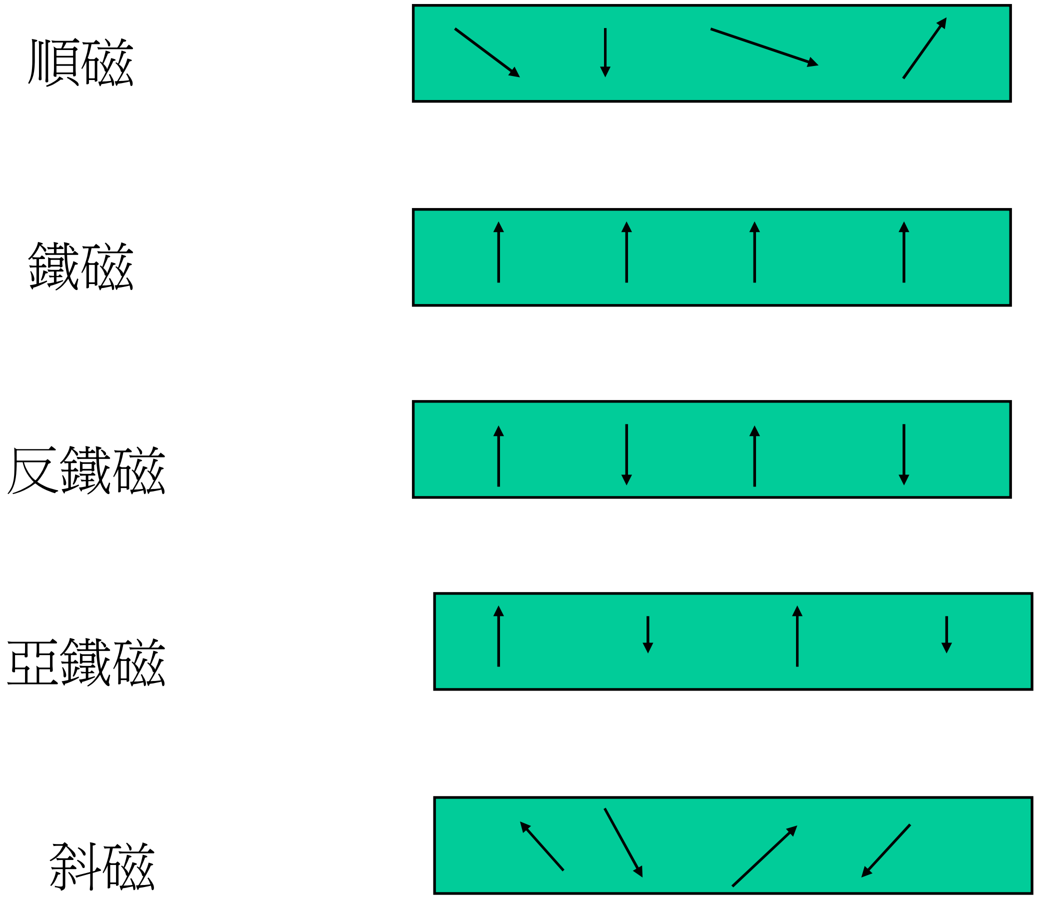




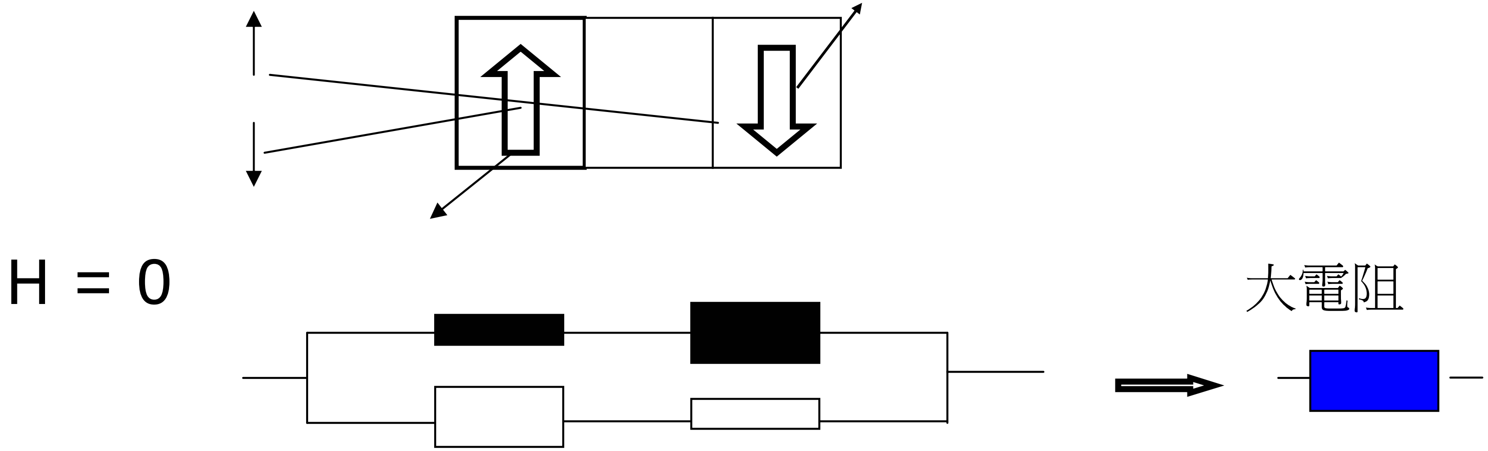
$\rho = R*A/w$

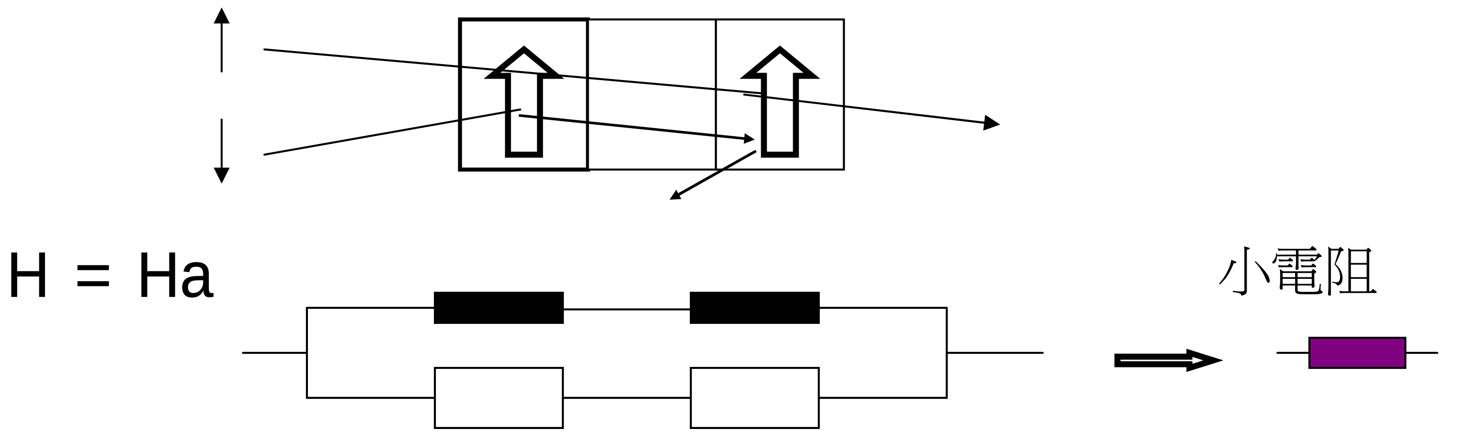


Introduction ---物質的基本磁性

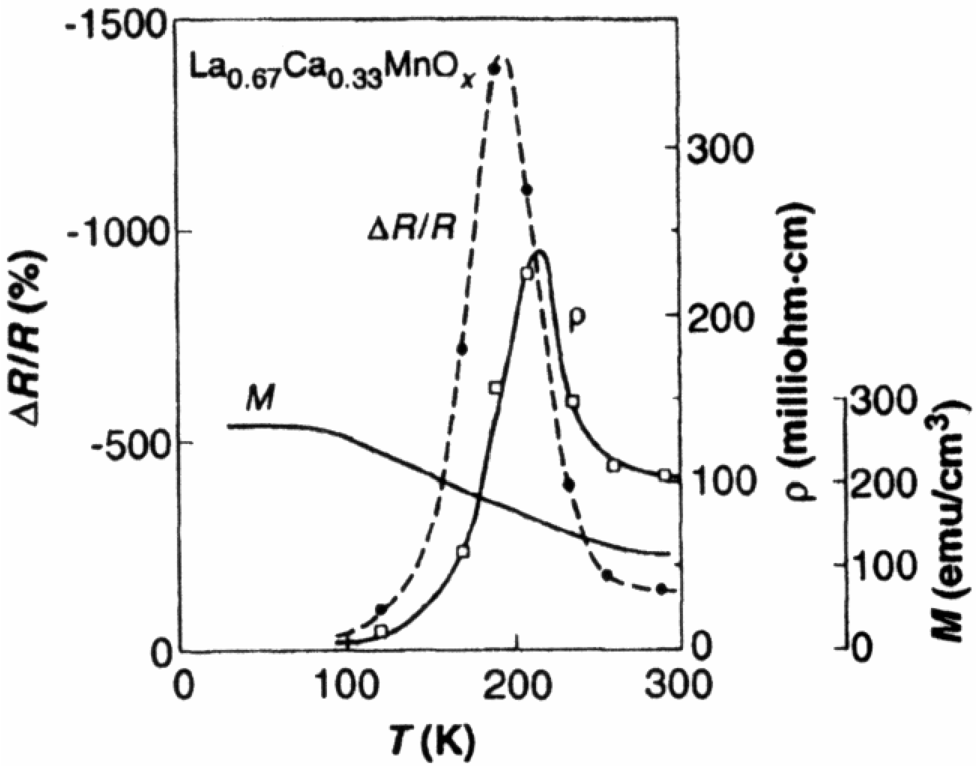


Introduction ---何謂磁阻





Introduction --- 何謂龐磁阻(CMR)



S. Jin et al. Science 265 (1994)

1) Very high magnetoresistance (99% at Tp)
2) Polarization 100% (half metal)
3) M- I (F/AF) transition
4) Phase separation
5) Charge ordering
6) Orbital ordering

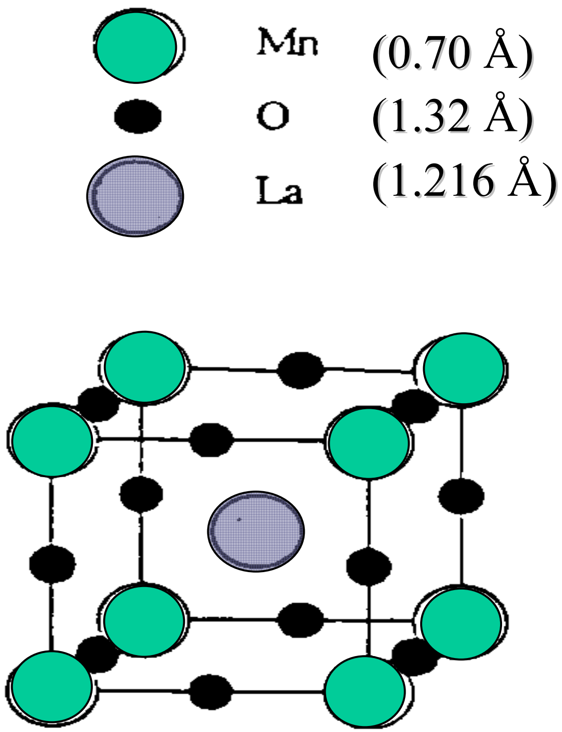
MR =[ρ(H)-ρ(0)]/ρ(H=6 tesla)

MR ratio of spintronic materials

Type	MR	Field	Temp.	Sample
OMR	0.01%	~Tesla	RT	Cu,Al
AMR	2 %	10 Oe	RT	Fe,Co,Ni
GMR	10 %	2 Oe	RT	Fe/Cr/Fe
TMR	40 %	10 Oe	RT	Co/AlO/Co
CMR	10 (99)%	~Tesla	RT(LT)	La-Sr-Mn-O

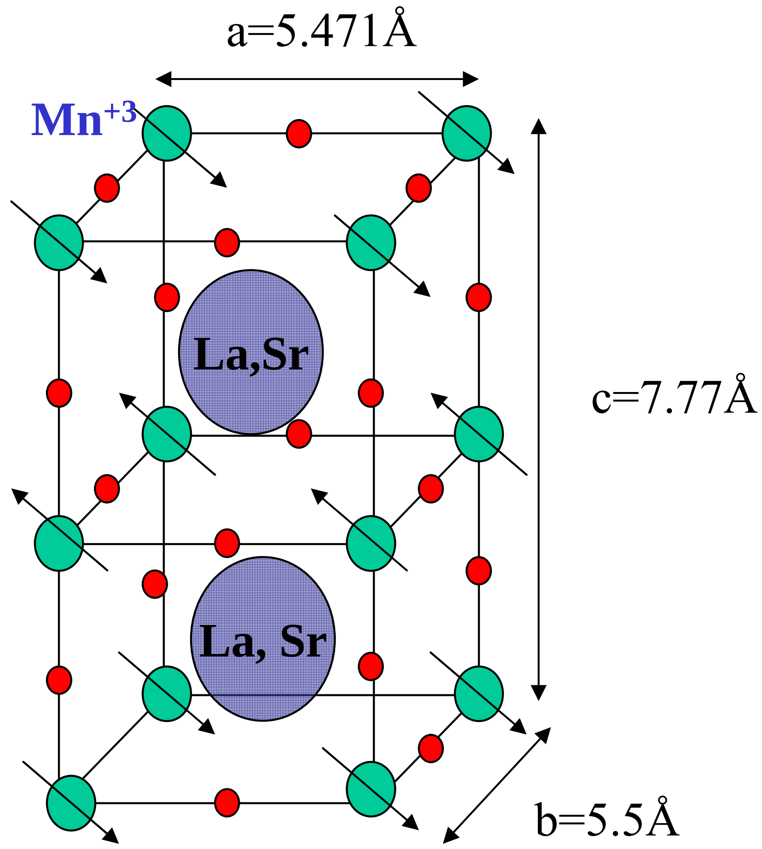
2. Material Perovskite layered structure

$LaMnO_3$: Cubic (insulator, Antiferromagnetic)

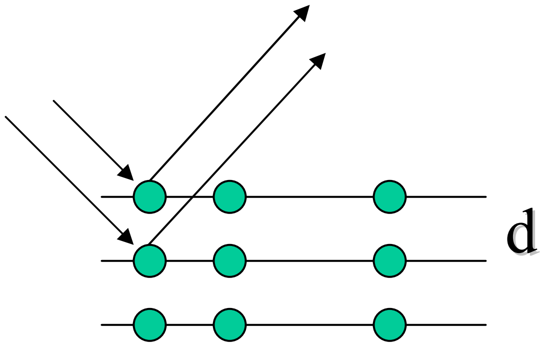


$(La,Sr)MnO_3$: orthorhombic

(Metal, Ferromagnetic)

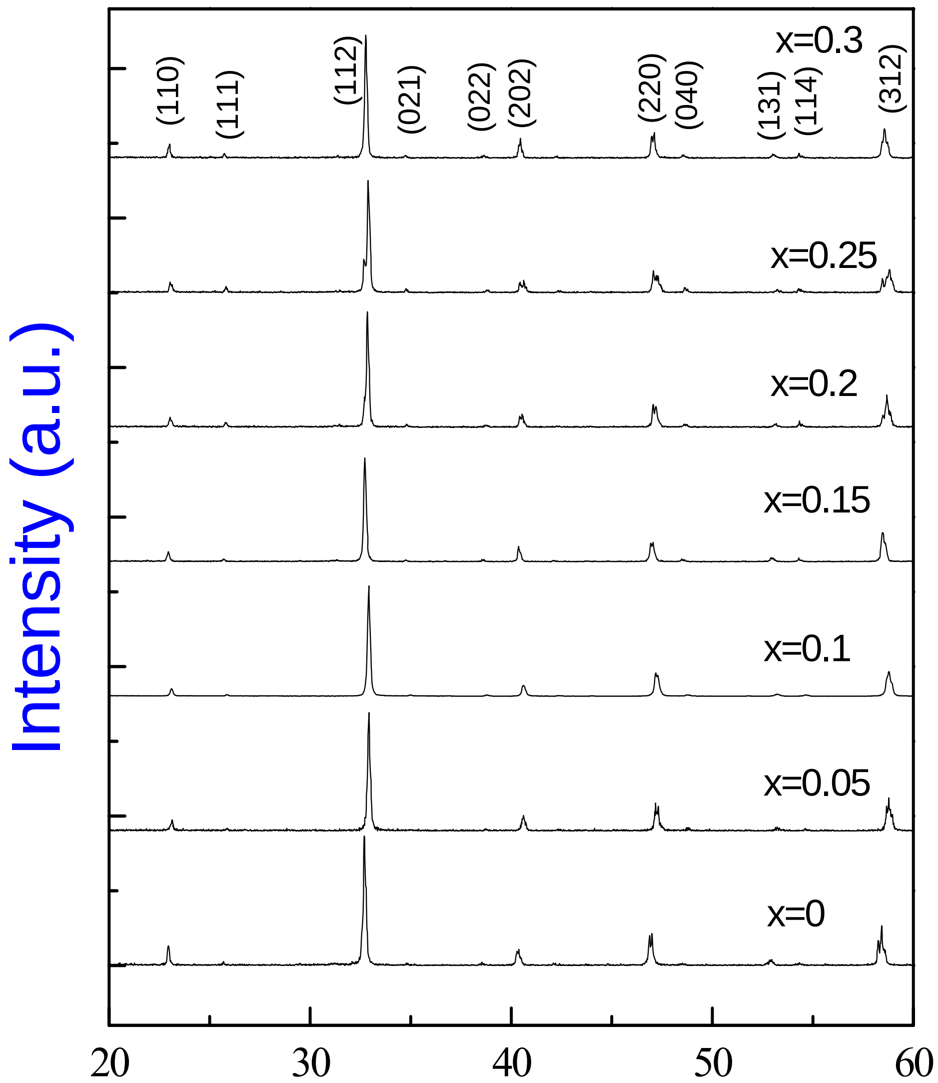


X – ray diffraction pattern



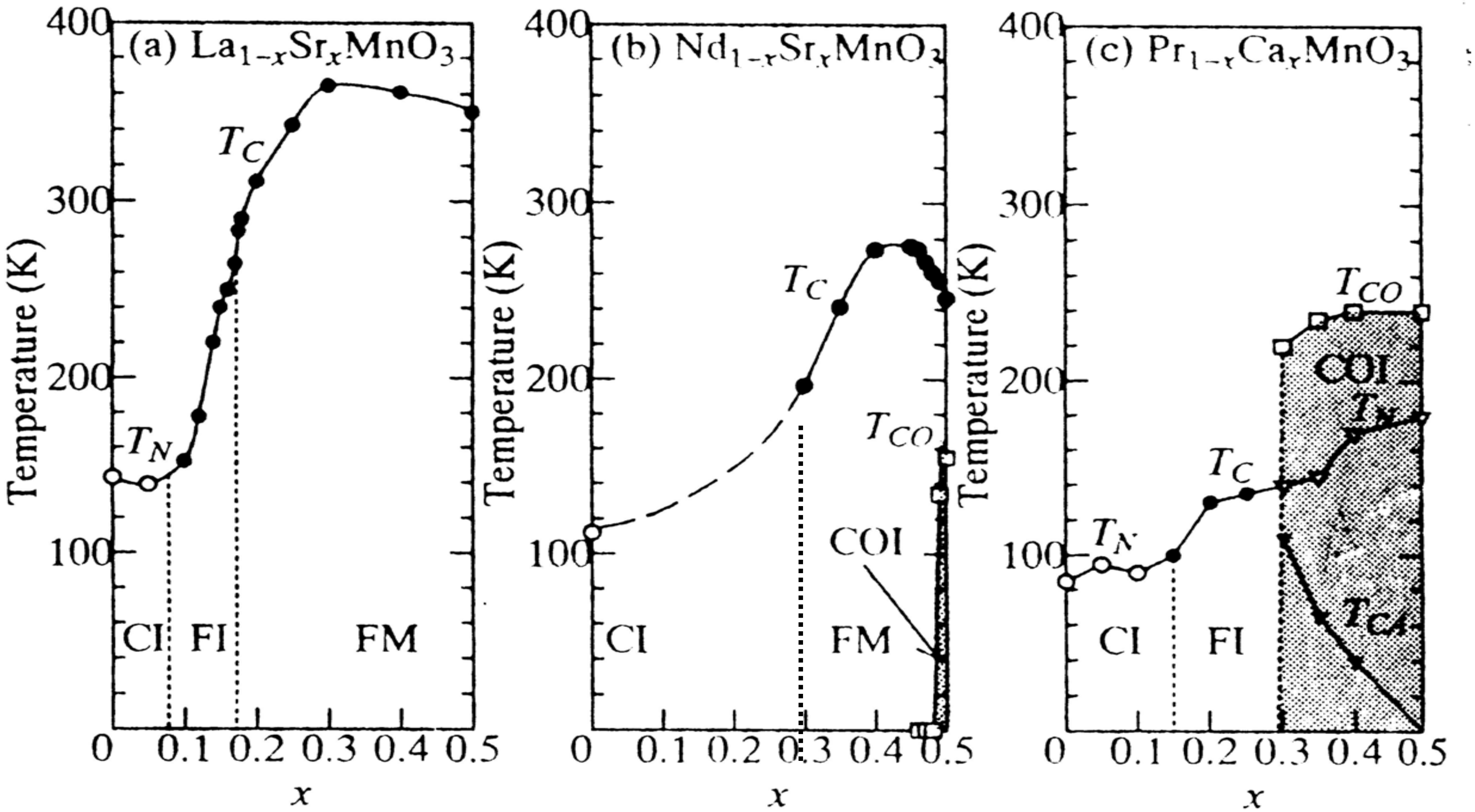
Bragg condition:

$$n\lambda = 2d\sin\theta$$



Phase diagrams of $R_{1-x}A_xMnO_3$

degrees of freedom: charge, spin, orbital

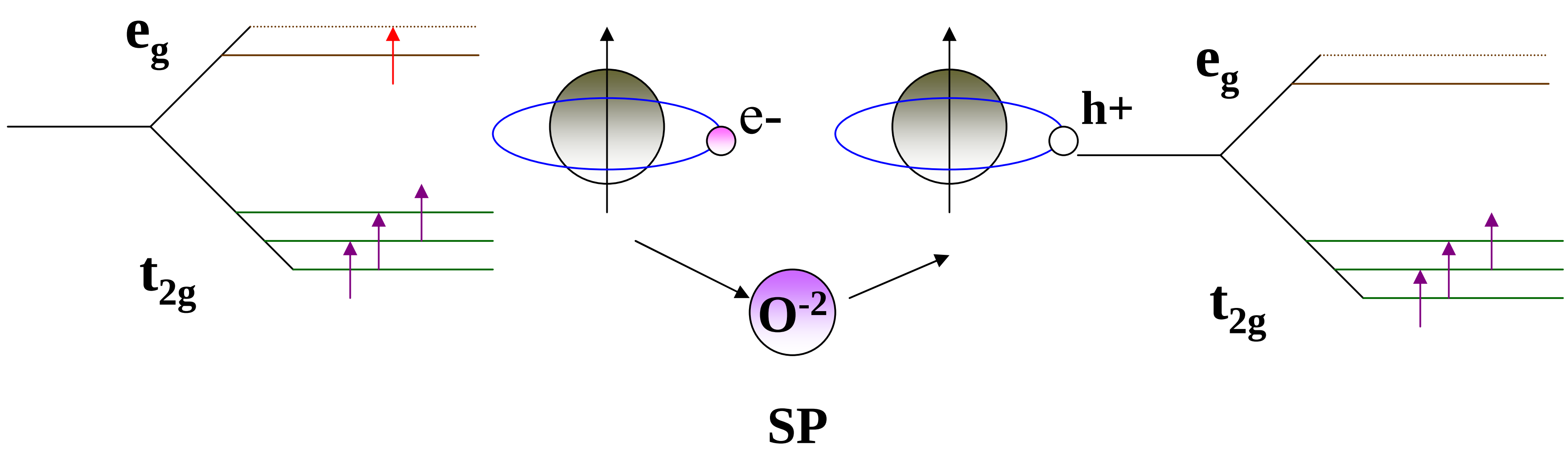


3. Physical properties & mechanism

What's the new physics

Double exchange (1951, Zener)

$(La,Sr,)MnO_3$



Mn^{+3}: $3d^4$ $(t_{2g}^{3}\ e_g^{1})$

Mn^{+4}: $3d^3$ $(t_{2g}^{3}\ e_g^{0})$

Mechanism vs. degree of freedom

- FM/AFM superexchange --- spin
- Charge/orbital ordering --- charge/orbital
- double exchange --- charge/spin
- Lattice distortion (John-Teller effect) --- lattice

Physical Review B, Volume 58, Number 17 1 November 1998-I

Phase diagram of manganese oxide

Ryo Maezono, Surnio Ishihara, and Naoto Nagaosa
Department of Applied Physics, University of Tokyo, Bunkyo-ku, Tokyo 113, Japan

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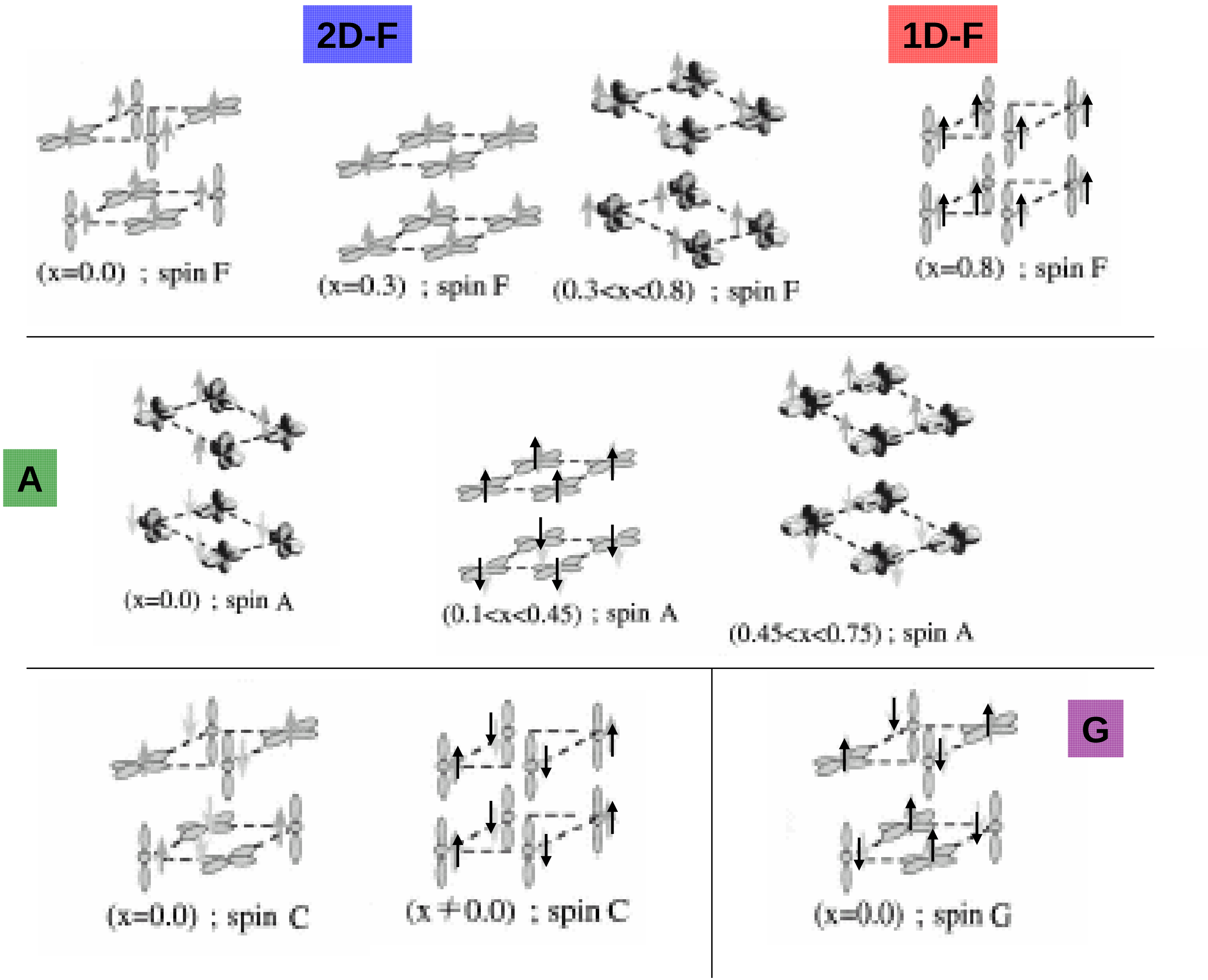
$$H = H_K + H_{Hund} + H_{on\,site} + H_S$$

$$H_K = \sum_{\sigma\gamma\gamma'(ij)} t_{ij}^{\gamma\gamma'} d_{i\sigma\gamma}^{+} d_{j\sigma\gamma'}$$ (Kinetic energy of e_g electrons)

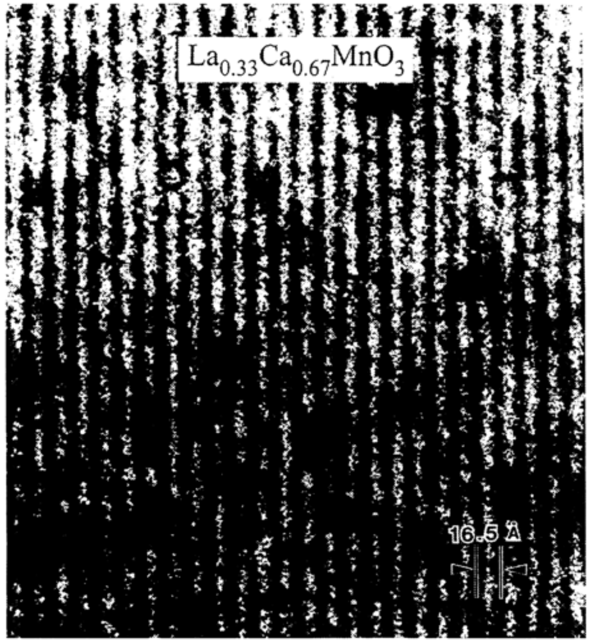
$$H_{Hund} = -J_H \sum_i \vec{S}_{t2gi} \cdot \vec{S}_{egi}$$ (Hund coupling between e_g & t_{2g} spins)

$$H_{on\,site} = -\sum_{\mathrm{i}} (\tilde{\beta} T_i^2 + \tilde{\alpha} S_{egi}^2)$$ (Coulomb interaction between e_g-electrons)

$$H_S = J_S \sum_{(ij)} \vec{S}_{t2gi} \cdot \vec{S}_{t2gj}$$ (Super exchange t_{2g} spins)

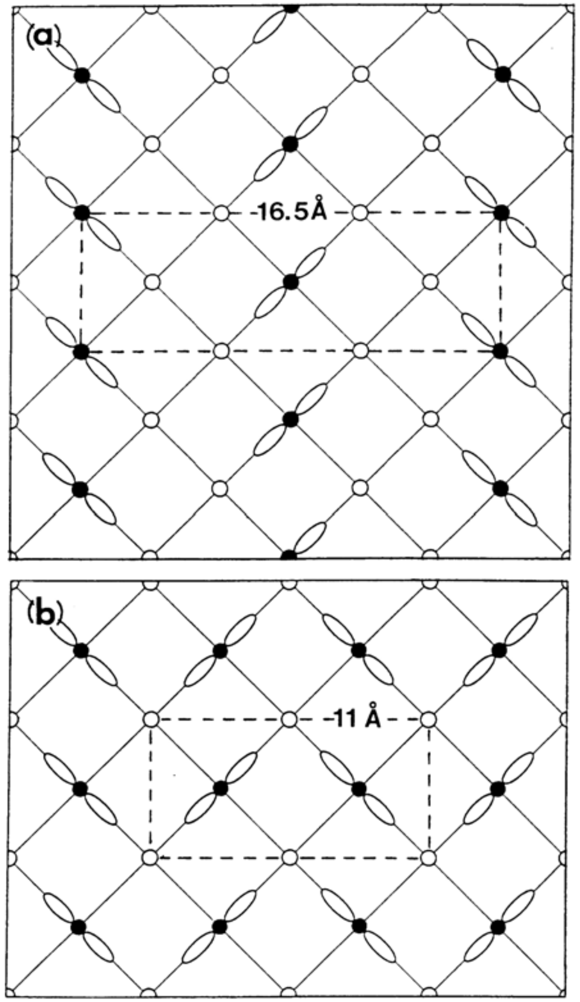

2D-F
1D-F
(x=0.0) ; spin F
(x=0.3) ; spin F
(0.3<x<0.8) ; spin F
(x=0.8) ; spin F
A
(x=0.0) ; spin A
(0.1<x<0.45) ; spin A
(0.45<x<0.75) ; spin A
(x=0.0) ; spin C
(x ≠ 0.0) ; spin C
G
(x=0.0) ; spin G

Charge Ordering (Mn^{+4}/Mn^{+3} = n/8)



Chen et al., J. Appl. Phys. 81 (1997)

Mn^{+3} Mn^{+4} Mn^{+4} Mn^{+3}

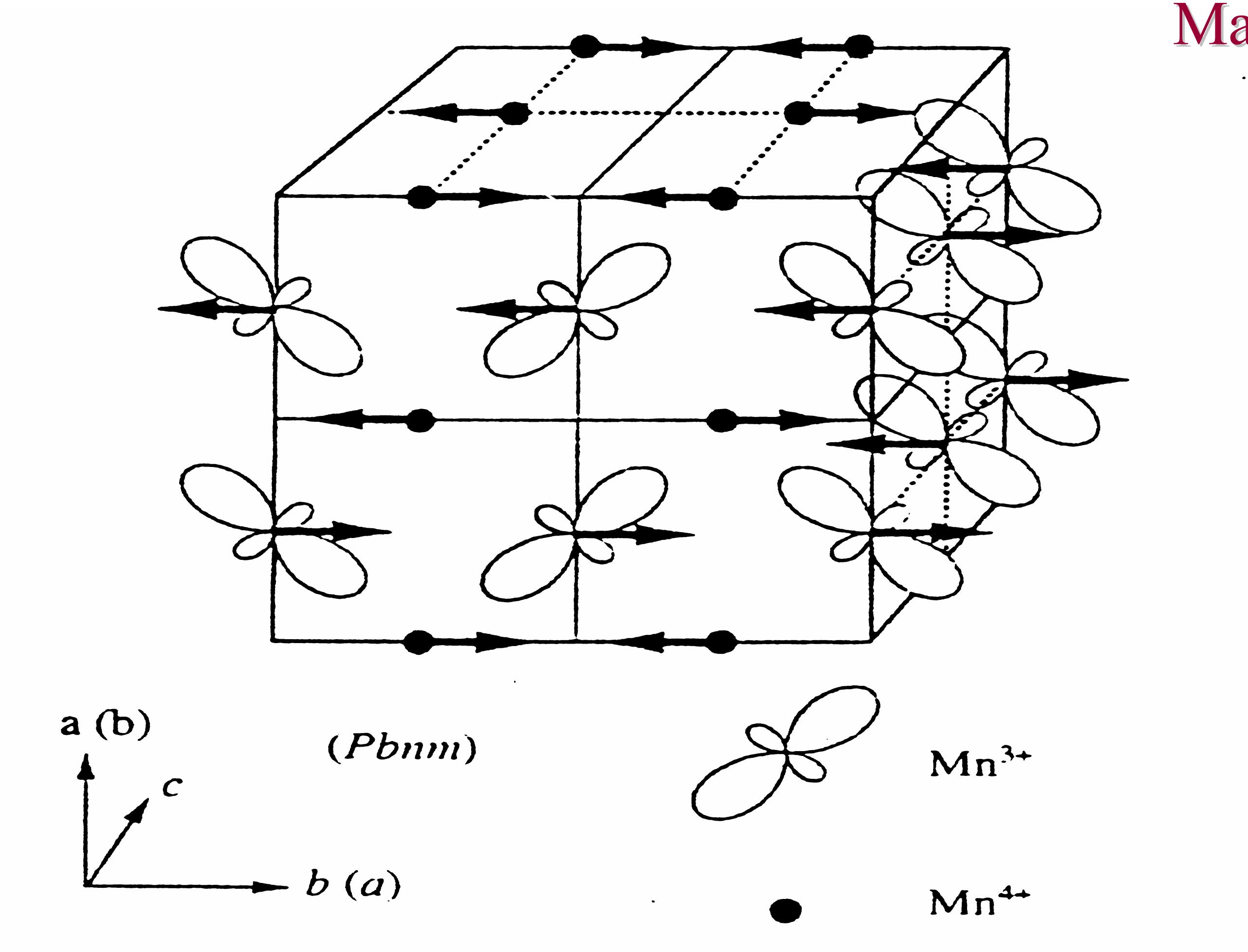


(0.67~5/8)

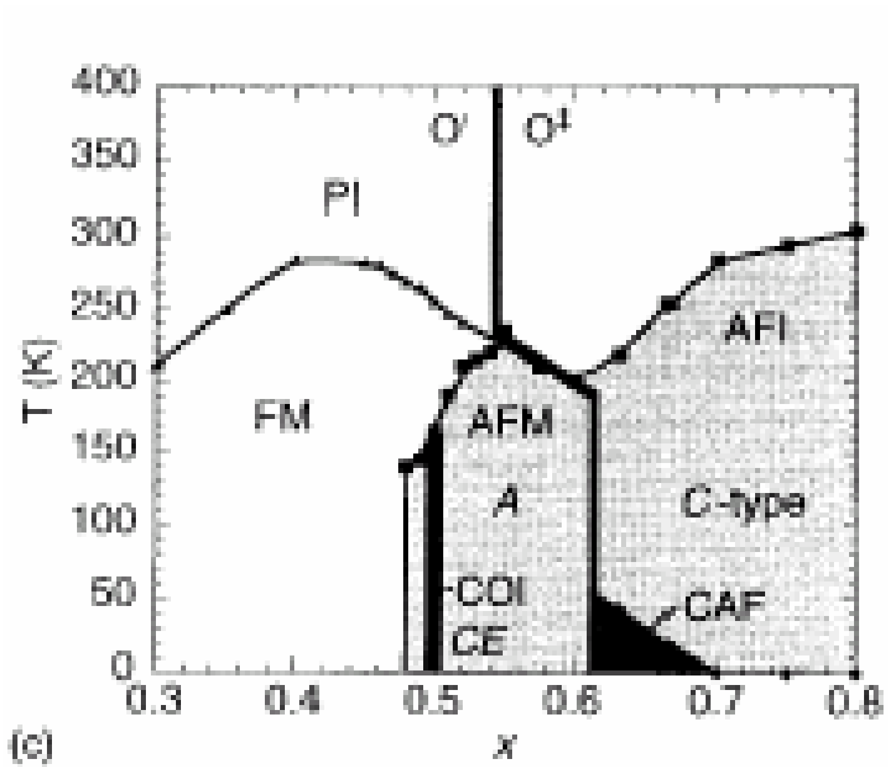
(0.5=4/8)

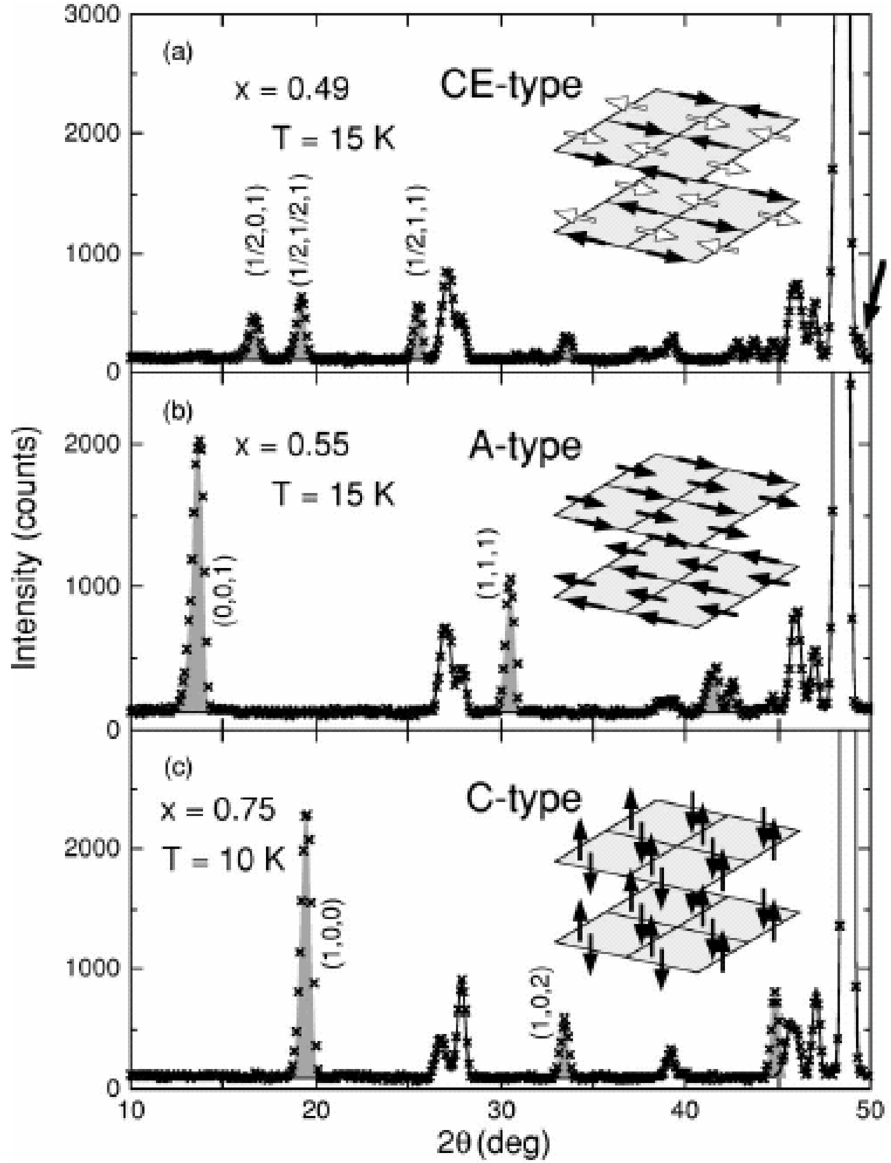
Spin/orbital structure

Neutron diffraction
Magnetic Dichroism



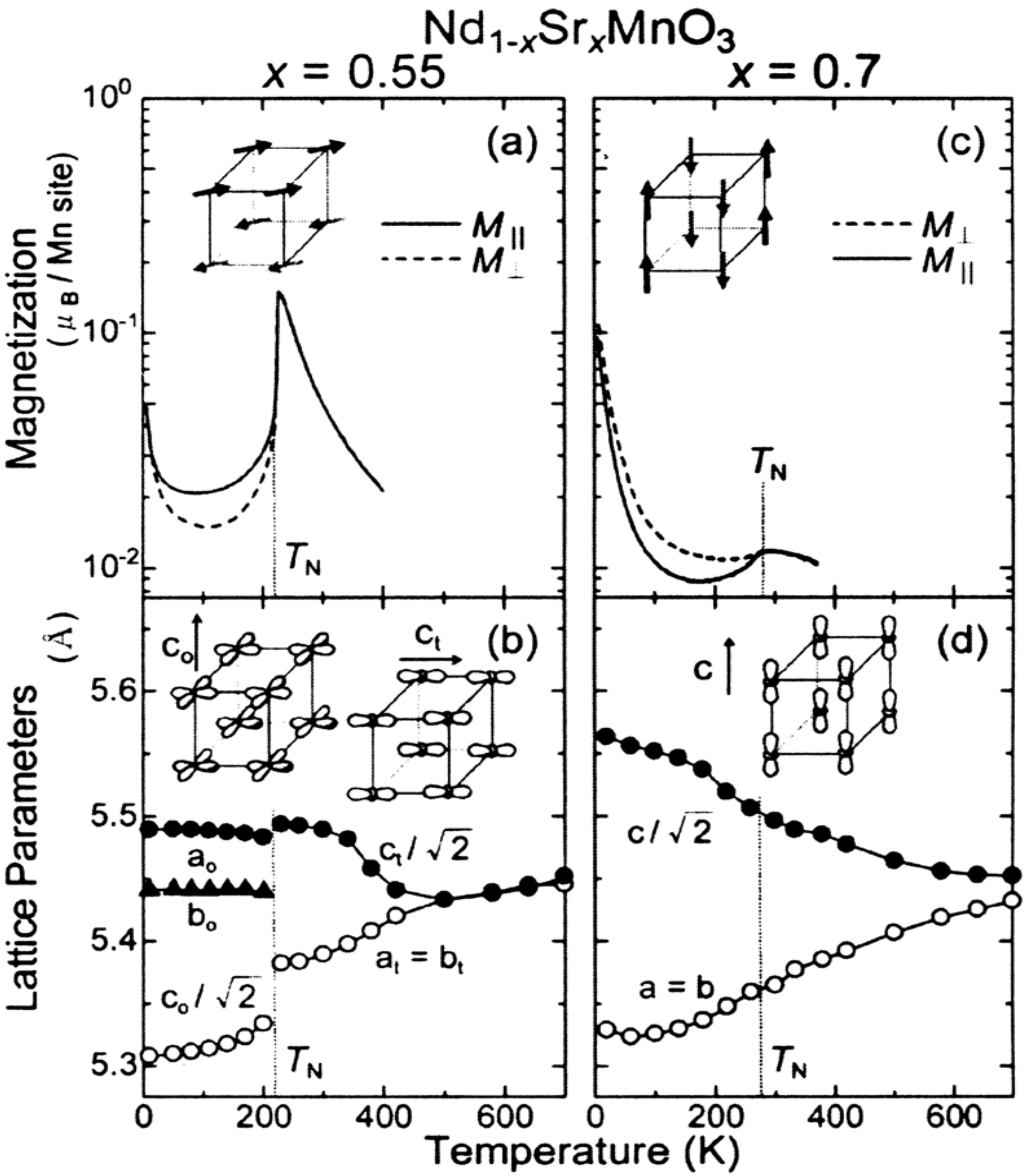
$Nd_{1-x}Sr_xMnO_3$





Kajimoto et al., PRB 30, 9506 (1999)

Orbital switching in A-type spin state



Spin A-Type :

from d_{3z2-r2} to d_{x2-y2}

at T_N ~ 220 K

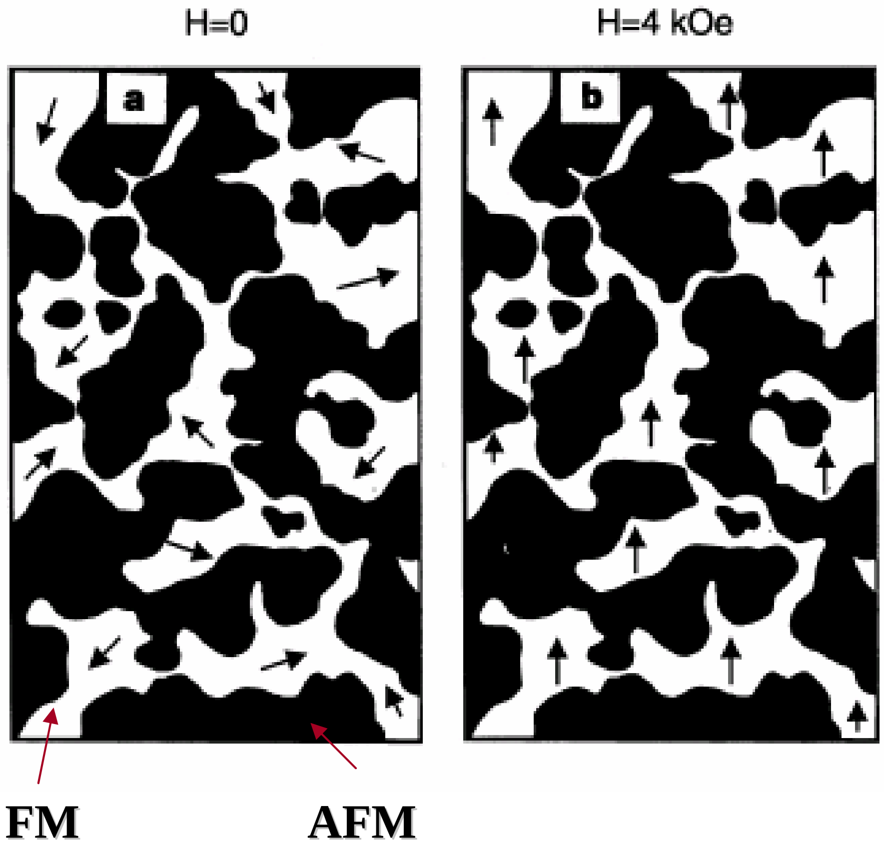
Spin C-type :

remains d_{3z2-r2}

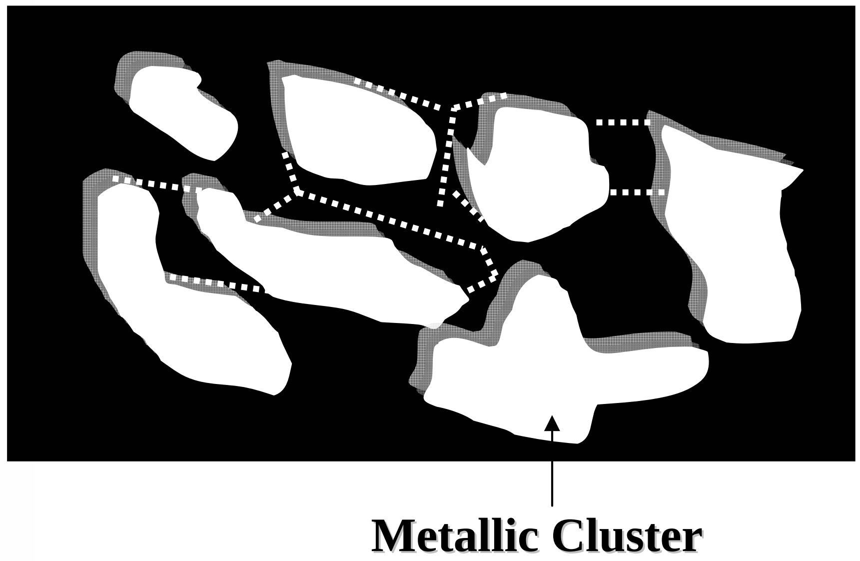
with T_N ~ 270 K

Tobe, PRB 67, 140402 (2003)

Origin of CMR: Phase separation



AFM Melting as H≧1 Tesla (insulator)



Spin line up (metal)

M. Uehara, et all, Nature 399 (1999)

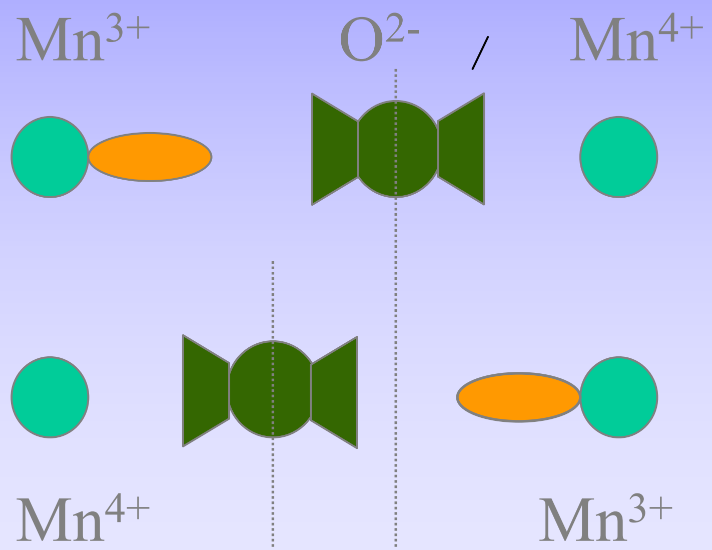
4-a. Experiment --- bulk

Phase separation

--- fine tuning the MR value by Ionic radius size

Key parameters

- hole concentration (charge/orbital/spin)
- radius of A-site (lattice/ spin)



Bulk Making

Step 1 Pre-heating R_2O_3: 900℃/ 3 h .

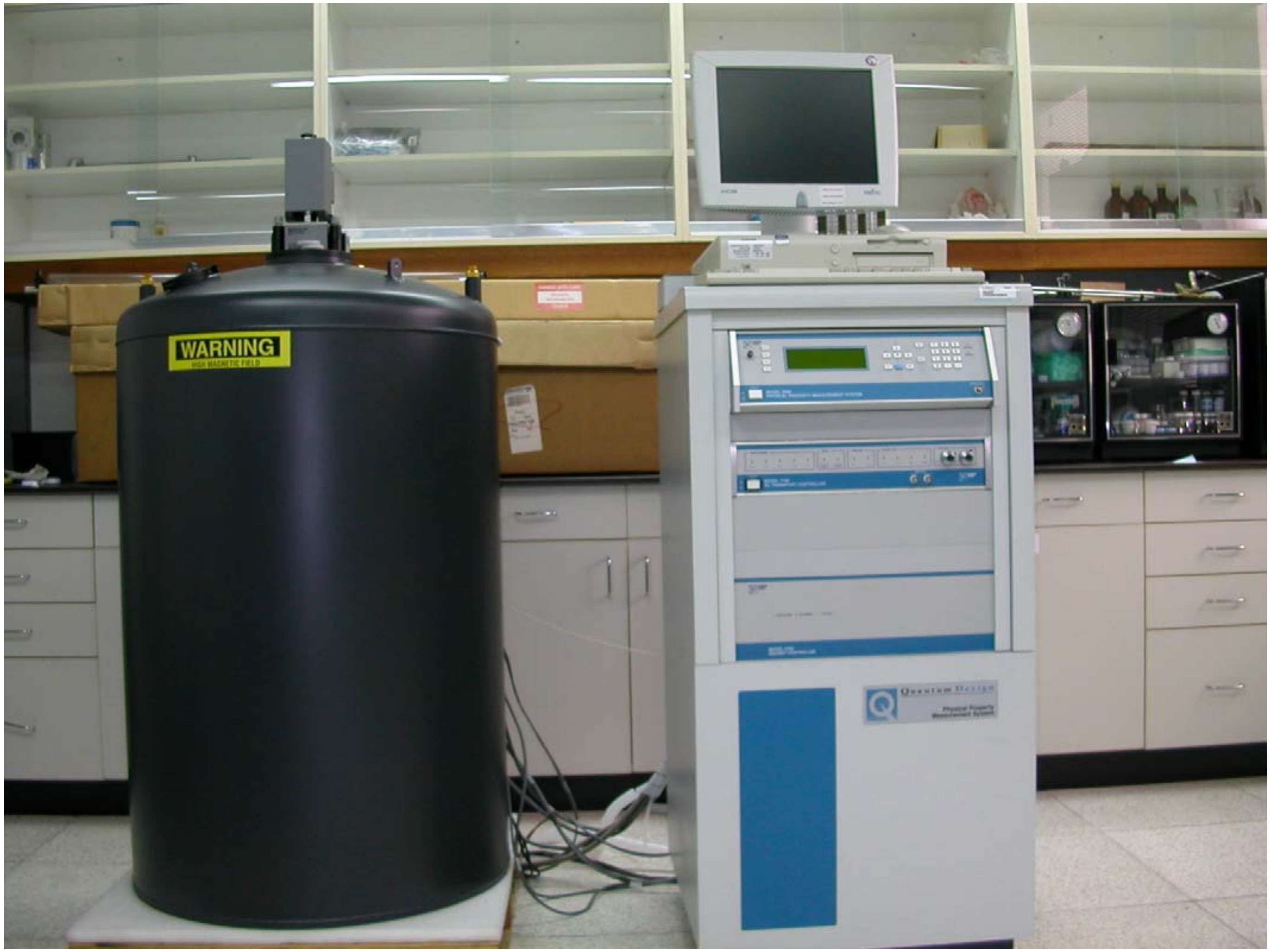
Step 2 Mix R_2O_3 ,$CaCO_3$, $SrCO_3$, $MnCO_3$

Step 3 Reaction: 1200℃/ 24 h

Step 4 Pellet :d = 1 cm, Thickness =3 mm, 3 tons/ cm^{-2}

Step 5 Anneal: 1400℃ /16 h

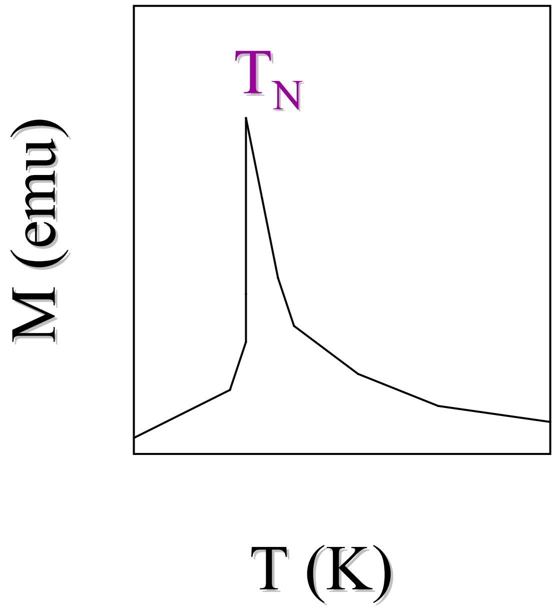
Physical Property measurement System (PPMS)



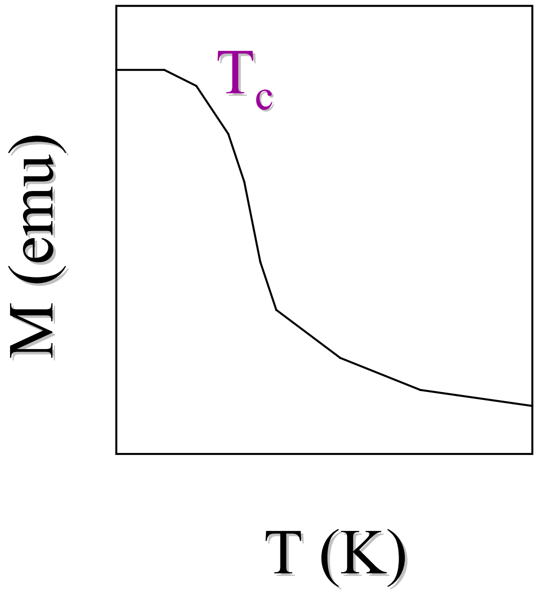
0 – 7 Tesla
1.4 – 400 K
Hall effect
resisitivity
AC susceptibility

Basic M – T curve

Para- to antiferromagnetic



Para- to ferromagnetic

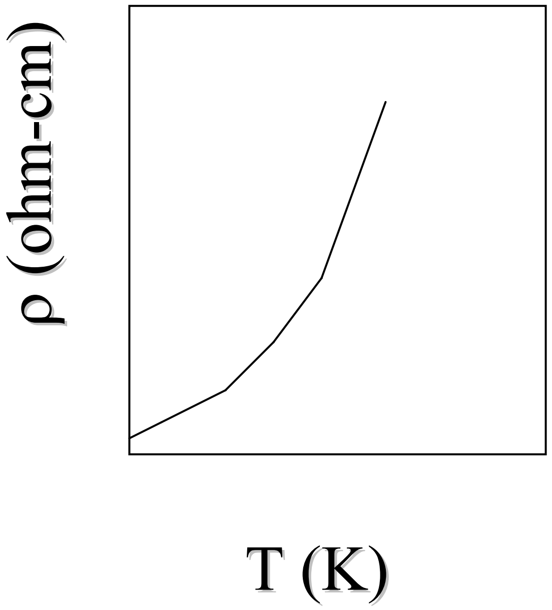


$M \sim C/T$, $C/(T+T_N)$, $C/(T-Tc)$

Basic R – T curve

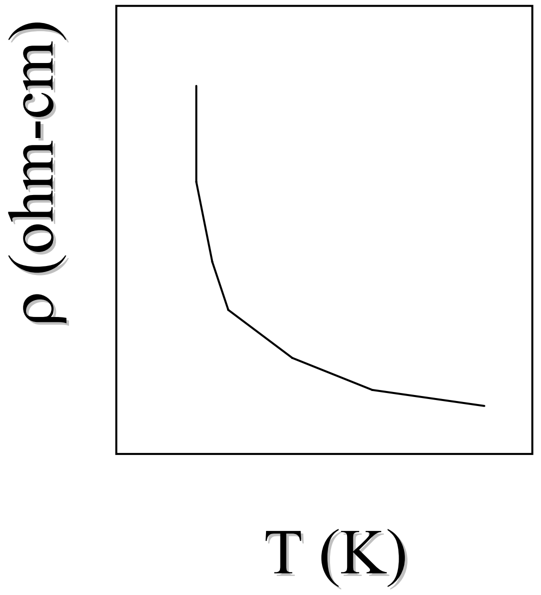
$\rho = \rho_o + \rho_1 T^{\alpha}$

metal



$\rho = \rho_o \exp(C/T^{\beta})$

insulator

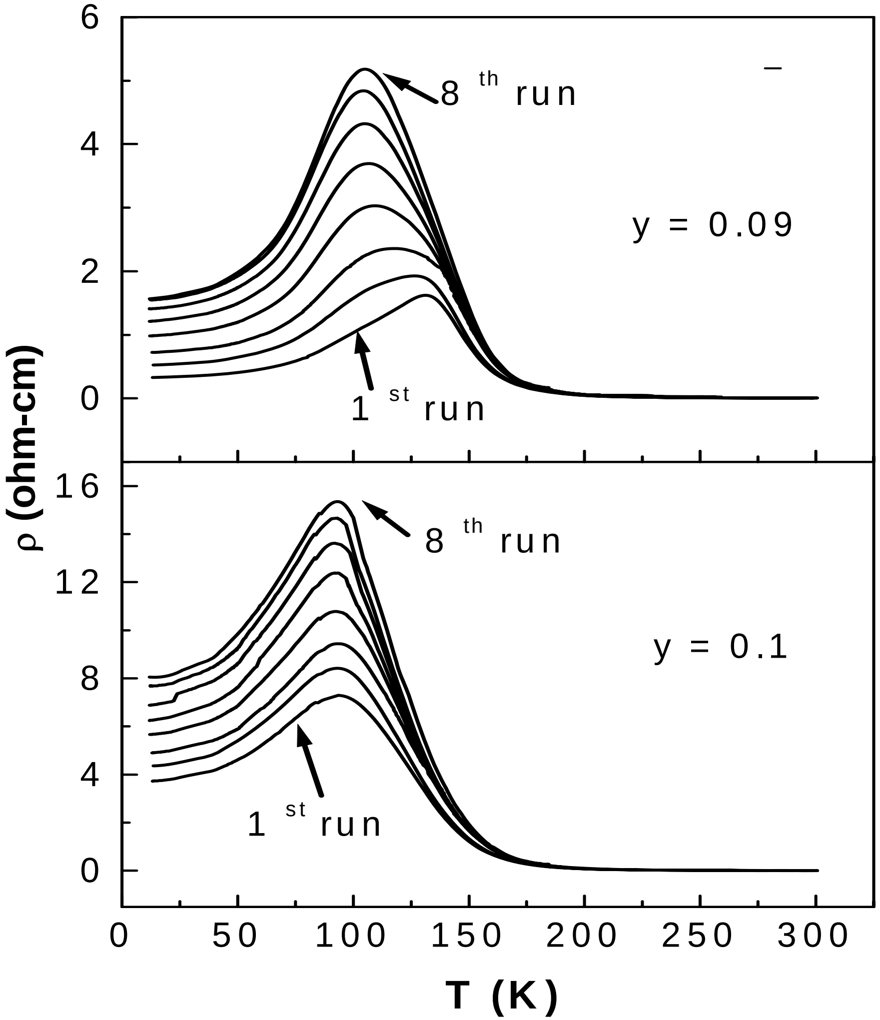


PHYSICAL REVIEW B, VOLUME 65, 024422 (2001)

Thermal and magnetic instability near the percolation threshold of $Nd_{0.5}Ca_{0.5-y}Sr_yMnO_3$

C. W. Chang,* A. K. Debnath,† and J. G. Lin‡

Center for Condensed Matter Science, National Taiwan University, Taipei, Taiwan



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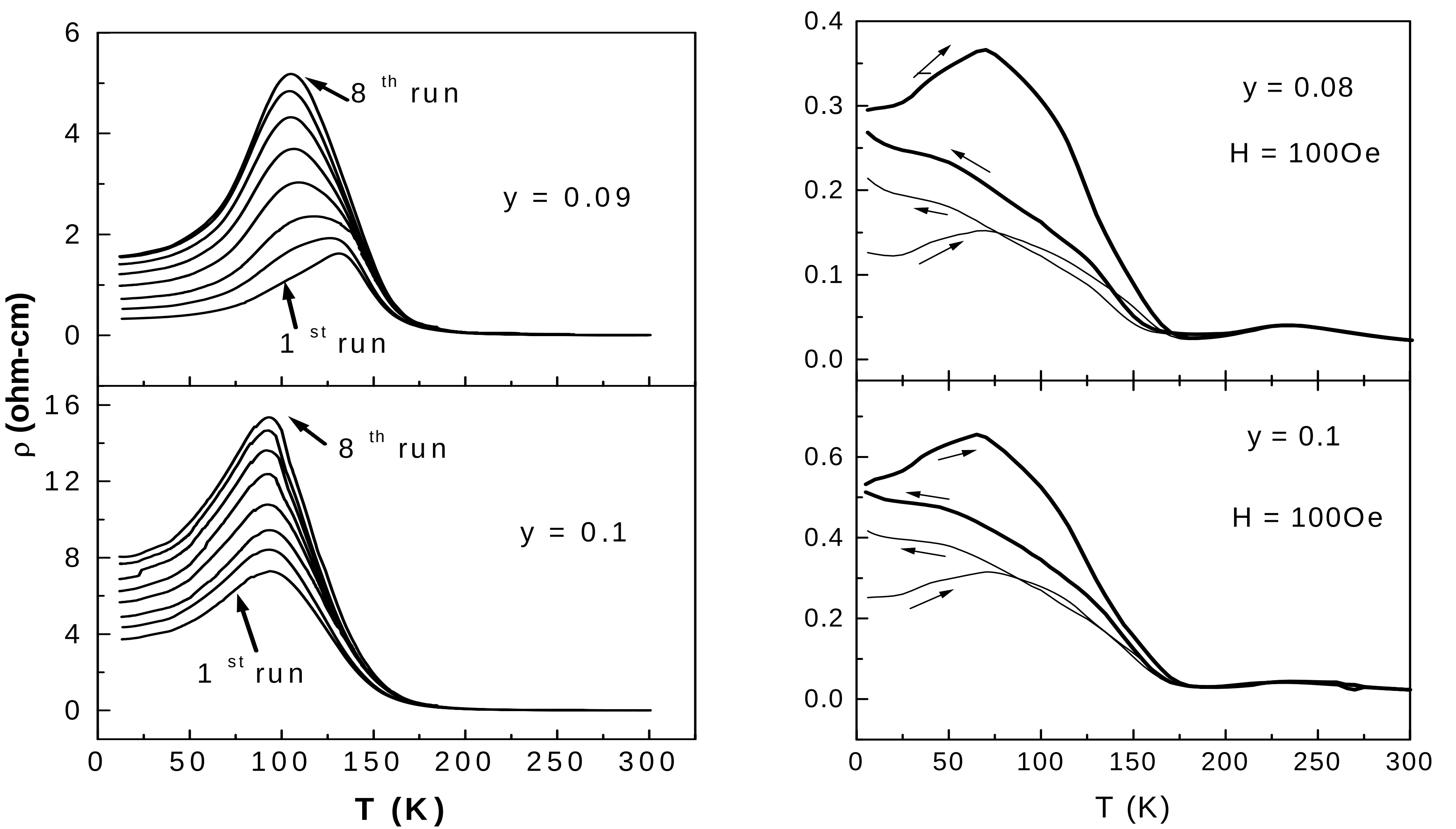

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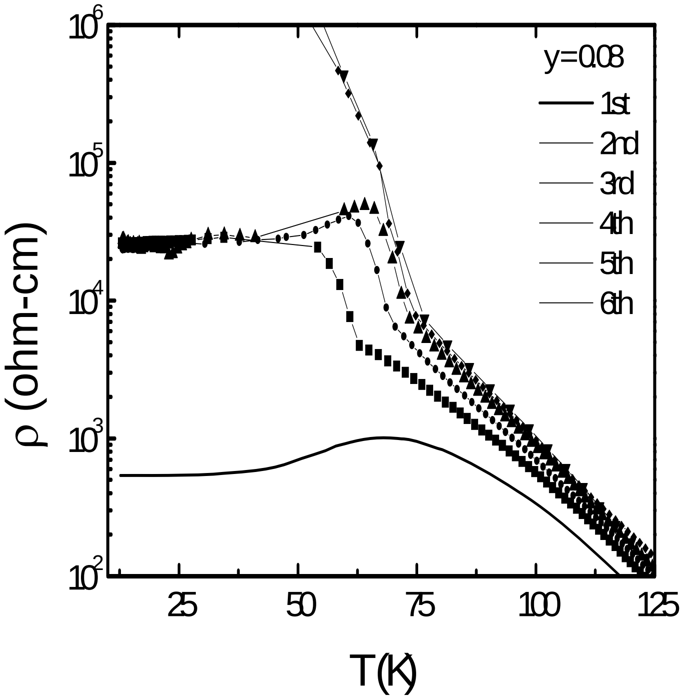

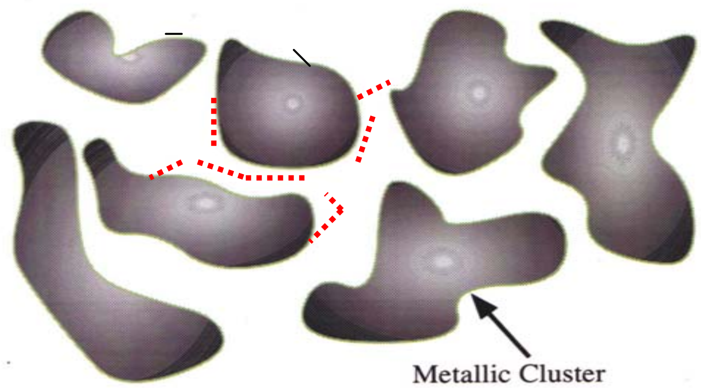

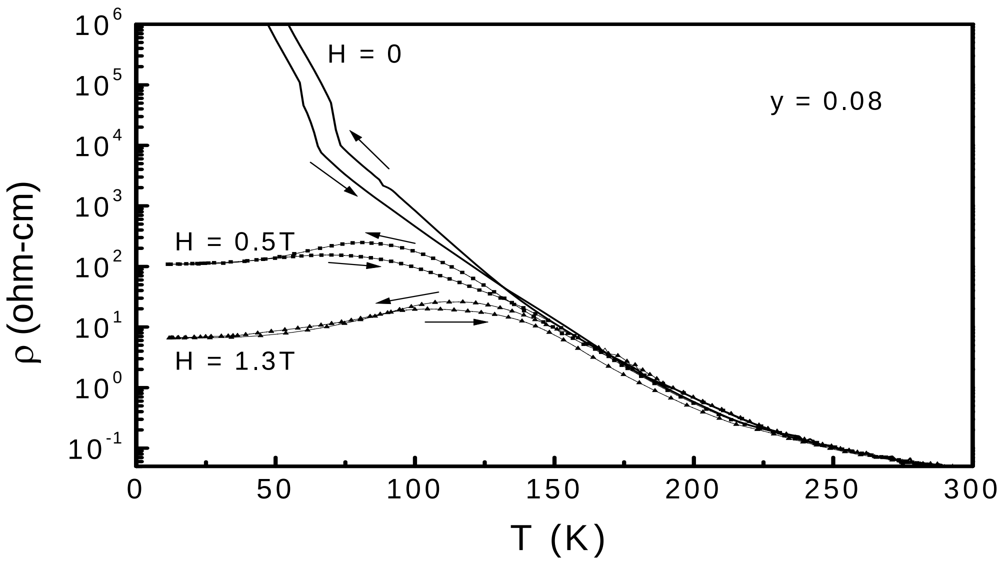

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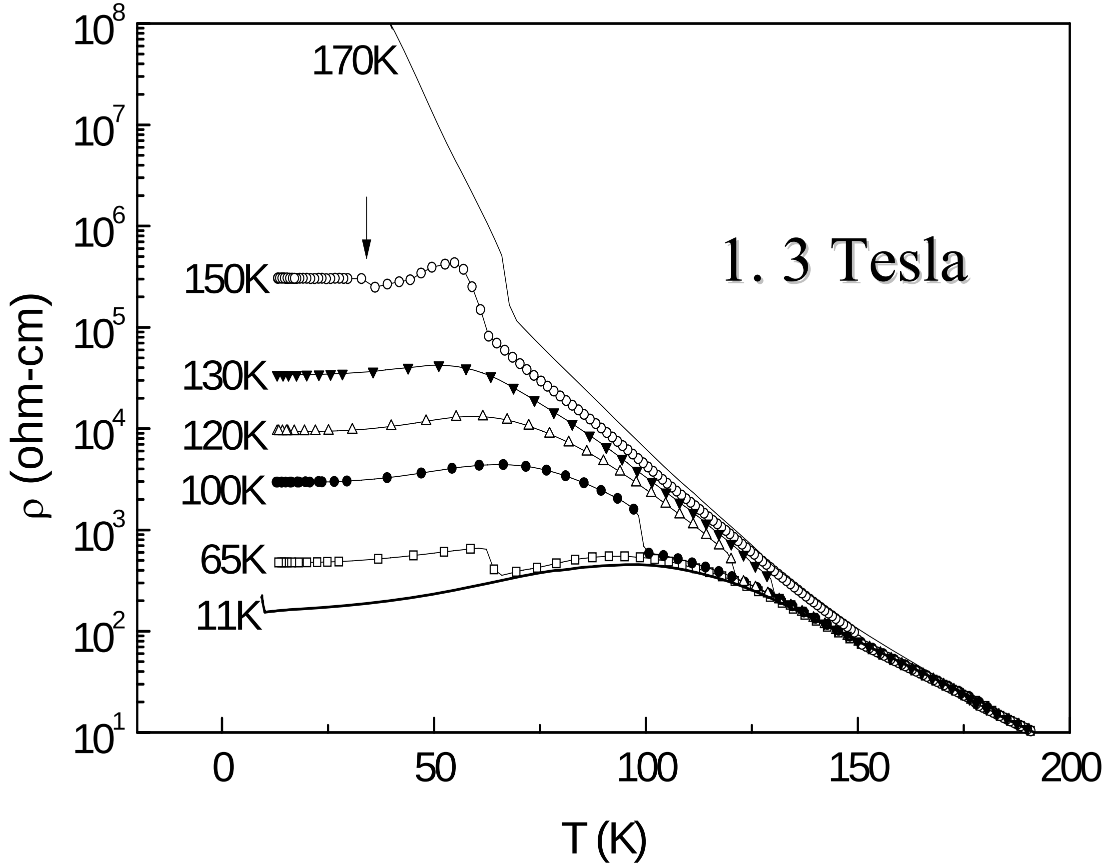

PHYSICAL REVIEW B, VOLUME 65, 024422 (2001)

Thermal and magnetic instability near the percolation threshold of $Nd_{0.5}Ca_{0.5-y}Sr_yMnO_3$

C. W. Chang,* A. K. Debnath,† and J. G. Lin‡

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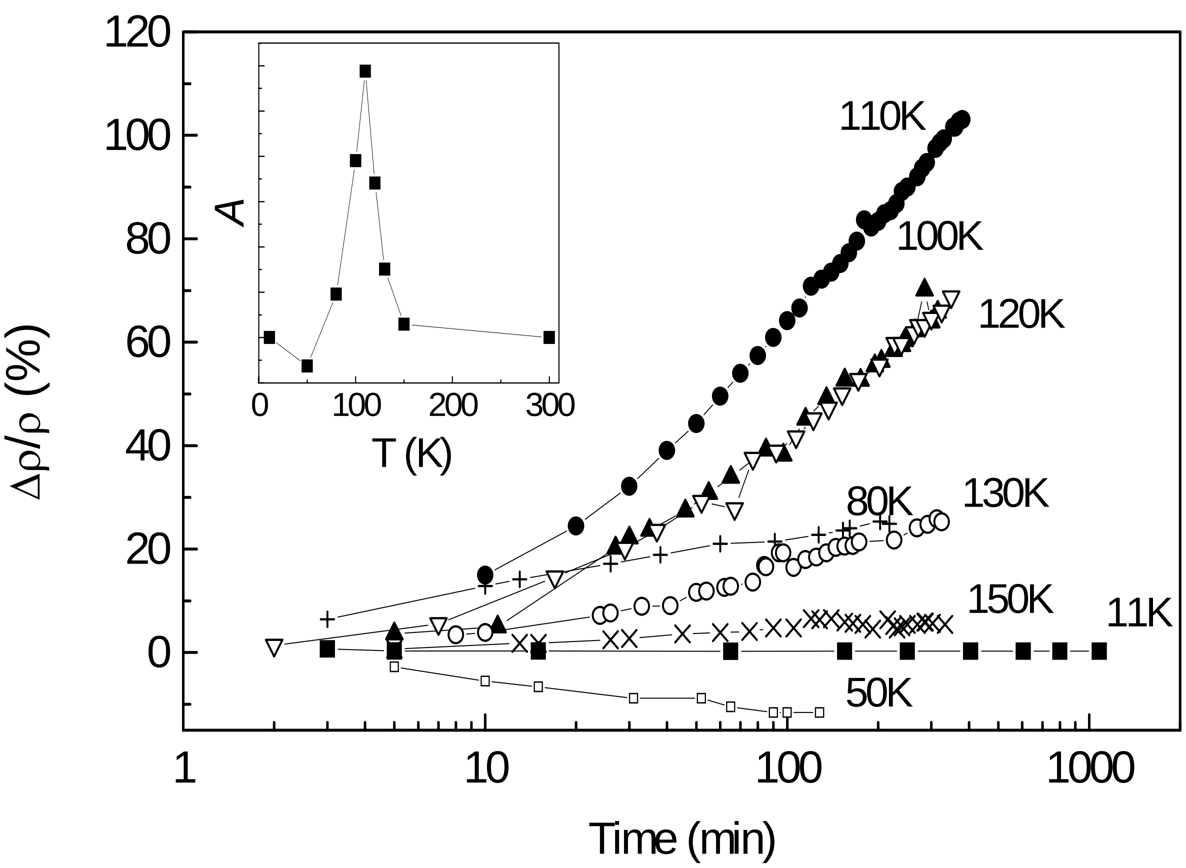

$$\Delta\rho/\rho = A\ \ln t + \text{const.}$$

J OF APPLIED PHYSICS VOLUME 90 (2001)

Enhancement of magnetoresistance in the intermediate state of Pr0.5Sr0.3Ca0.2MnO3

C. W. Chang and J. G. Lin

Center for Condensed Matter Science, National Taiwan University, Taipei, Taiwan 10764

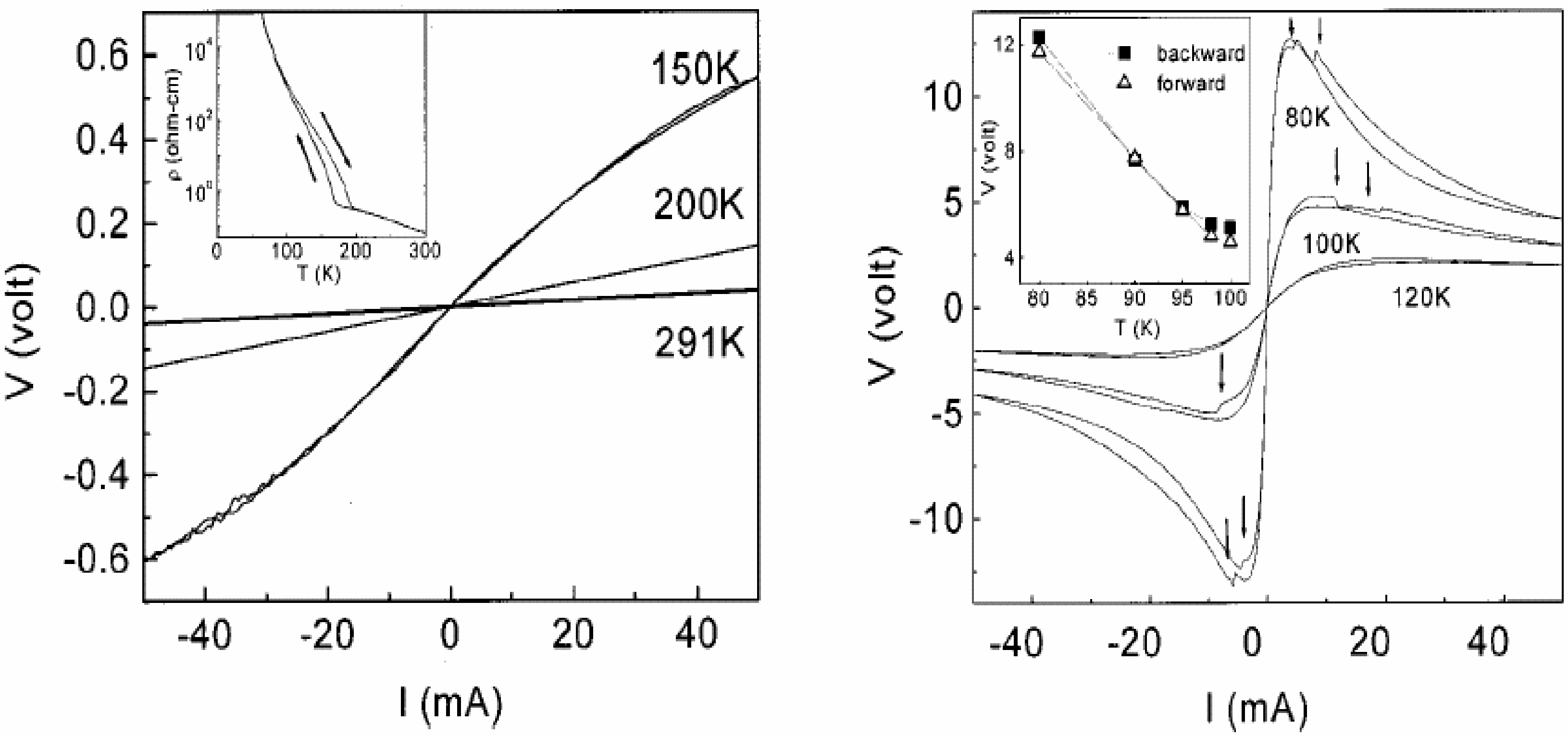

FIG. 2. *I –V* curves for Pr0.5Sr0.3Ca0.2MnO3 at 120, 100, and 80 K, respectively. The arrows denote the jumps of voltage. The inset shows the temperature dependence of the onset voltage of the jump with the forward scan (increasing) and the backward scan (decreasing).

J OF APPLIED PHYSICS VOLUME 90 (2001)

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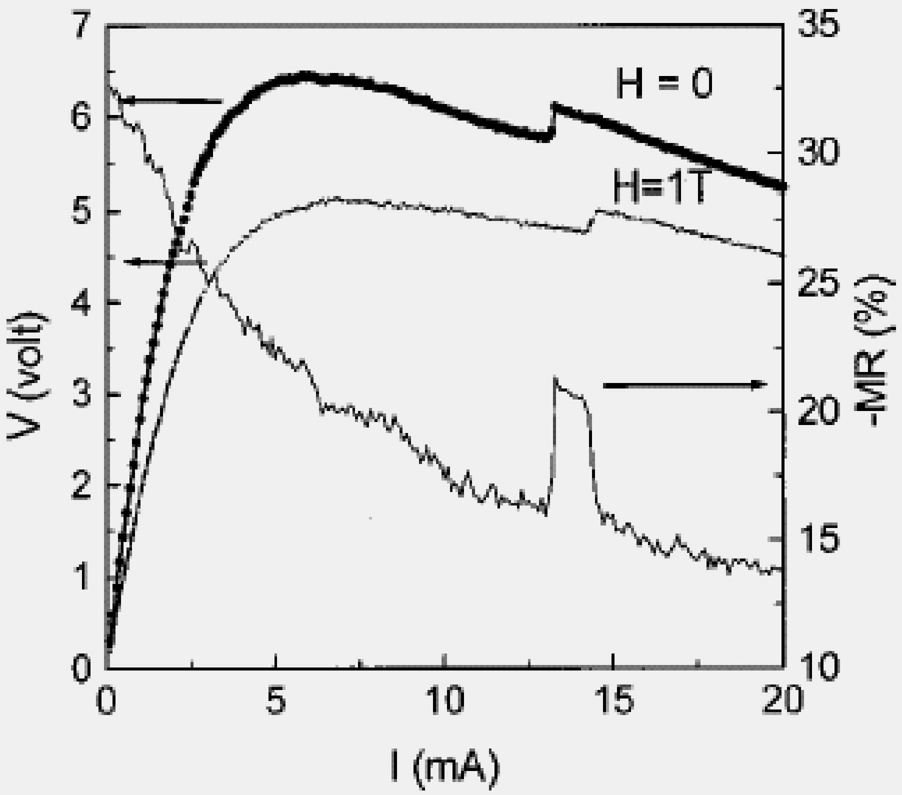

FIG. 3. Current dependence of *V* and MR ratio at 95 K with different magnetic fields.

$$MR=\frac{R(H)-R(0)}{R(0)}=\frac{V(H)-V(0)}{V(0)},$$

J OF APPLIED PHYSICS VOLUME 90 (2001)

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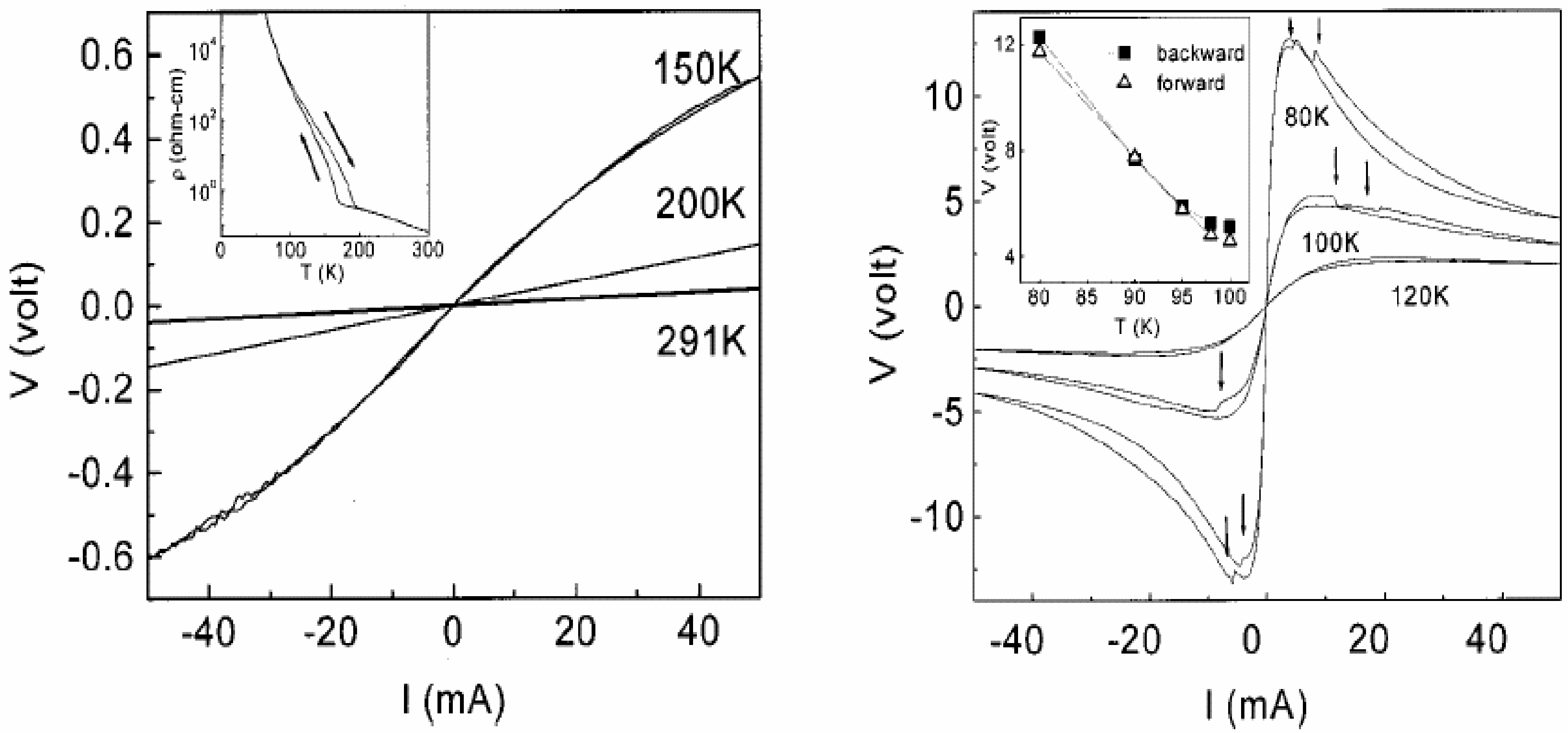


FIG. 2. *I –V* curves for Pr0.5Sr0.3Ca0.2MnO3 at 120, 100, and 80 K, respectively. The arrows denote the jumps of voltage. The inset shows the temperature dependence of the onset voltage of the jump with the forward scan (increasing) and the backward scan (decreasing).

4-b. Experiment : films & bilayers

Subject (1): Nanocrystalline LSMO films

--- High electroresistance

& low field MR effects

Experiment Flow Chart

Target preparation (Solid State Reaction)

↓

Structure Analysis (XRD)

↓

Thin Film Deposition (Sputtering,PLD)

↓

Thin Film Charaterization

Composition — *EDS*

Magnetism — *PPMS & SQUID*

Resistance — *PPMS*

Morphology — *TEM*

Target preparation

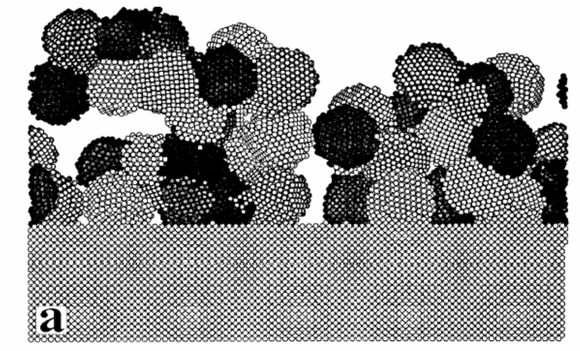
❖ $Nd_{0.7}Ca_{0.3}MnO_3$ (NCMO):

Nd_2O_3, $CaCO_3$, $MnCO_3$ Powder
100°C (3 hrs.) in air
1100°C (24 hrs.) in air

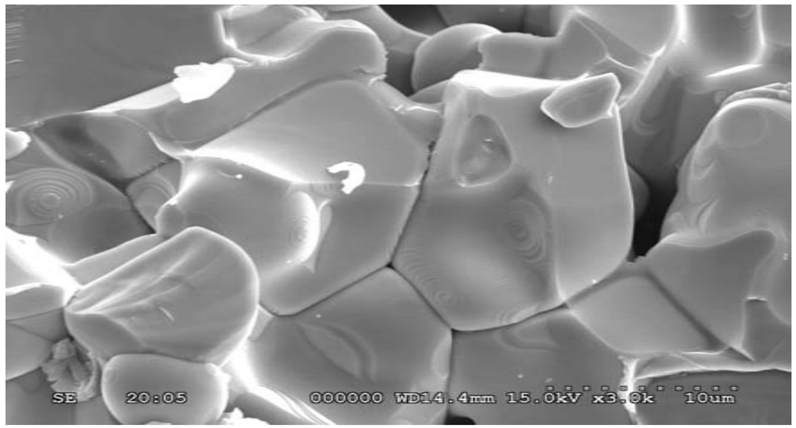
❖ $YBa_2Cu_3O_7$ (YBCO)

Y_2O_3, $BaCO_3$, CuO_2 Powder
100°C (3 hrs.) in air
1050°C (36 hrs.) in oxygen
400°C (12 hrs.) in oxygen

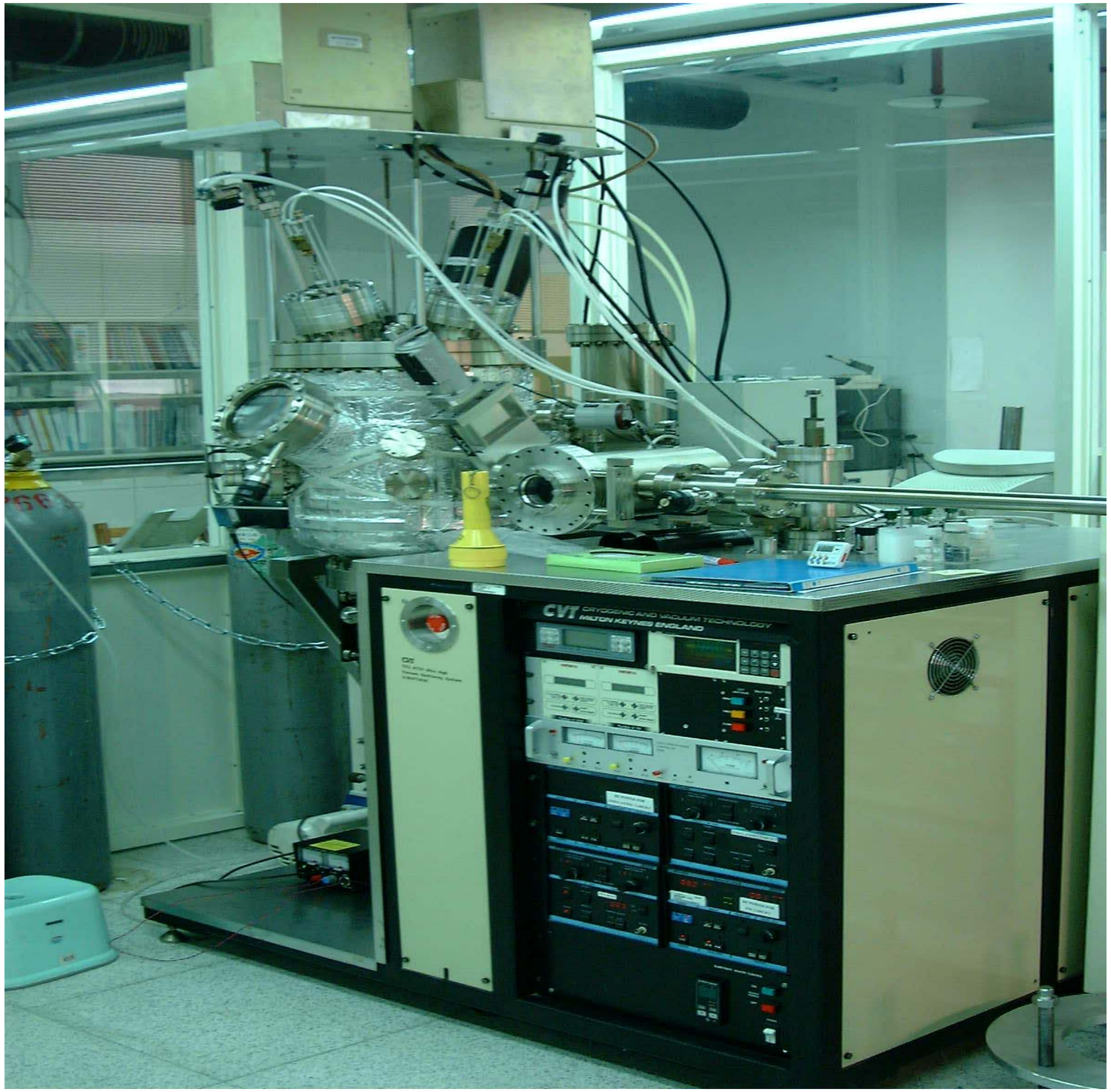
Powders



Solid state reaction

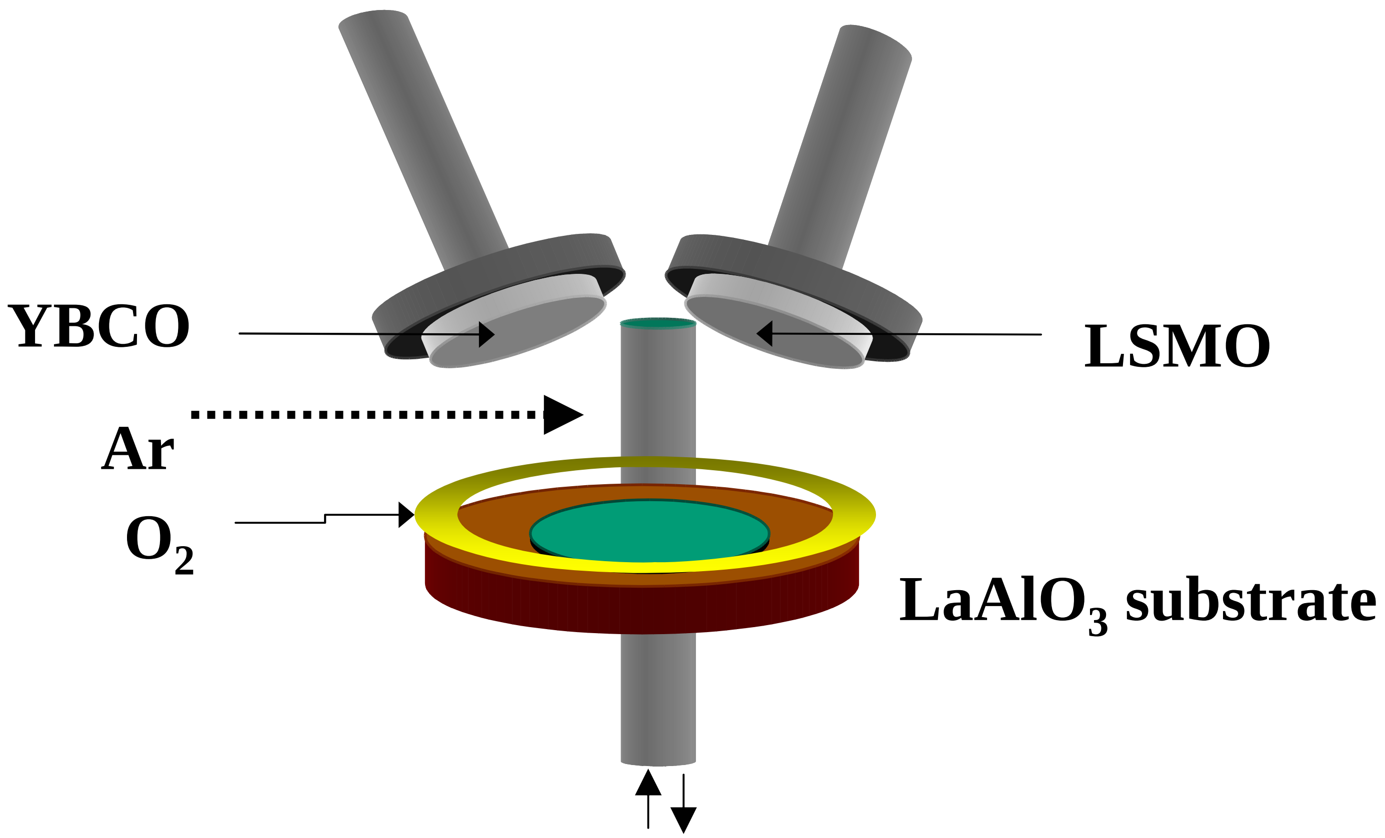


RF sputter system (Millton CVT, 13.6 MHz)



10^{-8} torr
four guns
1000 °C

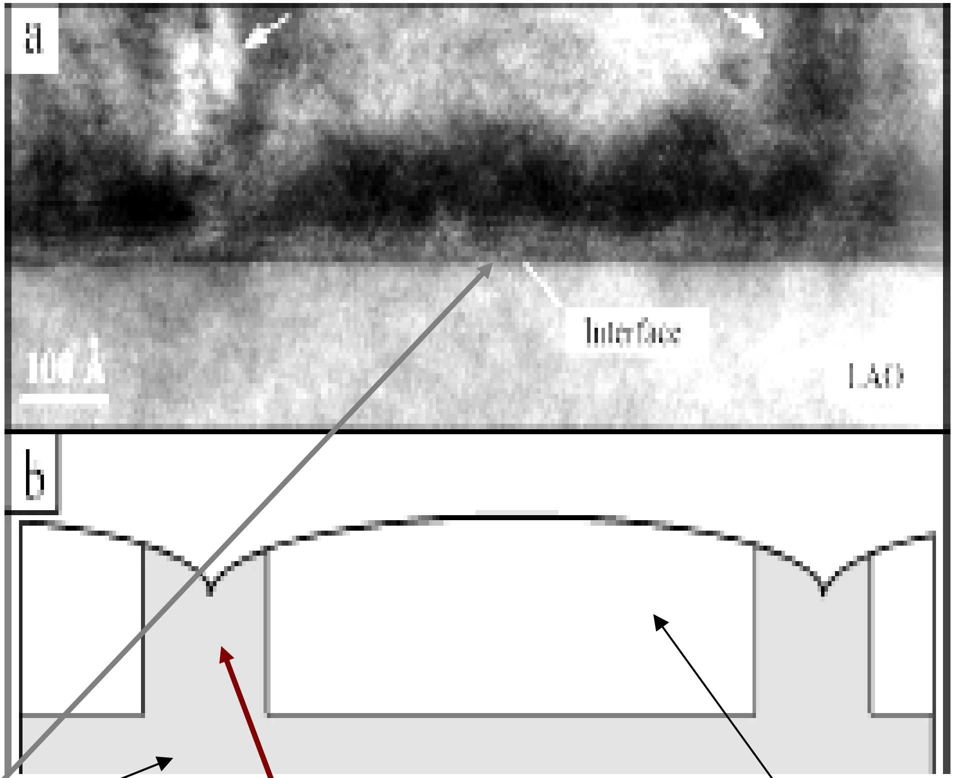
Film making



Reactive co-sputtering process

• Substrate ⇒	Si (100) , $LaAlO_3$(100)
• Target ⇒	YBCO, LSMO
• RF power ⇒	80 Watt
• Base pressure ⇒	3×10^{-7} torr
• Mixed gas ⇒	$Ar:O_2=98:2$
• Sputtering pressure	70 mtorr
• Base temperature ⇒	Room temperature
• pre-sputtering ⇒	3 minutes
• Working distance ⇒	10 cm
• Annealing tempert.	800 – 920 ℃ (700 ℃)
• Annealing time ⇒	1 hrs

Non-uniform strain in $La_{0.67}Ca_{0.33}MnO_3$ film



Lattice parameter a

LCMO=3.86Å

NGO(110)=3.86 Å (no strain)

LAO(100)=3.79 Å (Compressive)

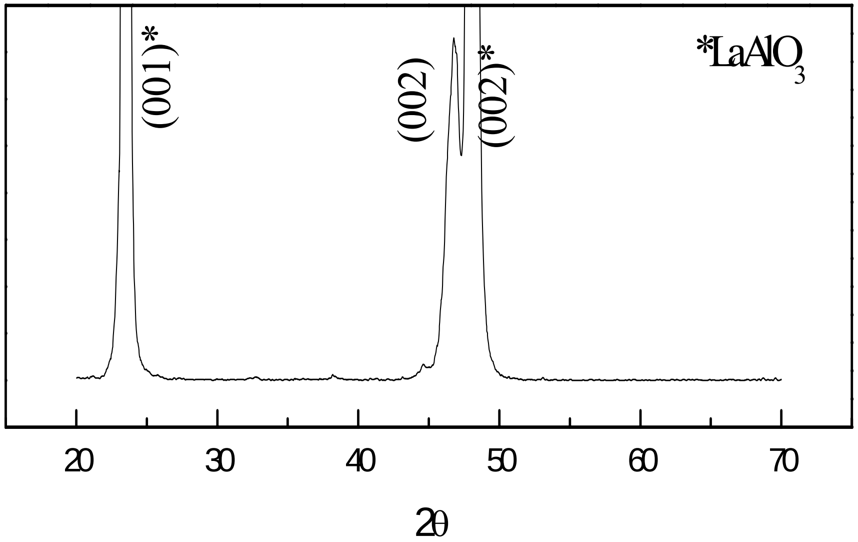
STO(100)=3.91 Å (Tensile)

PRB 61, 9665 (2000)

Structures of LSMO layer

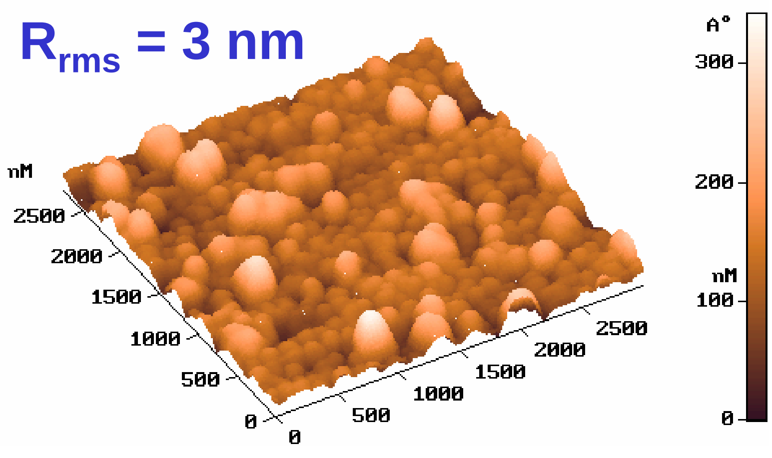
XRD – monoclinic

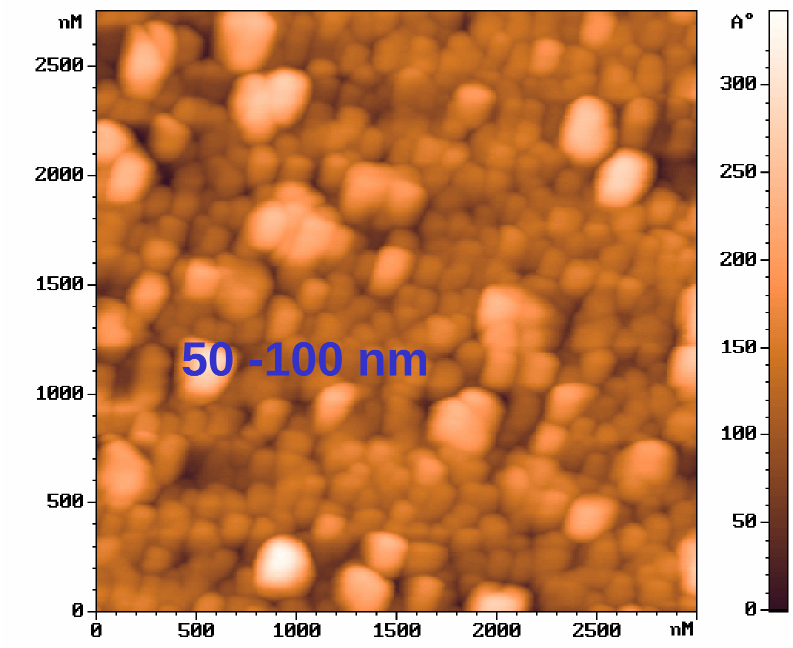
c-oriented



AFM - granular

R_{rms} = 3 nm

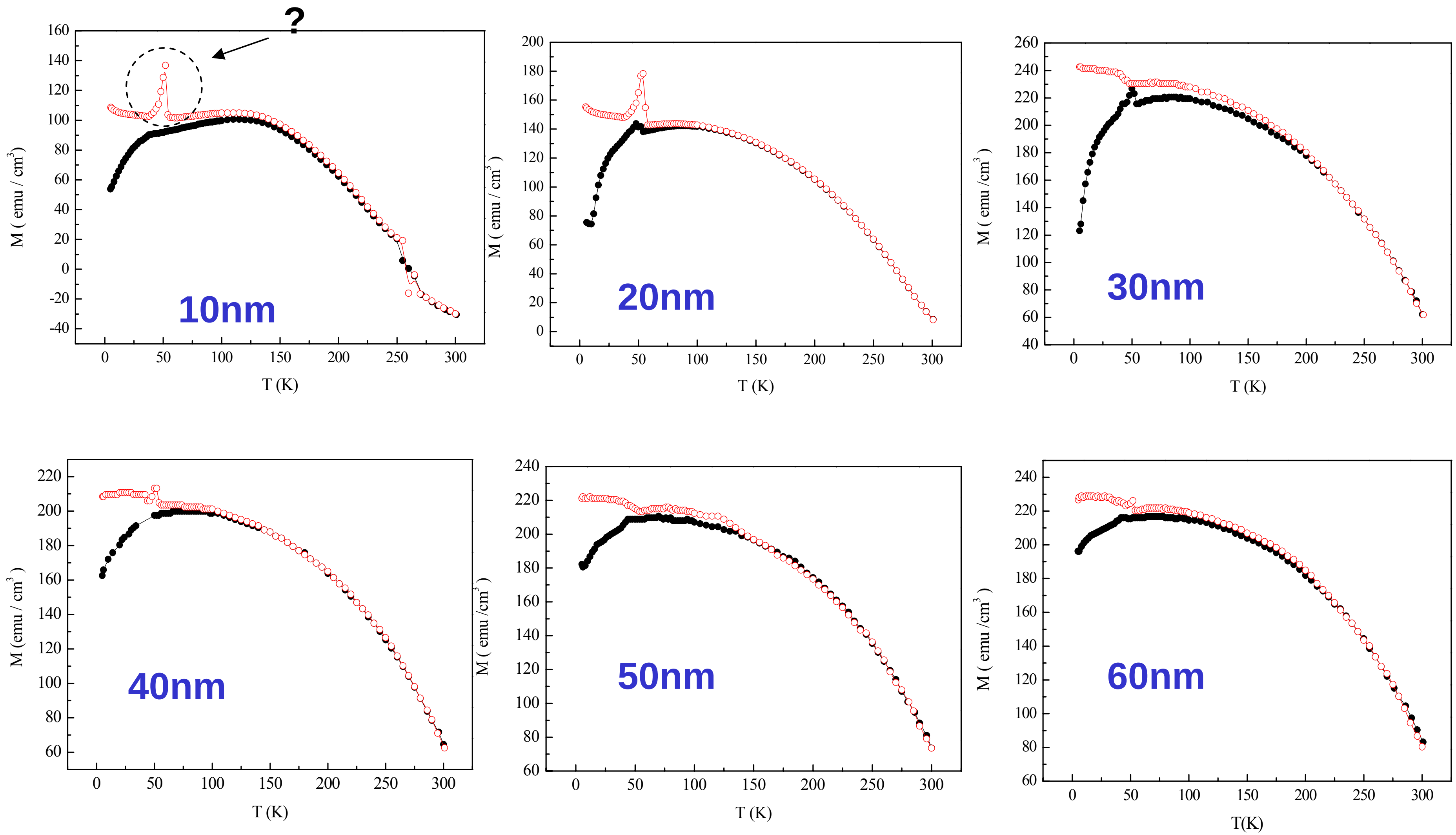


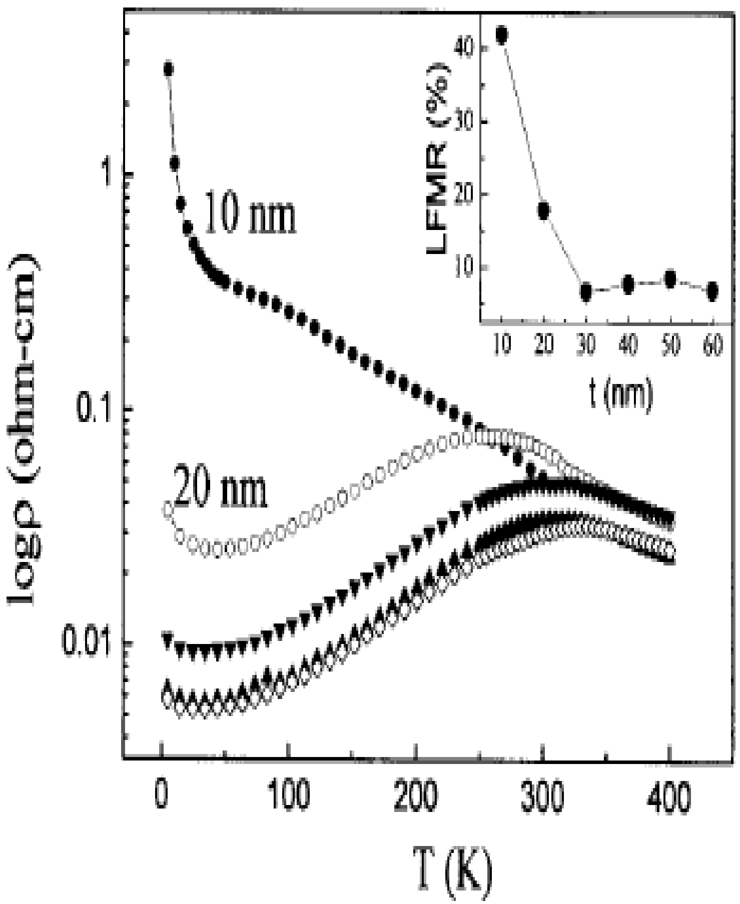


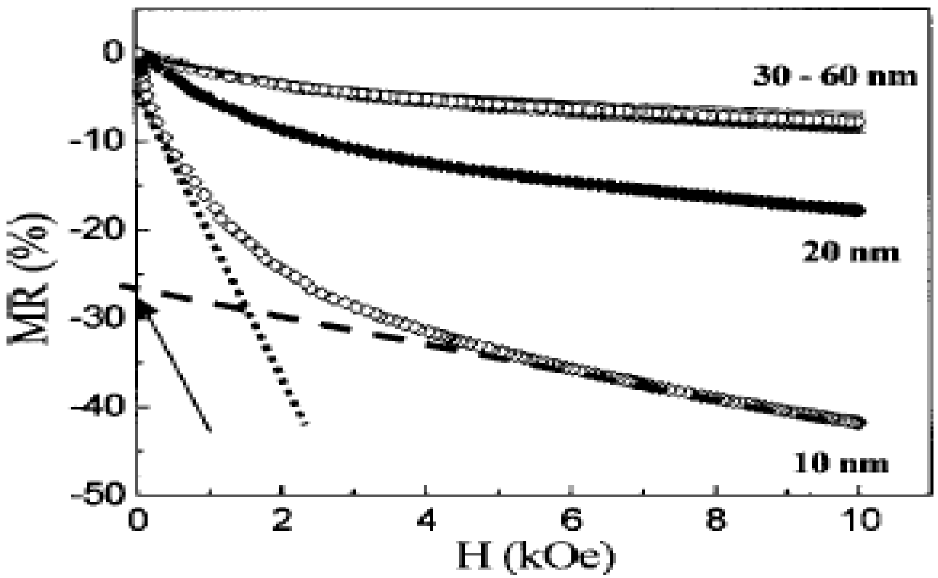
Magnetization for LSMO

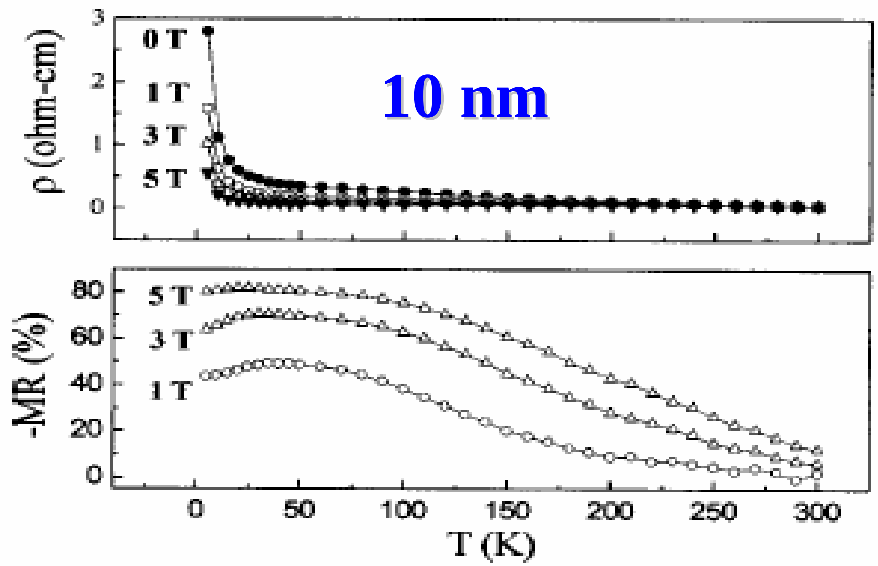
H=500 Gauss

ZFC(black line); FC (red line)




logρ (ohm-cm)
1
0.1
0.01
10 nm
20 nm
0
100
200
300
400
T (K)
LFMR (%)
40
30
20
10
10
20
30
40
50
60
t (nm)


MR (%)
0
-10
-20
-30
-40
-50
30 - 60 nm
20 nm
10 nm
0
2
4
6
8
10
H (kOe)


10 nm
ρ (ohm-cm)
3
2
1
0
0 T
1 T
3 T
5 T
-MR (%)
80
60
40
20
0
5 T
3 T
1 T
0
50
100
150
200
250
300
T (K)

Low-field magnetoresistance in nanocrystalline $La_{0.7}Sr_{0.3}MnO_3$ films

S. L. Cheng and J. G. Lin[a)]
Center for Condensed Matter Sciences/Center for Nanostorage Research, National Taiwan University, Taipei 106, Taiwan

(Received 28 March 2005; accepted 1 November 2005; published online 14 December 2005)

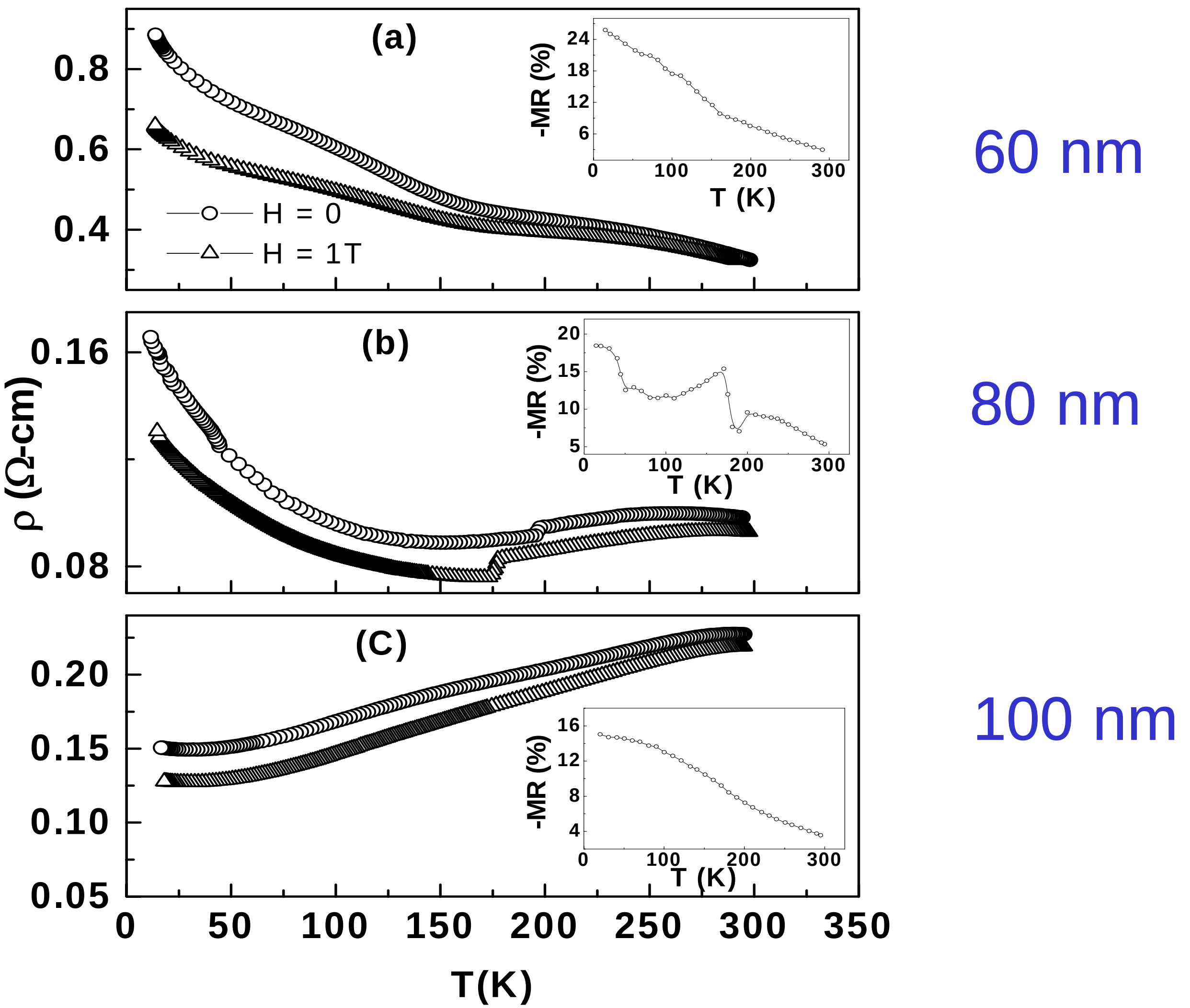
Nanocrystalline $La_{0.7}Sr_{0.3}MnO_3$ films with thickness t=10-60 nm were grown on $LaAlO_3$(100) substrates by radio-frequency magnetron sputtering. Their electrical resistivity and low-field magnetoresistance (MR) were measured. Metal-insulator transitions occur above 275 K for films with t=20-60 nm, but the electron localization prevails in the 10 nm thick film. Furthermore, only the 10 nm thick film has an MR that depends on the inverse of temperature, consistent with the model of spin-polarized tunneling. This relationship may reflect a critical aspect of the structure of grain/grain-boundaries. Accordingly, the tunneling MR in this film is 27% at 75 K. © *2005 American Institute of Physics*. [DOI: 10.1063/1.2140081]

PHYSICAL REVIEW B **67**, 064412 (2003)

Current-induced giant electroresistance in La0.7Sr0.3MnO3 thin films

A. K. Debnath* and J. G. Lin†

Center for Condensed Matter Sciences, National Taiwan University, Taipei 10617, Taiwan



PHYSICAL REVIEW B **67**, 064412 (2003)

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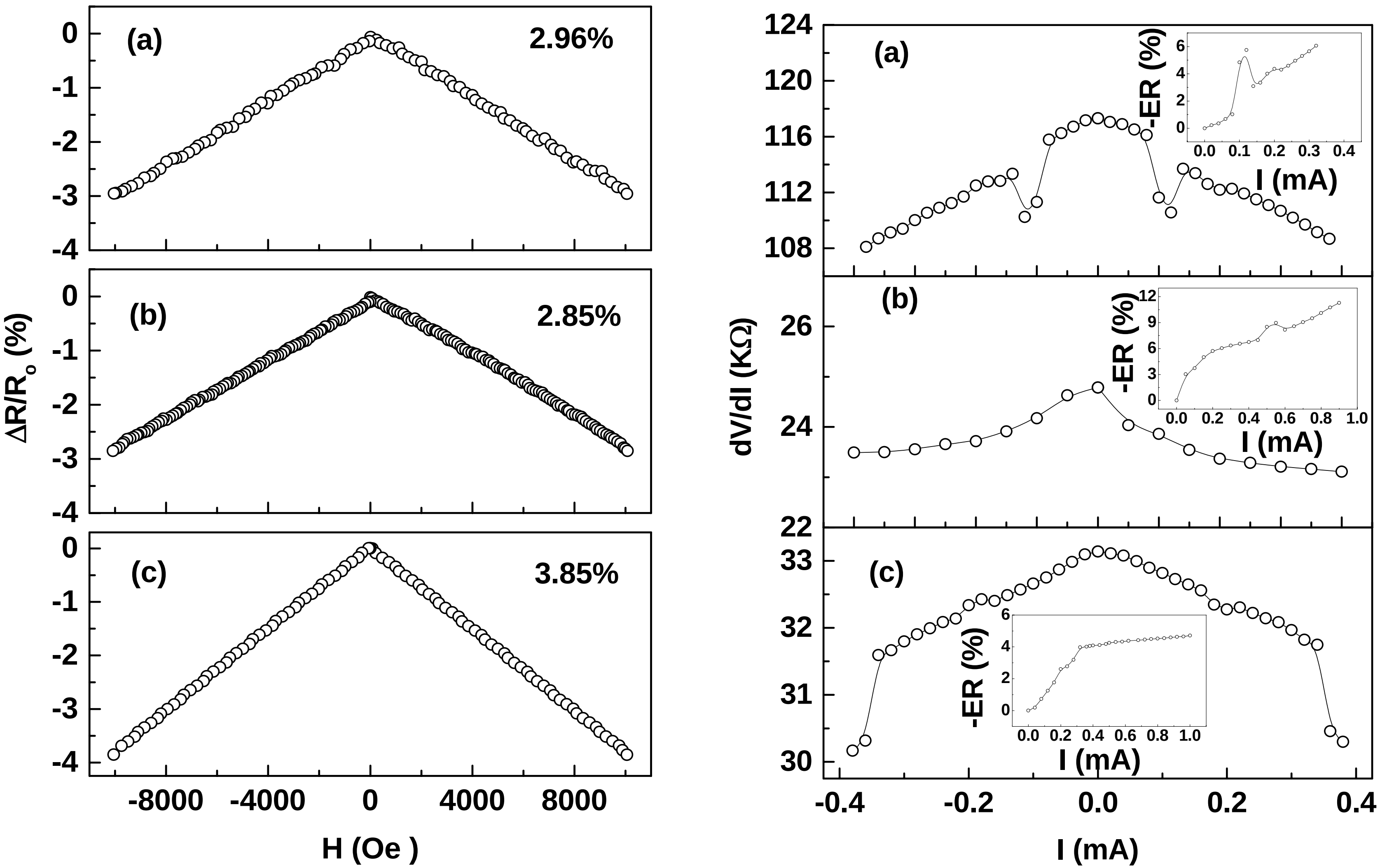


PHYSICAL REVIEW B **67**, 064412 (2003)

Current-induced giant electroresistance in La0.7Sr0.3MnO3 thin films

A. K. Debnath* and J. G. Lin†

Center for Condensed Matter Sciences, National Taiwan University, Taipei 10617, Taiwan

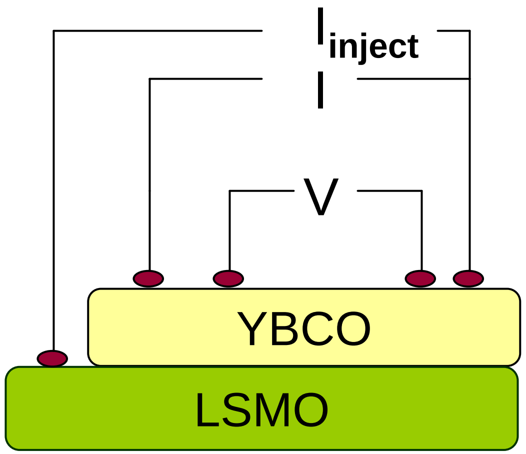
Current in nanowire of LSMO induced magnetic field, and the thermal energy delocalized the electrons.

thickness (nm)	MR_H (%)	MR_I (%)	ratio
60	2.96	5.6 (I = 0.3)	1.9
80	2.85	6.4 (I = 0.3)	2.3
		11.3 (I = 0.9)	4.0
100	3.85	3.5 (I = 0.3)	0.9
		4.6 (I = 0.9)	1.2

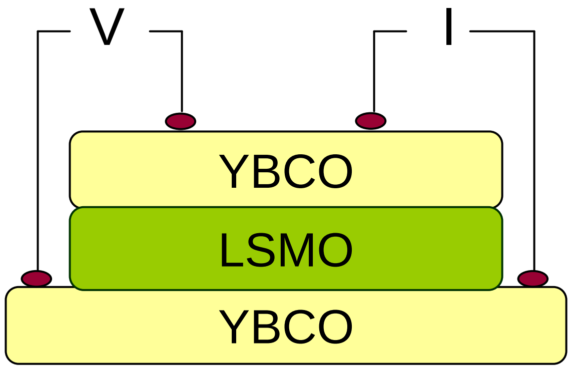
Subject (2): Nanocrystalline YBCO/LSMO bilayers

--- Proximity effect, spin injection & vortex pinning

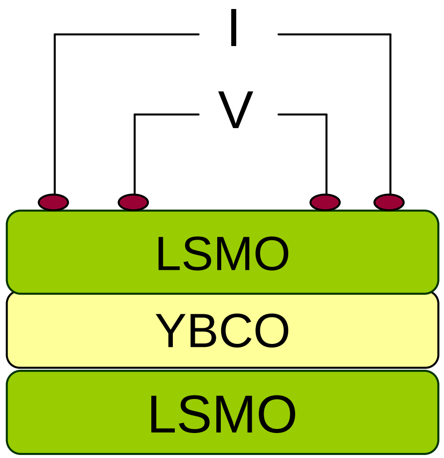
Multilayer devices



Spin-injection
I_c - depression

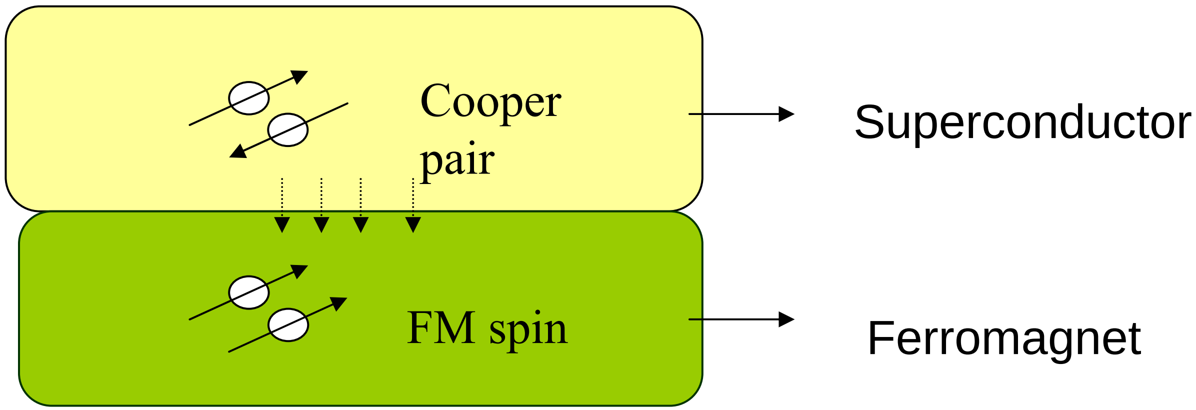


proximity effect
T_c - depression



FM/N/FM structure
MR enhancement

Interesting topics on YBCO/LSMO



1) **Proximity effects**
2) **Andreev reflection**
3) **Spin injection**
4) **Spin-accumulation**
5) π **–phase shift**

Proximity effect

1) Influence of magnetism on supercond.

Intermixing $\Rightarrow$ effective exchange field

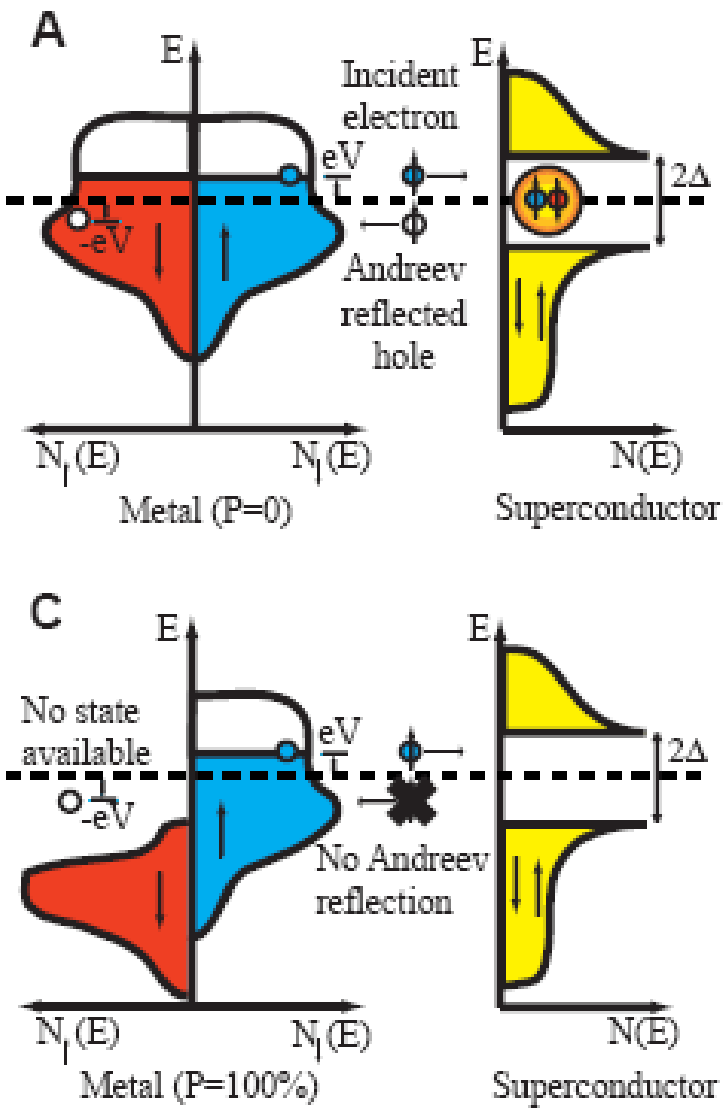
$H_{effect} = H_{ex}\,[d_F/(d_s+d_F)]$, **$T_c$ oscillation**

2) Influence of supercond. on magnetism

Intermixing $\Rightarrow$ reconstruction of magnetic order

T_{curie} oscillation

Spin-injection

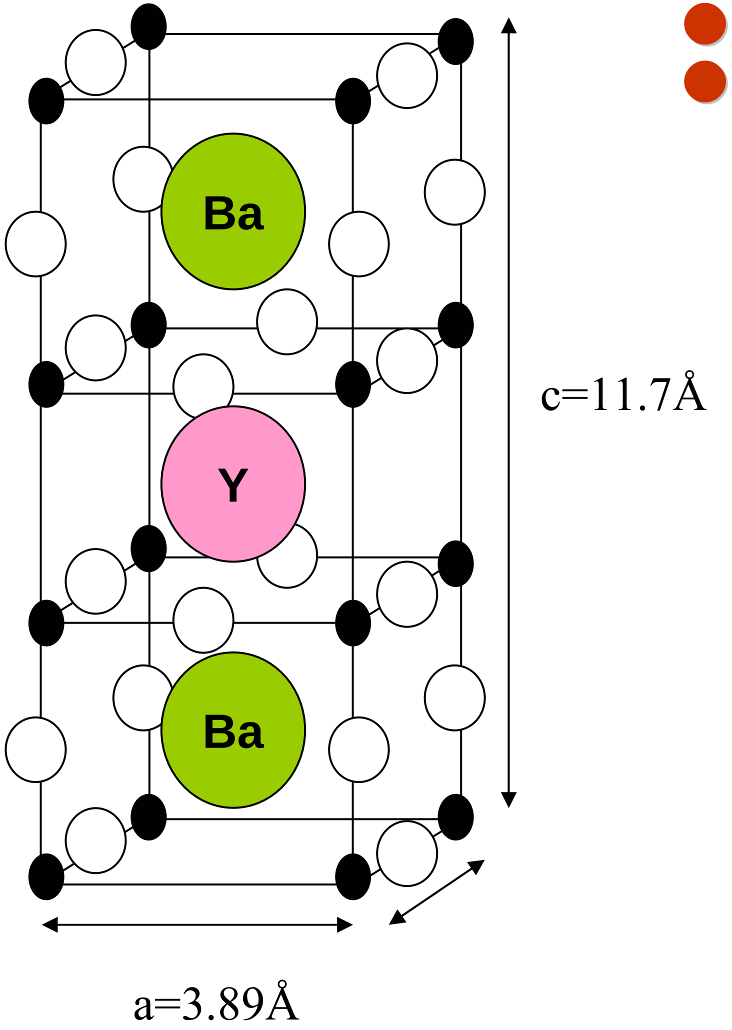


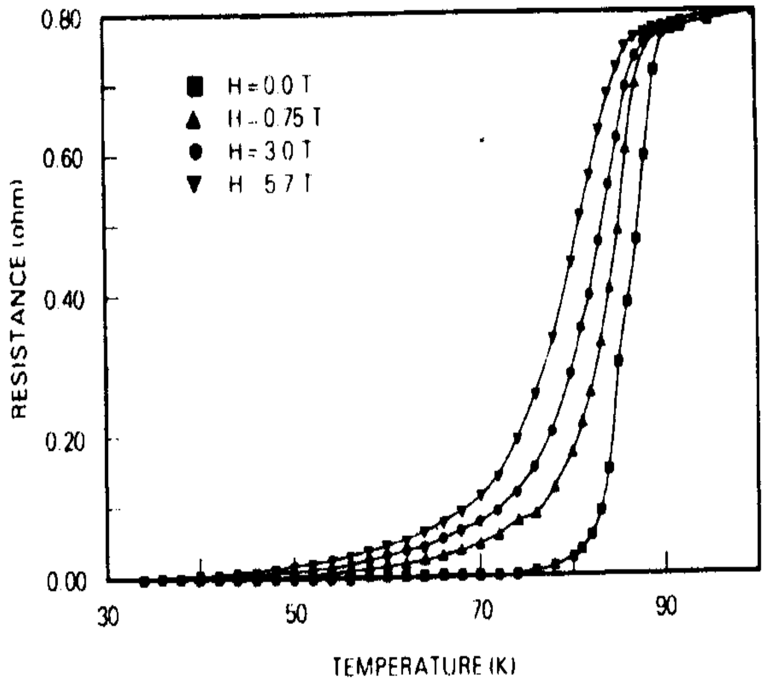
- Normal electron injected to superconductor will be reflected as a hole;
- Polarized electron injected into superconductor will kill a Cooper pair.

SCIENCE VOL 282 2 OCTOBER 1998

Superconductor: $RBa_2Cu_3O_7$, R = Y or rare earth element

- $YBa_2Cu_3O_7$ ξ_S ~ 3-10 nm
- Critical parameter T_c = 90 K, H_{c2} ~ 165 T
- SQUID, Bolometer, Filter, Resonator…

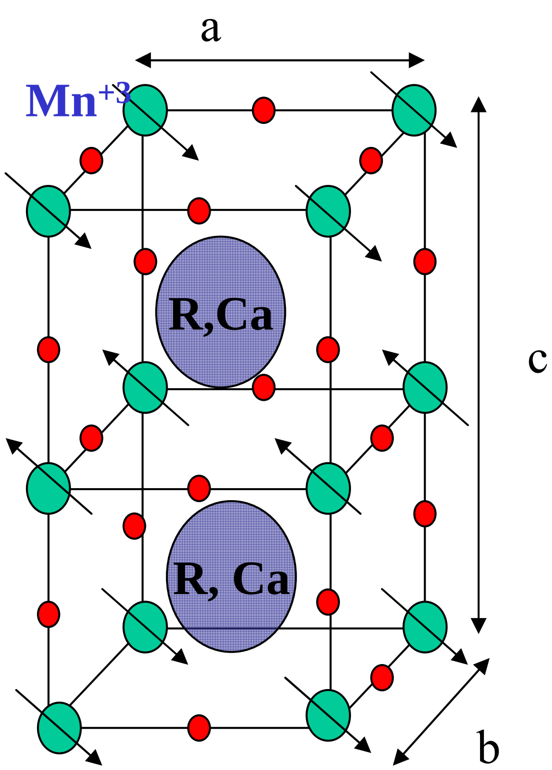


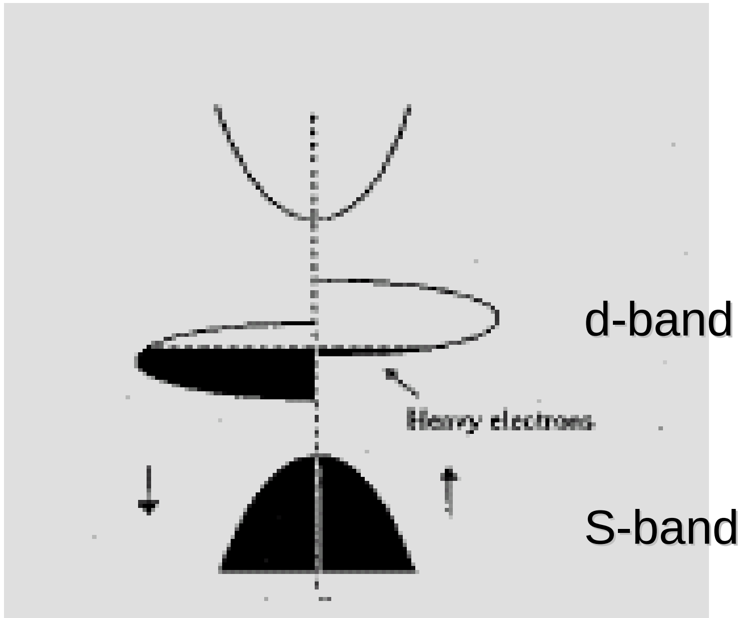


Wu, Chu et al., PRL (1987)

Half metal: (R,Ca)MnO_3, R = La, Nd, Pr….

- Ferromagnetic with 90 % polarization for LCMO. ξ_M ~ 10 -15 nm
- Colossal magnetoresistance (CMR)
- Sensor, pick-up head, MRAM





Jin et al, Science (1995)

APPLIED PHYSICS LETTERS VOLUME 81, NUMBER 24 9 DECEMBER 2002

Superconductivity depression in ultrathin $YBa_2Cu_3O_{7-\delta}$ layers in $La_{0.7}Ca_{0.3}MnO_3/YBa_2Cu_3O_{7-\delta}$ superlattices

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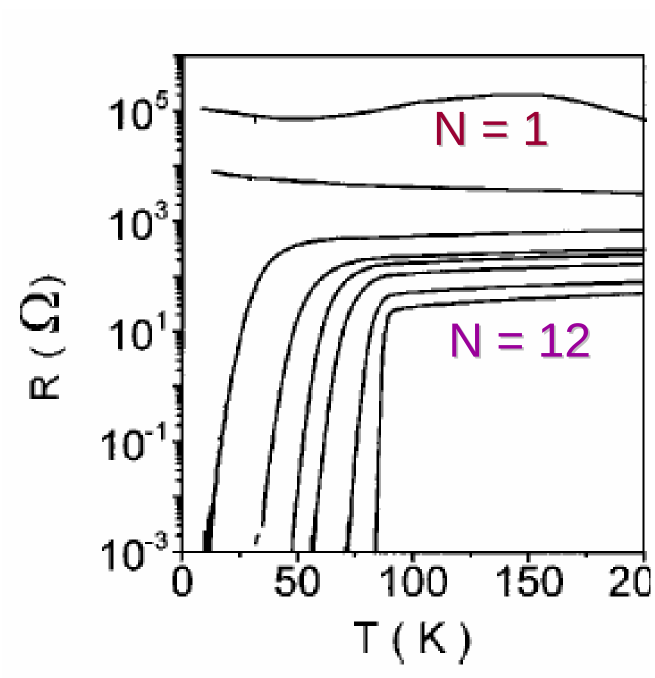

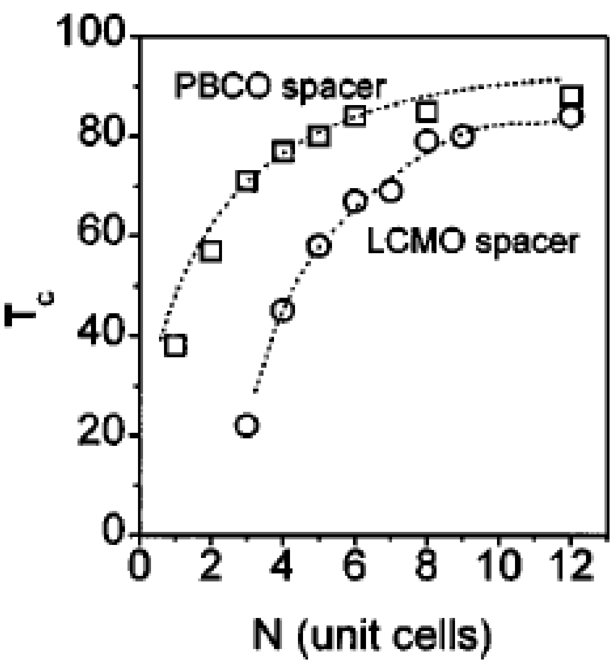

LCMO = 15 unit cell ~ 10 nm
N = unit cell of YBCO
N = 3 to 12
~ 3.5 to 14 nm
ΔTc ~ 62 to 2 K

$E_{eff\text{-}ex}$ ~ 0.74 to 0.42 of E_{ex}

JOURNAL OF APPLIED PHYSICS **98**, 023910 (2005)

Superconducting and transport properties of $YBa_2Cu_3O_7/La_{0.7}Sr_{0.3}MnO_3$ bilayers

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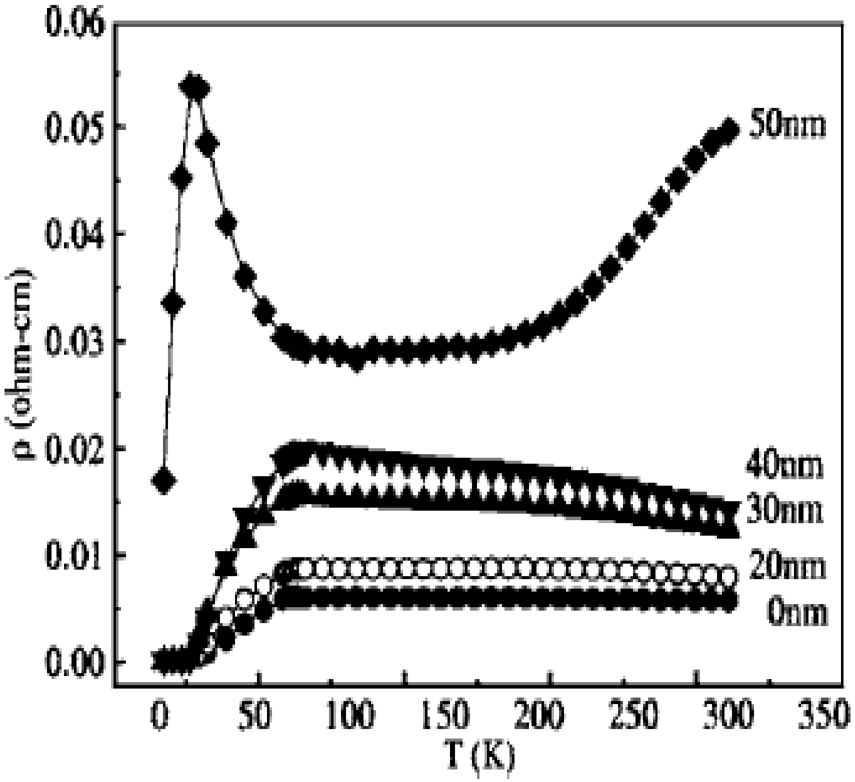
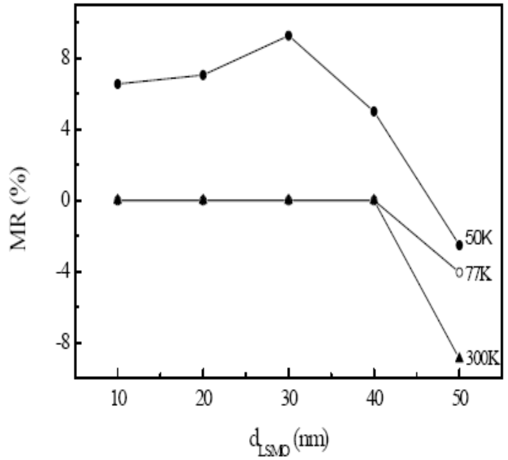
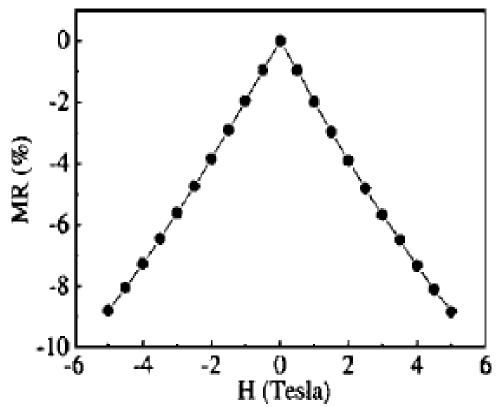
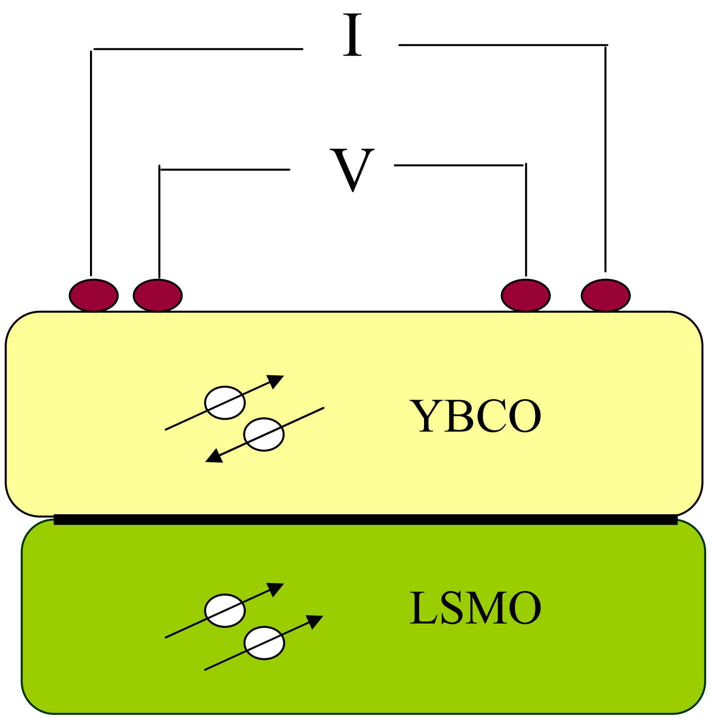
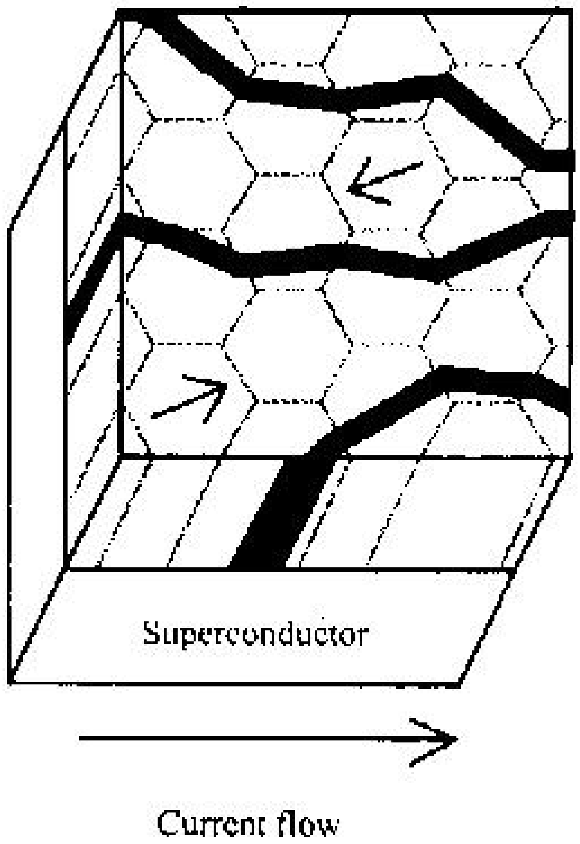




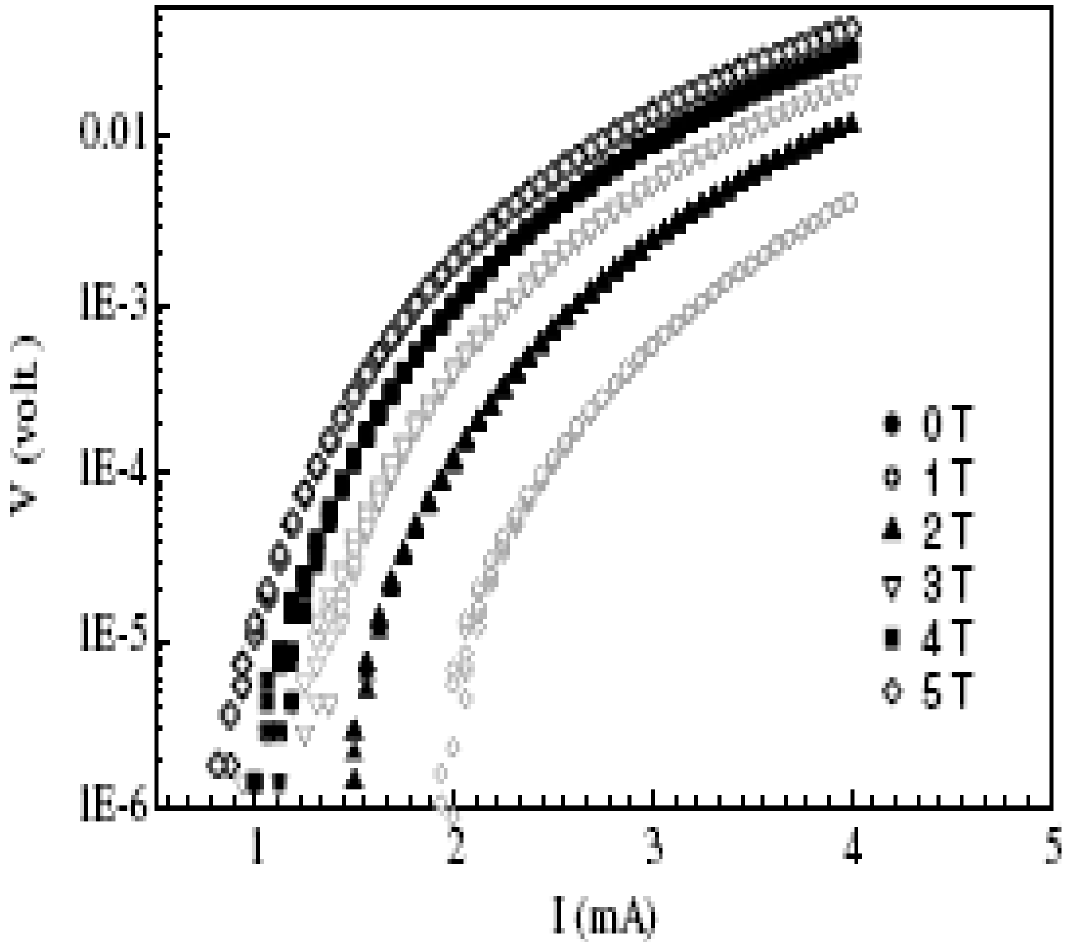


Vortex pining device





I-V characteristic



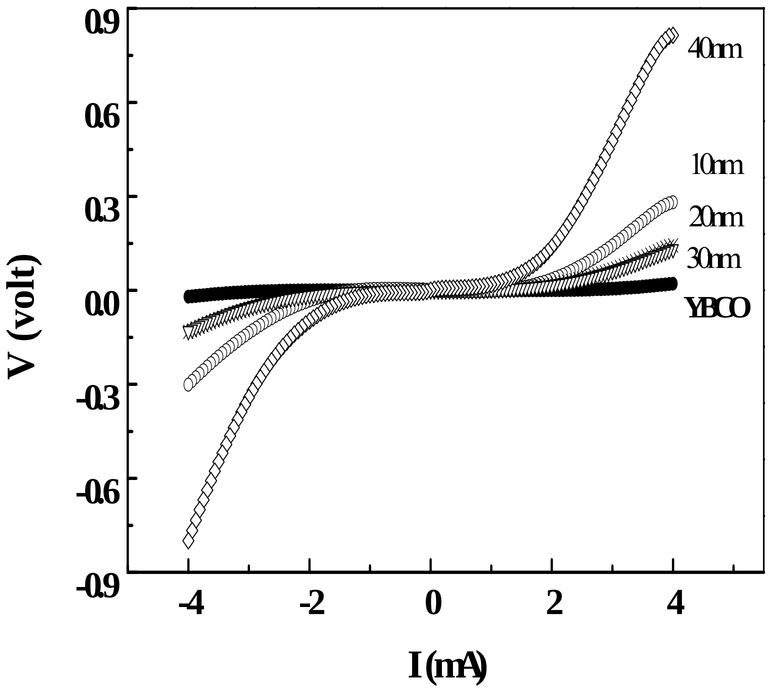
Critical current I_C
Normal state resistance Rn
Energy gap Δ

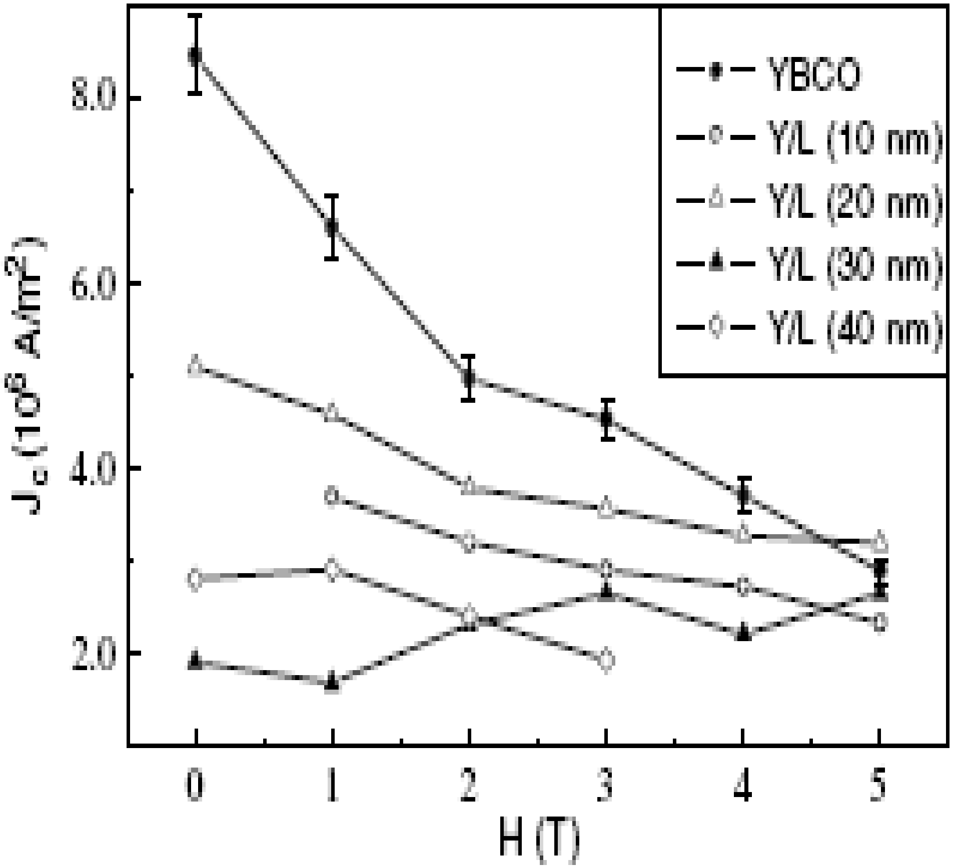
Ic = I(V=1 μV)

I_C Rn = $\pi\ \Delta / 2$

= $3.52\ \pi\ K_B\ T_C / 4$

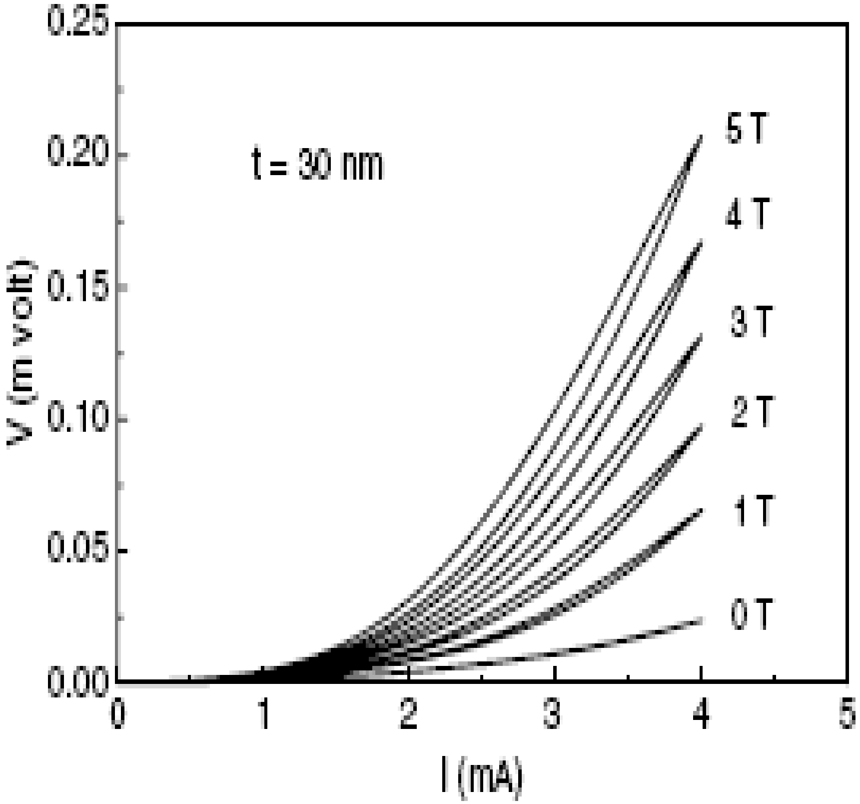
I- V Characteristic (1.9 K)

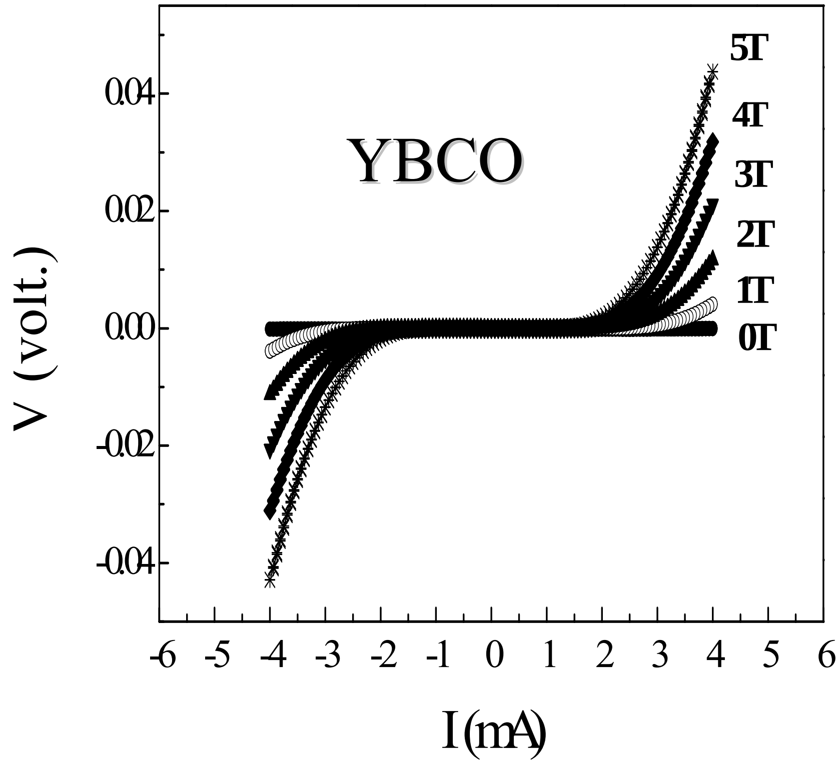




Ic decreases with increasing thickness of LSMO

Hysteresis in YBCO/LSMO(30nm)





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Physica C 437–438 (2006) 187–189

PHYSICA C

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High field pinning-effects in $YBa_2Cu_3O_7/La_{0.7}Sr_{0.3}MnO_3$ nanocrystalline bilayers

J.G. Lin [a,b,*], S.L. Cheng [b]

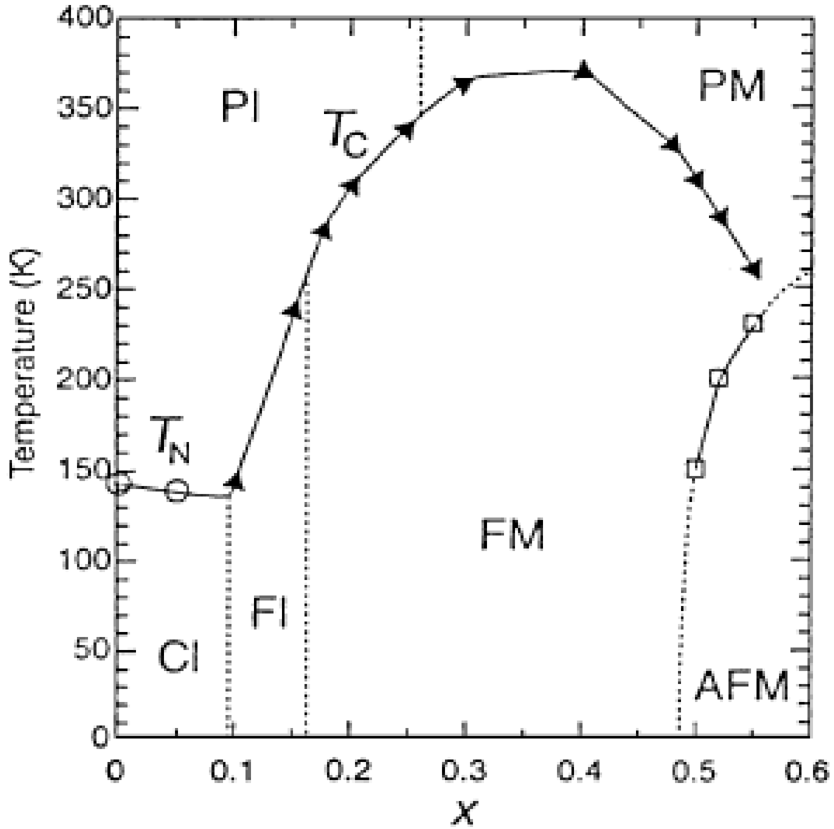
[a] Center for Condensed Matter Sciences, National Taiwan University, Taipei 106, Taiwan
[b] Center for Nanostorage Research, National Taiwan University, Taipei 106, Taiwan

Available online 26 January 2006

Subject (3): Epitaxial YBCO/NCMO bilayers

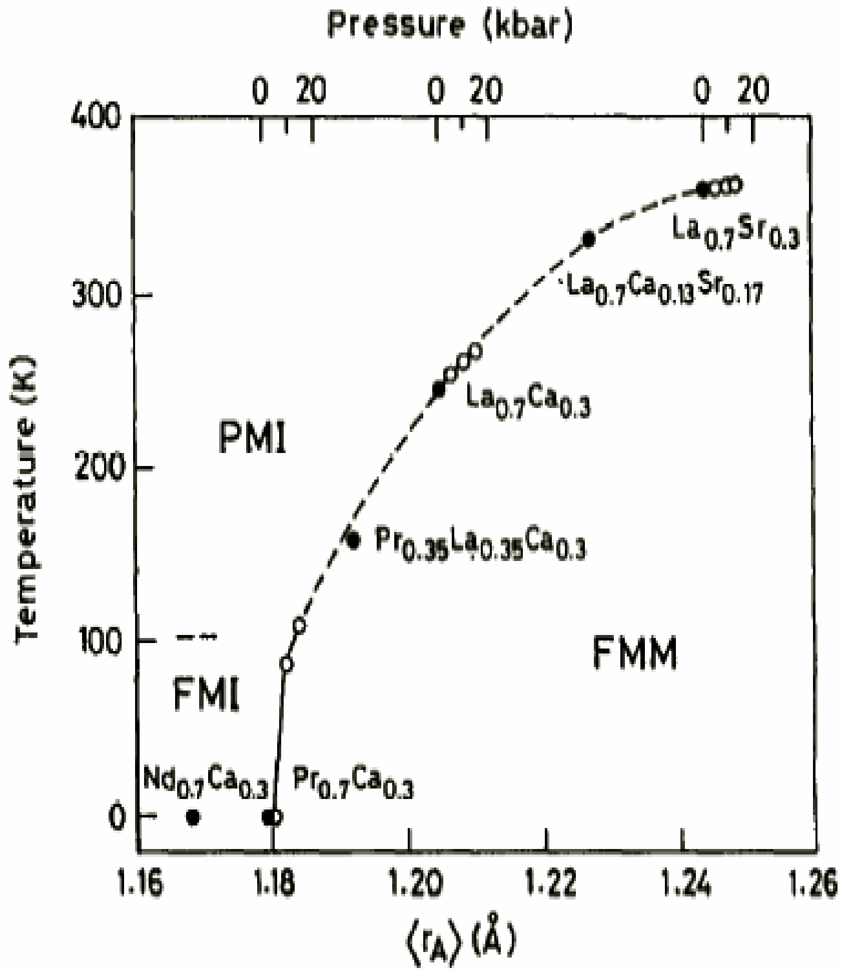
--- **How the superconductivity is affected by weak ferromagnetic insulator ?**

$La_{1-x}Sr_xMnO_3$



E. Dagotto *et. al.*, Physics Reports 344, 1-153 (2001)

$R_{0.7}(Ca,Sr)_{0.3}MnO_3$



C. N. R. Rao et al. J. Phys. Chem. Solids 59, 487 (1998)

Low-current-induced electrical hysteresis in $Nd_{0.7}Ca_{0.3}MnO_3$

Daniel Hsu
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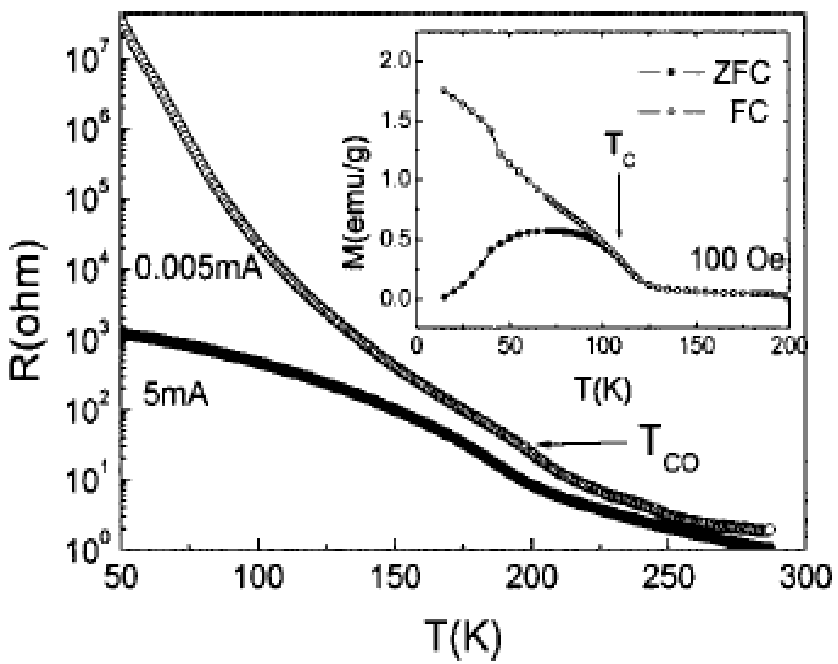
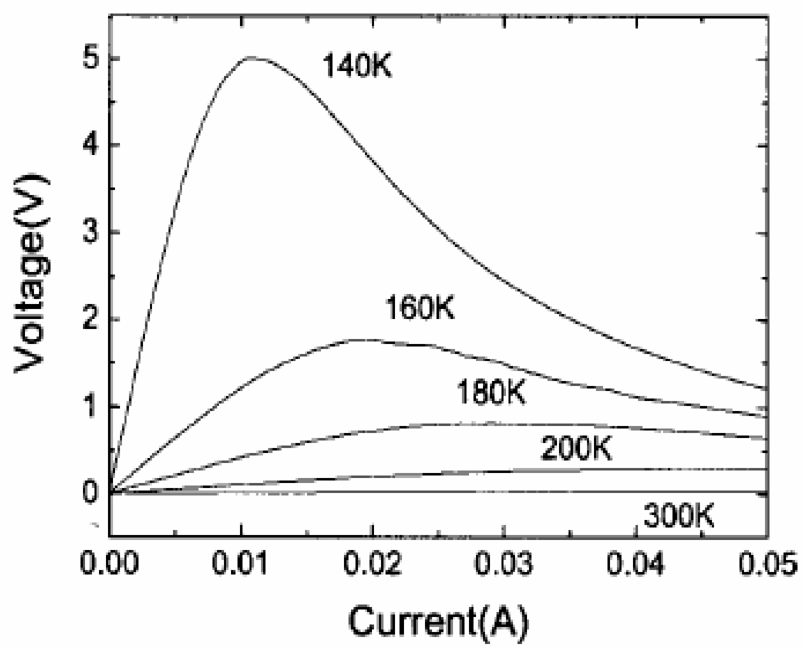
J. G. Lin[a)]
Center for Condensed Matter Science, National Taiwan University, Taipei 106, Taiwan and Center for Nanostorage Research, National Taiwan University, Taipei 106, Taiwan

W. F. Wu
Institute of Mechanical Engineering, National Taiwan University, Taipei 106, Taiwan





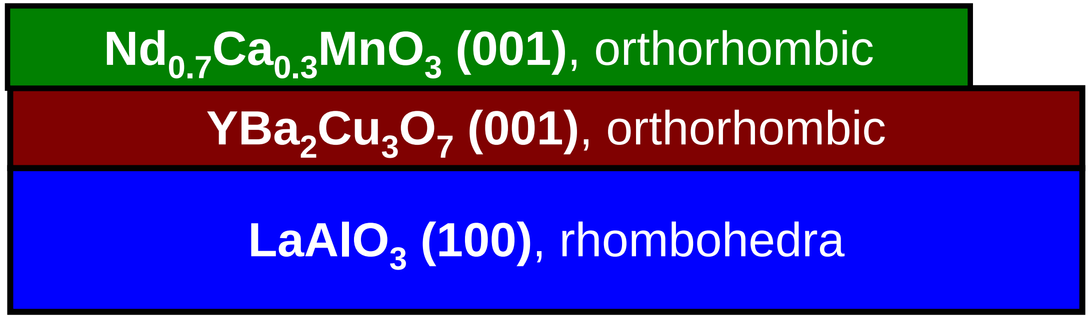
Substrate criteria

1. Atomically flat
2. Chemically comparable
3. Well lattice mismatch

   For YBCO, $LaAlO_3$ (100) is ~ 0.4%;

   $SrTiO_3$(100) is 0.2% for *a*-parameter, 0.3% for *b*-parameter;

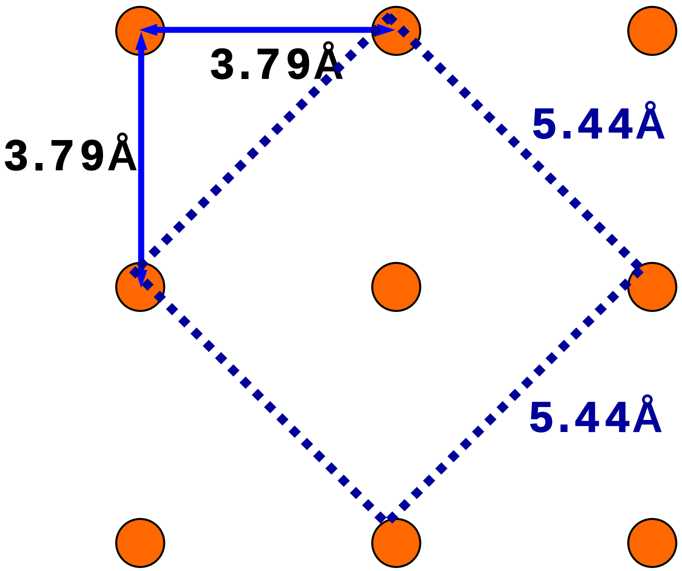
   $NdGaO_3$ (110) is 0.7 % for *a*- and *b*-parameters

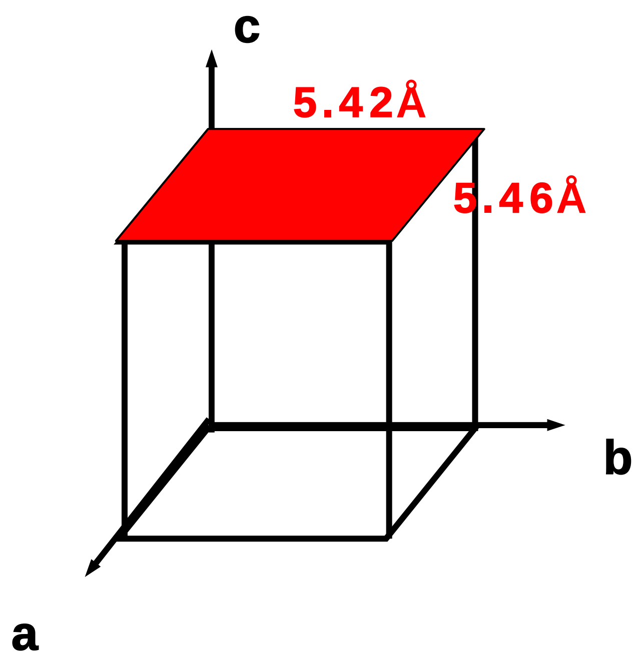


$\underline{LaAlO_3}$
a=b=c=3.79 Å
$\alpha=\beta=\gamma=90.12^o$

$\underline{o\text{-}YBa_2Cu_3O_7}$
a=3.82 Å, b=3.85 Å,
c=11.63 Å; $\alpha=\beta=\gamma=90^o$

$\underline{o\text{-}Nd_{0.7}Ca_{0.3}MnO_3}$
a=5.42 Å, b=5.46 Å,
c=7.72 Å;$\alpha=\beta=\gamma=$90o

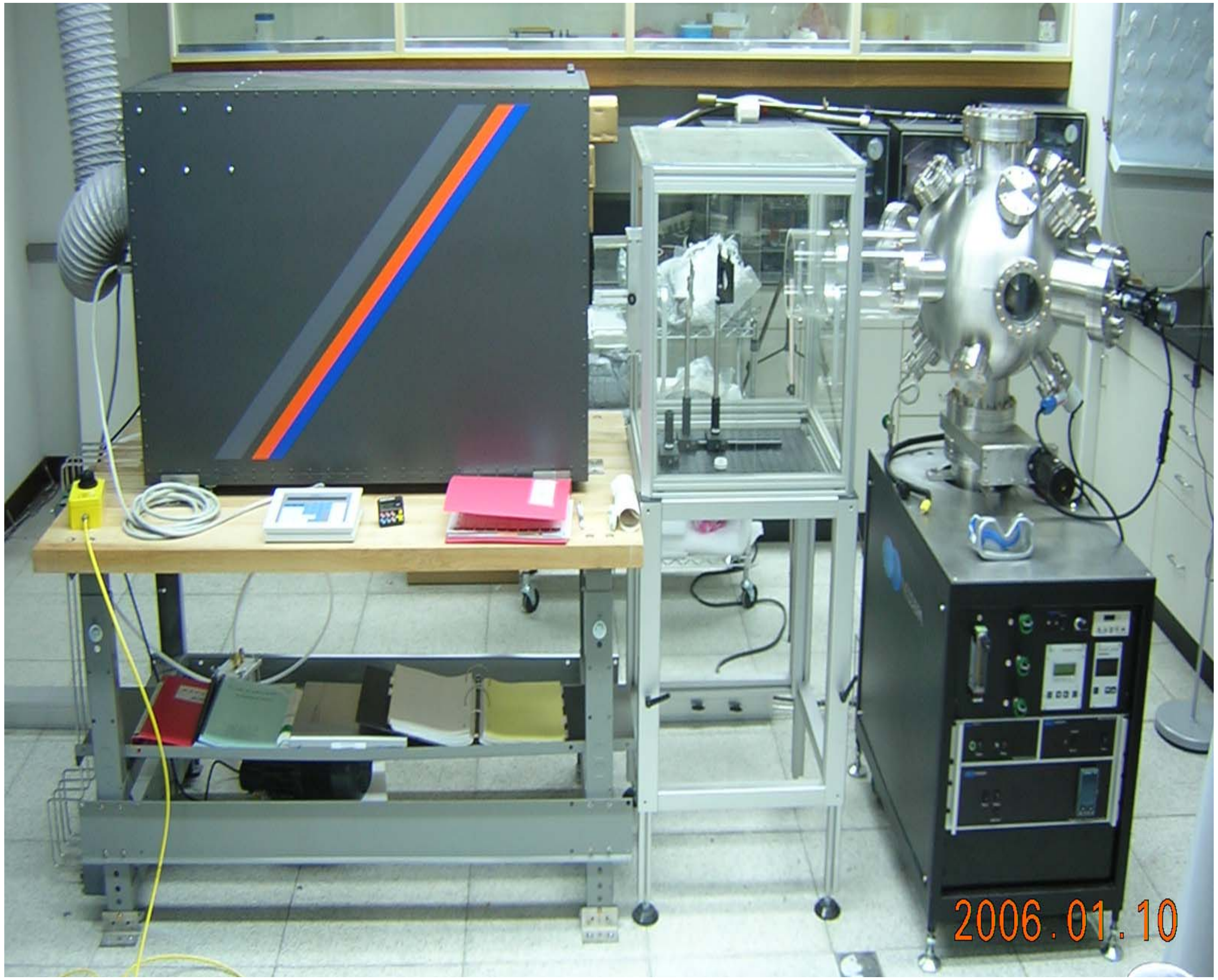




Pulse-Laser-Deposition system –Neocera 180

KrF Laser (248 nm)

Focus Len

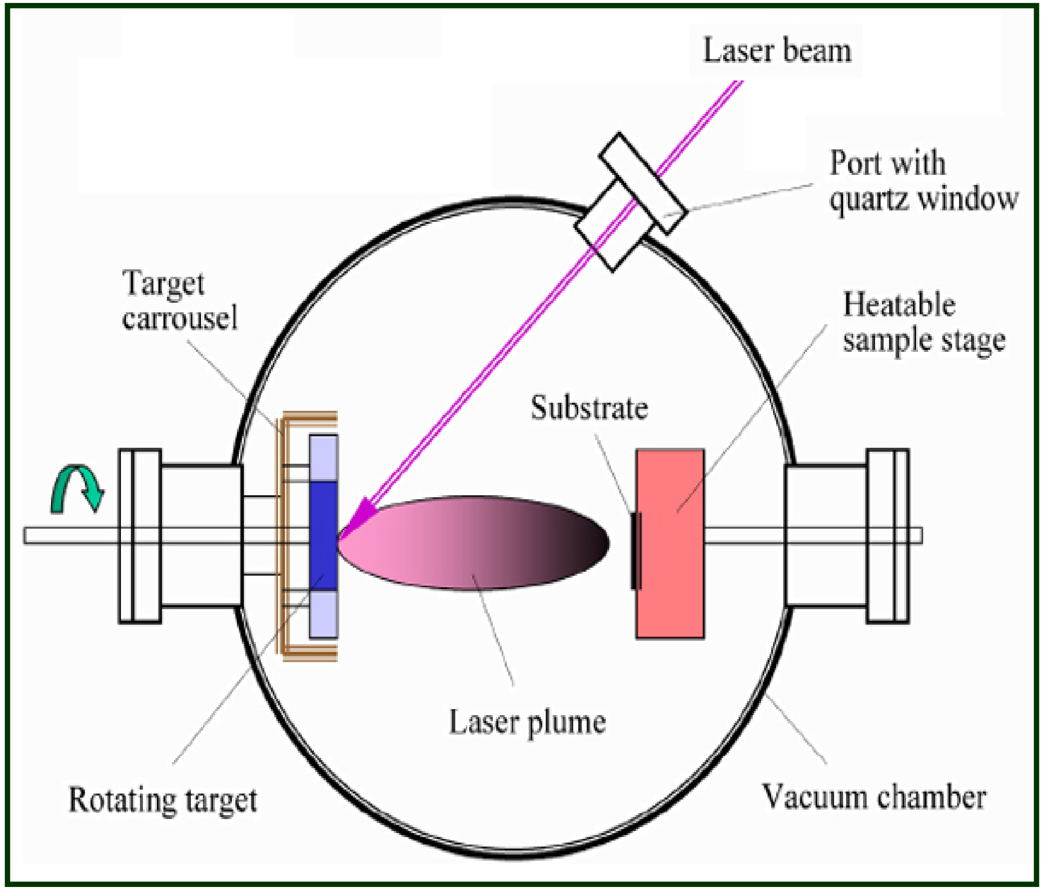


Chamber

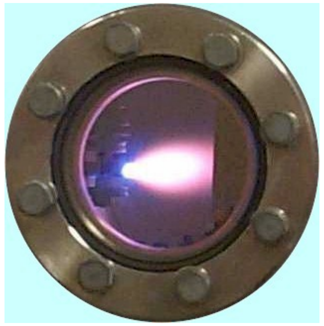
10^{-8} torr

Pump
Flow meter
Vacuum gauge

Pulse-Laser Deposition system (II) - Chamber



YBCO

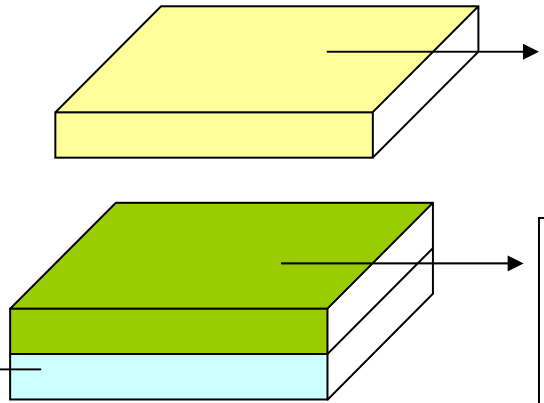


NCMO

NCMO/YBCO heterostructure

$Nd_{0.7}Ca_{0.3}MnO_3$, (40 -350 nm)

780°C for 1 hour in 50 mtorr O_2



$YBa_2Cu_3O_7$(100-600 nm)

850 °C for 1 hour

400 °C for 6 hours

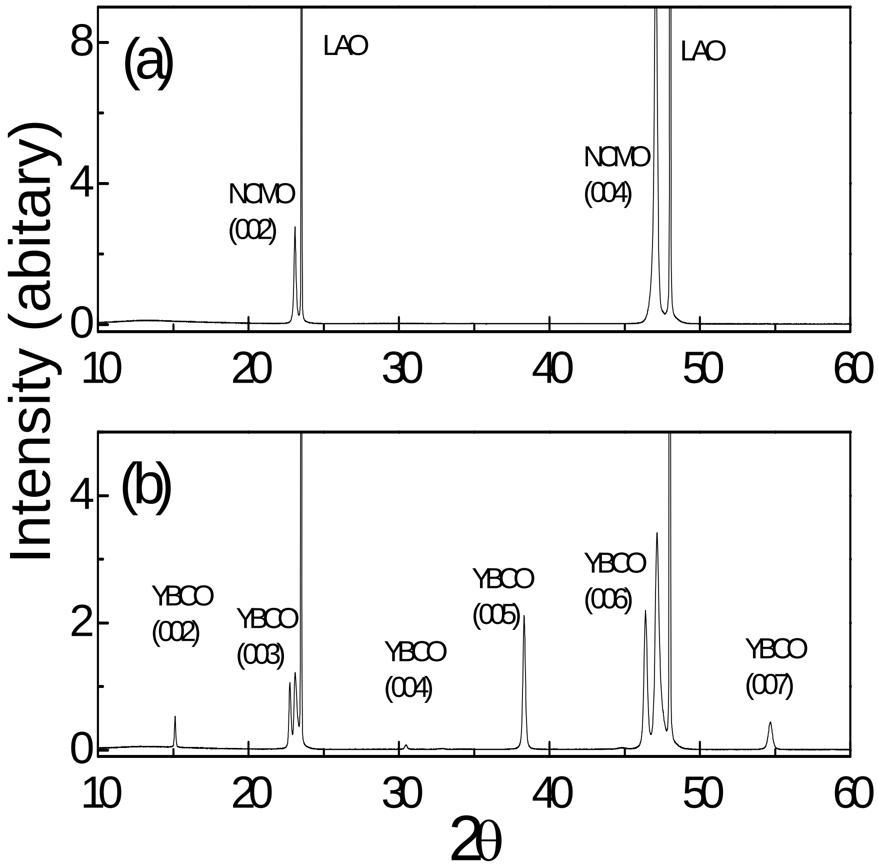
in 300 mtorr O_2

$LaAlO_3$

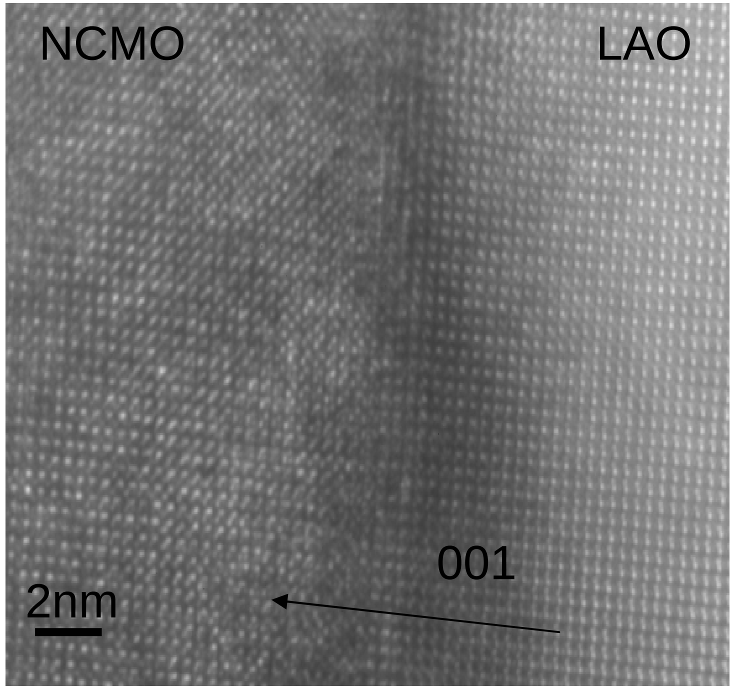
(100)

X-ray Diffraction Patterns & TEM image

- Both NCMO & YBCO are with c-axis perpendicular to the film surface

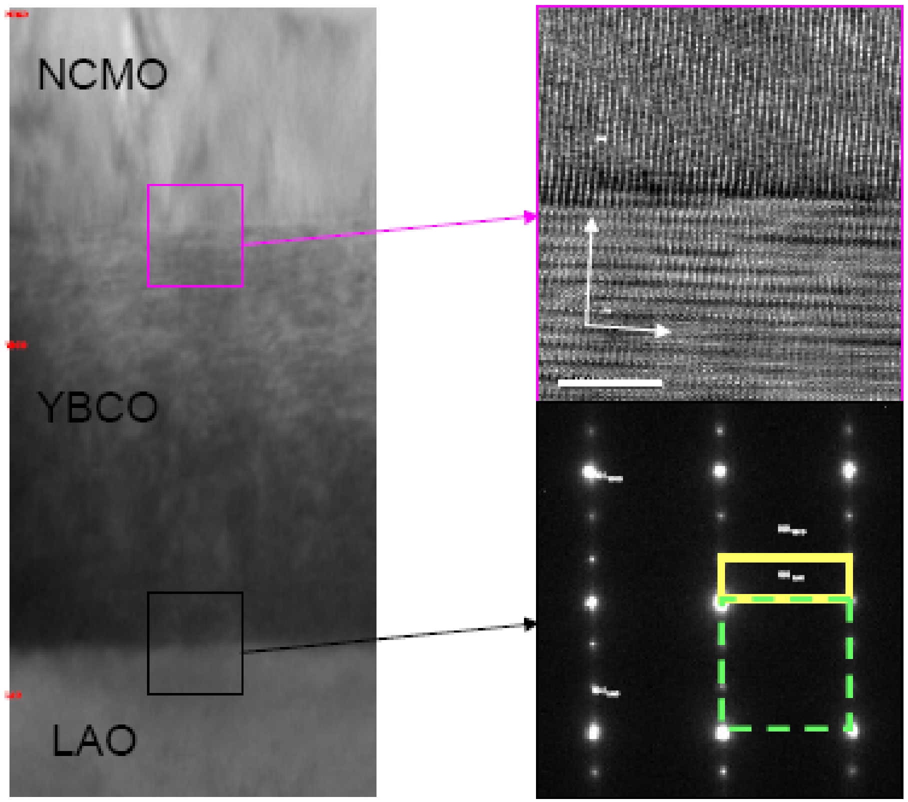


- Epitaxial growth along [001]

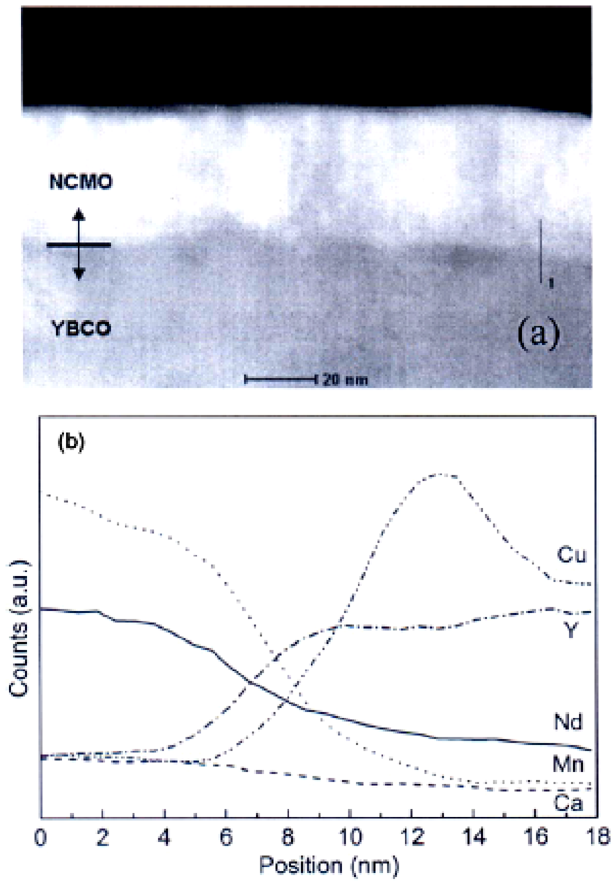


Basic characterization

(High resolution TEM & electron diffractions)

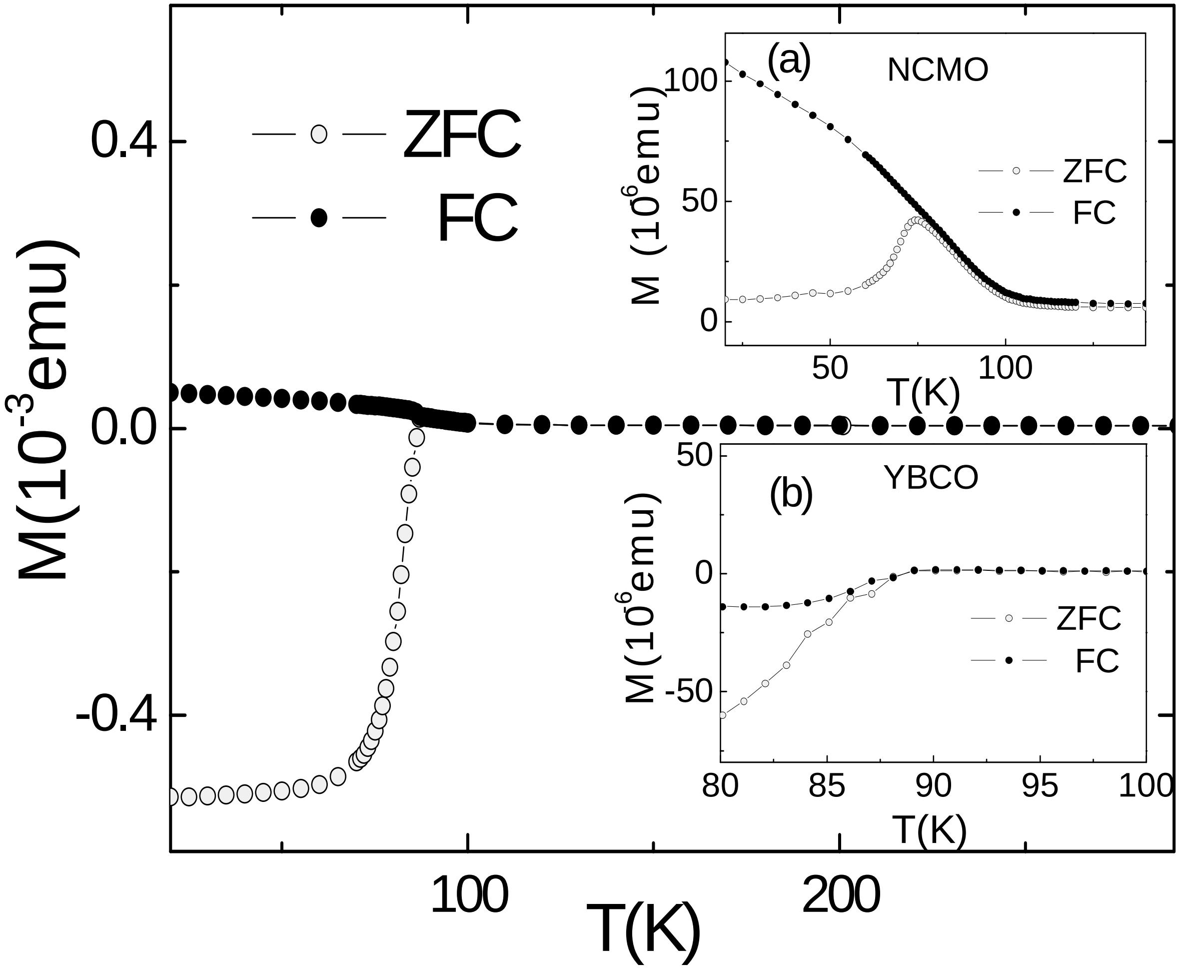


(Scanning EDX)



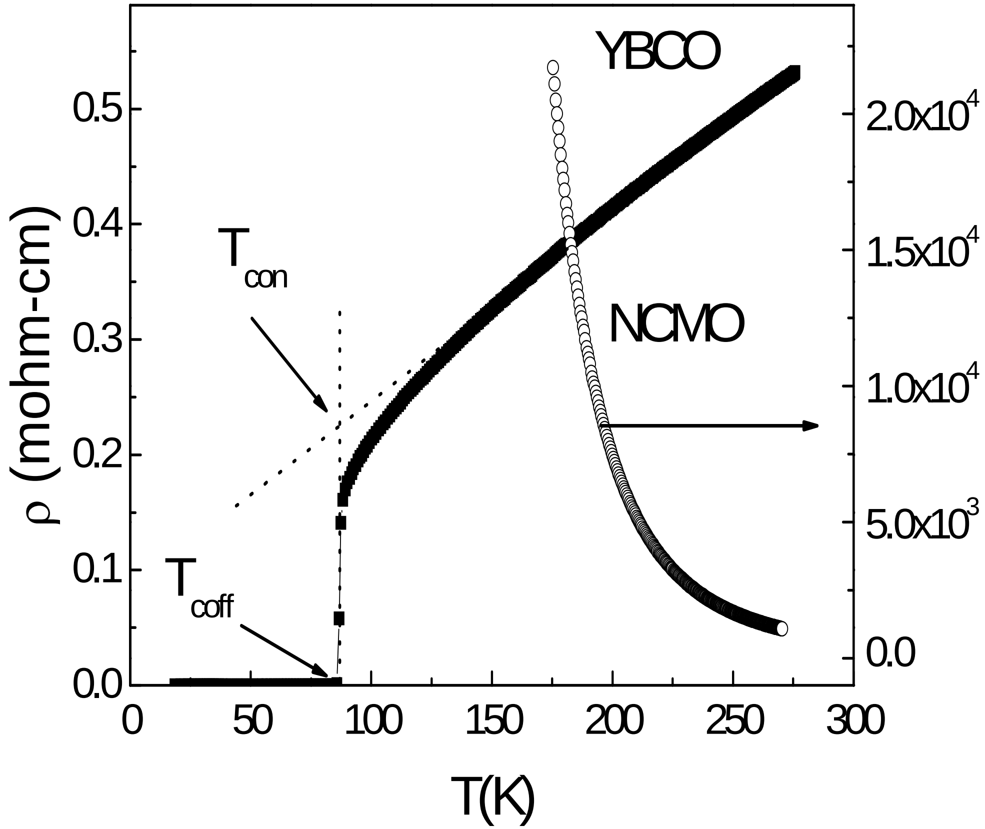
Magnetization(M) vs. Temperature (T) – by SQUID

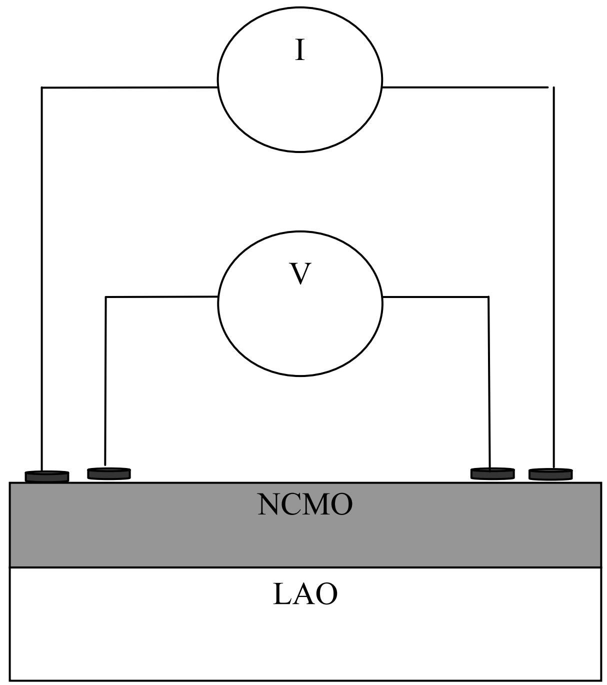
- T_c is 88 K for YBCO; T_N is ~ 75 K for NCMO; Both transition temperature do not change in NCMO(40nm)/YBCO(160nm).



4-Probe method

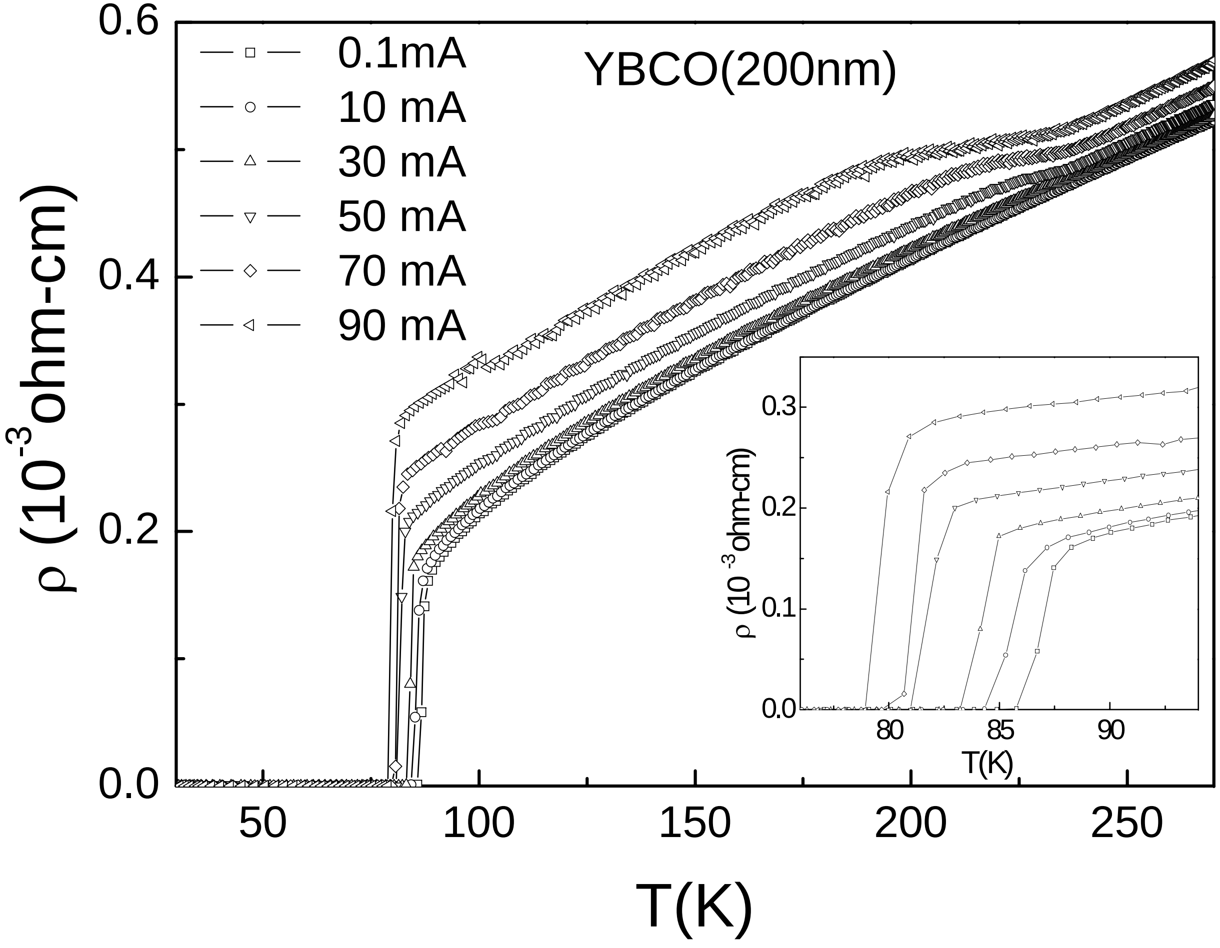
- YBCO : T_c ~ 88 K & ΔT < 2 K; NCMO: Insulating





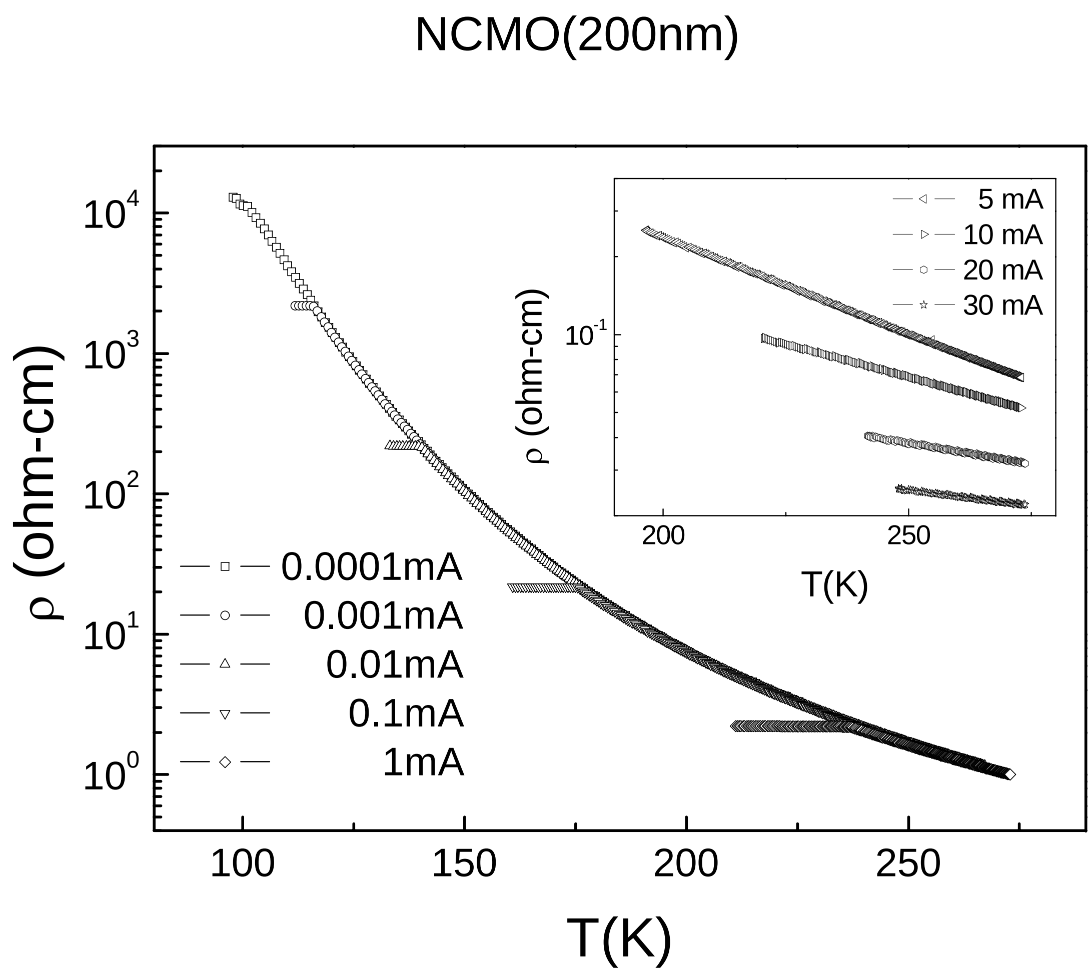
Resistivity (ρ) vs. Temperature (T)

- For I_a = 1 – 90 mA, normal state increases & T_c drops with a rate 0.1K/mA.
- open a gap at 230 K ?



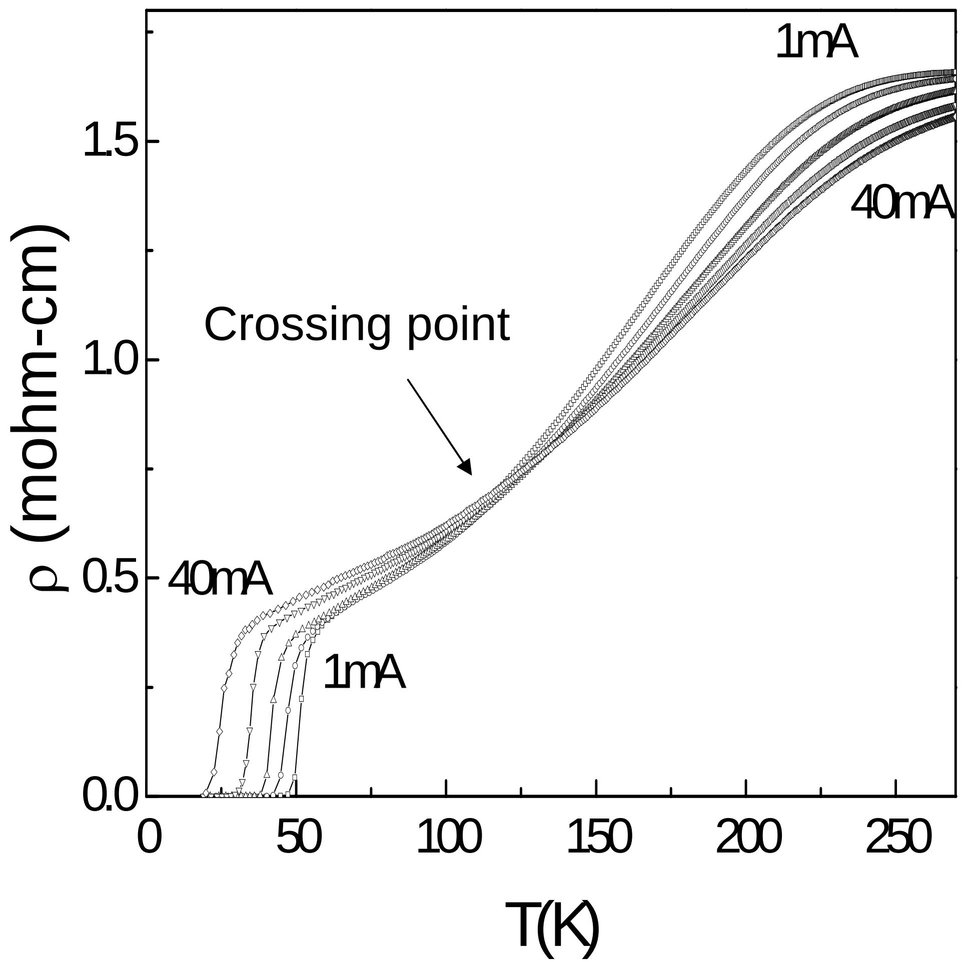
Resistivity (ρ) vs. Temperature (T)

- For I_a = 5 – 30 mA, resistivity decreases.

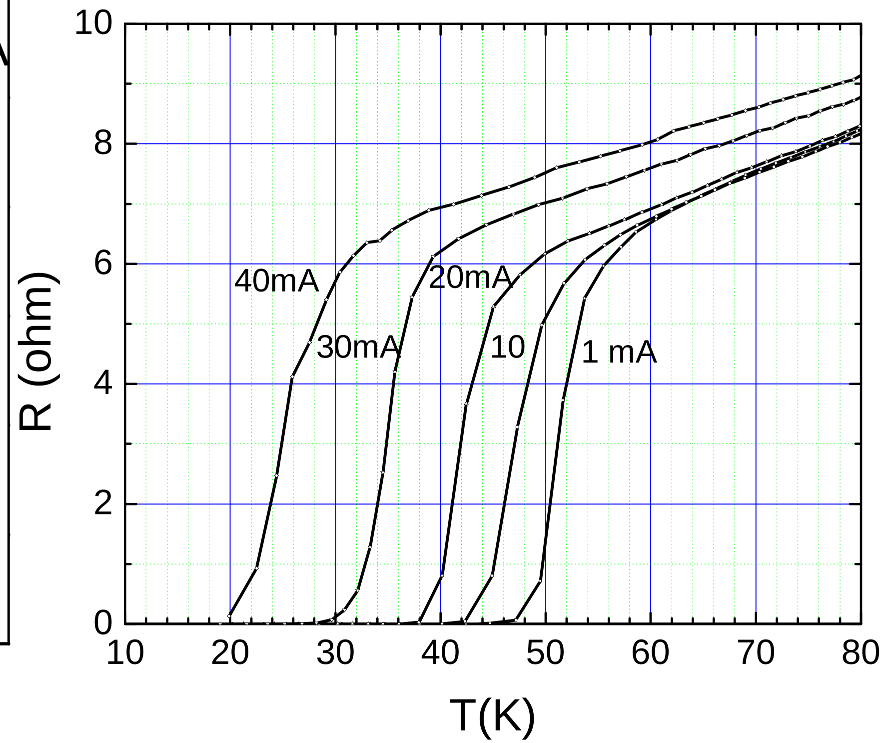


Resistivity (ρ) vs. Temperature (T)

NCMO(200nm)/YBCO(200nm)



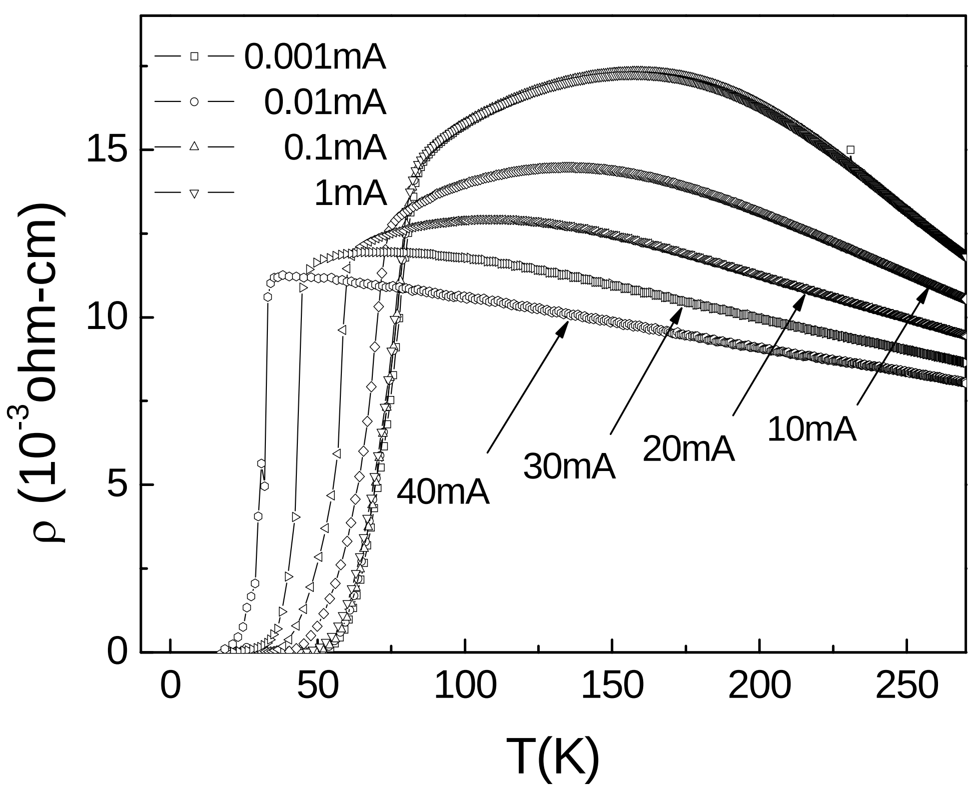
- For I_a = 1 – 40 mA, normal state increases & T_c drops with a rate of 1.0 K/mA.

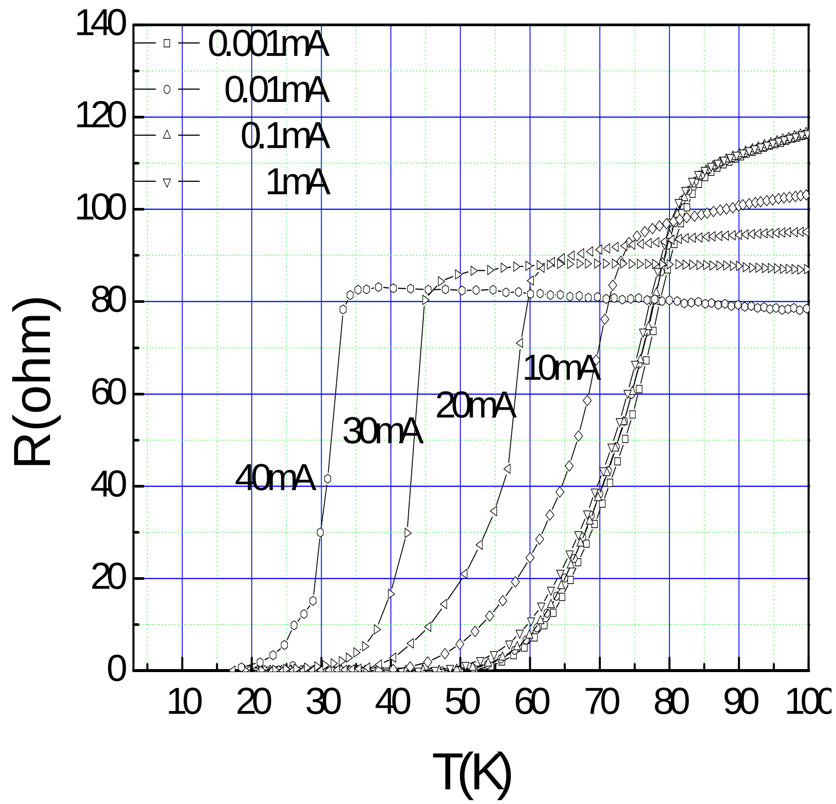


Resistivity (ρ) vs. Temperature (T)

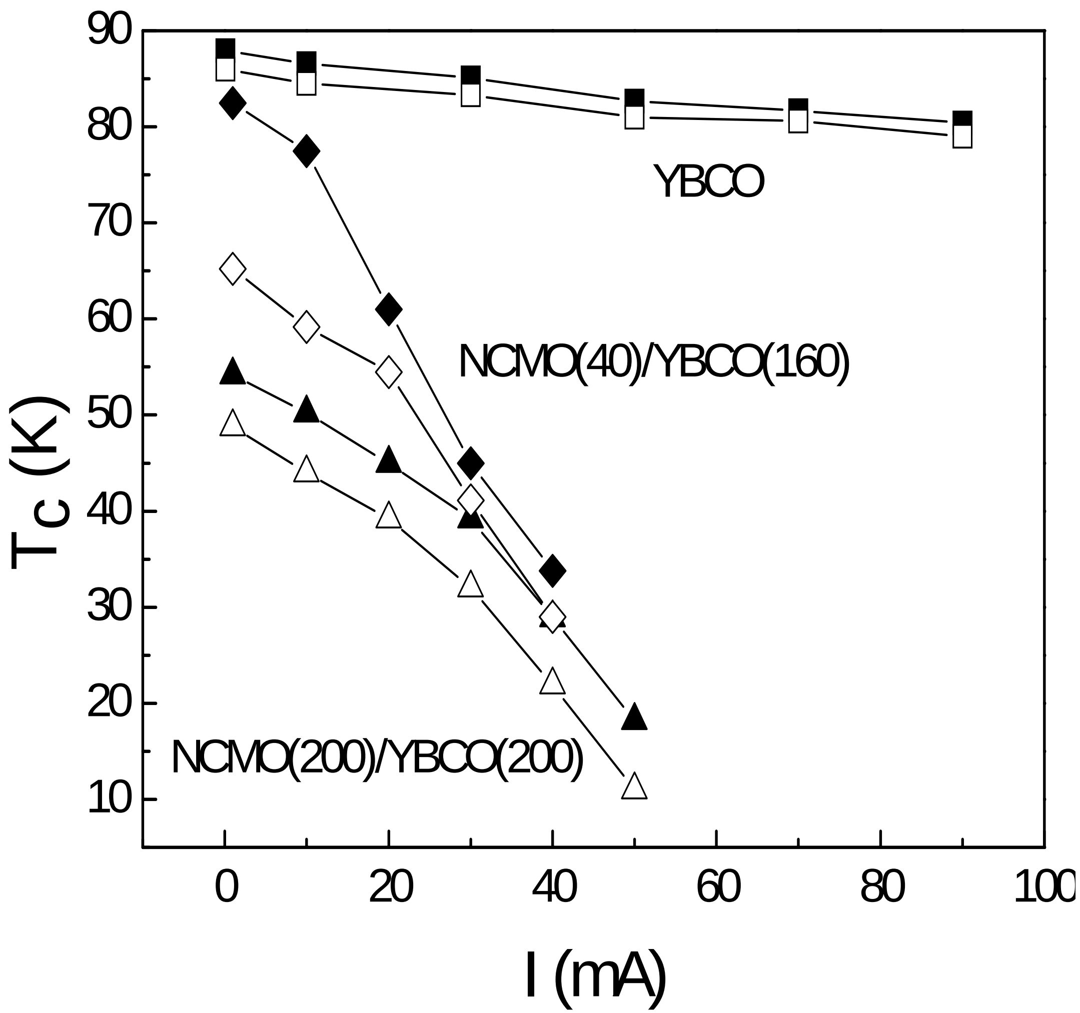
- For I_a = 1 – 40 mA, normal state decreases 30% & T_c drops with a rate of 1.4 K/mA.

NCMO(40nm)/YBCO(160nm)





Superconducting temperature vs. applying current



- T_c is suppressed by proximity in NCMO/YBCO at I < 1 mA.
- T_c-suppresion rate at I > 1 mA is one order higher in bilayer due to the spin-injection.

Proximity effect

$$H_{eff\text{-}ex} = H_{ex} \{d_F/(dS + d_F)\}$$

$0.5H_{ex}$ for NCMO(200nm)/YBCO(200nm)
$0.2H_{ex}$ for NCMO(40nm)/YBCO(160nm)

ΔT_c ~ 34K / 16K

Current induces spin-injection effect

Observe a threshold of effective current I > 1 mA

Large T_c-suppresion in a rate of 1.4 K/mA.

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Proximity effect of superconductivity and magnetism in the $Nd_{0.7}Ca_{0.3}MnO_3/YBa_2Cu_3O_7$ bilayer

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Current enhanced magnetic proximity in $Nd_{0.7}Ca_{0.3}MnO_3/YBa_2Cu_3O_7$ bilayer

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Thickness dependent spin-injection effects in $Nd_{0.7}Ca_{0.3}MnO_3/YBa_2Cu_3O_7$ bilayers

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Two $Nd_{0.7}Ca_{0.3}MnO_3/YBa_2Cu_3O_7$ (NCMO/YBCO) bilayers with different thickness ratios are fabricated and the spin-injection effects are investigated. The NCMO/YBCO samples have thicknesses of 100 nm/200 nm and 200 nm/200 nm, which are denoted as N/Y(1) and N/Y(2), respectively. It is shown that the current-induced suppression rate of superconducting transition temperature (dT_c/dI) in YBCO is enhanced by four to six times of magnitude in N/Y(1) and N/Y(2) compared with that in pure YBCO. Furthermore, dT_c/dI in N/Y(2) is larger than that in N/Y(1), which suggests that the thickness of NCMO has influence on the pair breaking in YBCO.
 [DOI: 10.1063/1.2833819]

Half metal is the future material for all kinds of spintronic devices !!!